

Arrays of indefinitely long uniform nanowires and nanotubes

Mecit Yaman^{1,2}, Tural Khudiyev^{1,2}, Erol Ozgur^{1,2}, Mehmet Kanik^{1,2}, Ozan Aktas^{1,3}, Ekin O. Ozgur^{1,2}, Hakan Deniz^{1,2}, Enes Korkut^{1,2} and Mehmet Bayindir^{1,2,3*}

Nanowires are arguably the most studied nanomaterial model to make functional devices and arrays^{1,2}. Although there is remarkable maturity in the chemical synthesis of complex nanowire structures^{3,4}, their integration and interfacing to macro systems with high yields and repeatability^{5–7} still require elaborate aligning, positioning and interfacing and post-synthesis techniques^{8,9}. Top-down fabrication methods for nanowire production, such as lithography and electrospinning, have not enjoyed comparable growth. Here we report a new thermal size-reduction process to produce well-ordered, globally oriented, indefinitely long nanowire and nanotube arrays with different materials. The new technique involves iterative co-drawing of hermetically sealed multimaterials in compatible polymer matrices similar to fibre drawing. Globally oriented, endlessly parallel, axially and radially uniform semiconducting and piezoelectric nanowire and nanotube arrays hundreds of metres long, with nanowire diameters less than 15 nm, are obtained. The resulting nanostructures are sealed inside a flexible substrate, facilitating the handling of and electrical contacting to the nanowires. Inexpensive, high-throughput, multimaterial nanowire arrays pave the way for applications including nanowire-based large-area flexible sensor platforms, phase-change memory, nanostructure-enhanced photovoltaics, semiconductor nanophotonics, dielectric metamaterials, linear and nonlinear photonics and nanowire-enabled high-performance composites.

One-dimensional nanostructures such as nanowires, nanotubes and nanoribbons are continuing to be at the forefront of nanoscience and nanotechnology. Two distinct approaches for the fabrication of these structures are the top-down and bottom-up philosophies¹⁰. During the past decade, chemical synthesis demonstrated the impressive success of the bottom-up approach in achieving controllable composition and morphology^{3,4} and prototype functional devices^{1,2,10}, but less so for the integration of high-density device assemblies. Post-synthesis assembly techniques, designed for this end, such as electric¹¹, magnetic-field-assisted alignment and dielectrophoresis⁵, optical and optoelectronic tweezers¹², laminar flow in microfluidic channels¹³ and micro contact printing¹⁴ still do not achieve low cost, high throughput, and good reproducibility with high-precision addressability⁸. For some applications chemical synthesis is simply not a fitting choice; for example, it is still difficult to produce very long aligned nanowires^{15–17}, and to obtain nanowires on large area substrates¹⁸, or polymer substrate nanowire integration¹⁹, requires separate high-temperature nanowire synthesis and subsequent transformation.

On the top-down side, patterning by lithography is a powerful but costly micro- and nanofabrication tool perfected

by the semiconductor industry. However, for one-dimensional nanostructure fabrication, it is not much favoured owing to scientific challenges such as resolution, surface roughness, limited chemical composition and its low throughput¹⁰. Various hybrid techniques are developed for nanowire fabrication to expand the toolkit, such as using photonic crystal fibres²⁰ as microfluidic reactors¹⁶ to synthesize nanowires (templating) and meniscus-controlled solution evaporation¹⁷ (growth and winding process) to obtain very long nanowires. A few other successfully fabricated top-to-bottom micro- or nanostructures are the photonic crystal fibre, which consists of microtubular enclaves inside a silica fibre²⁰ for optical guiding, thermally drawn composite fibres integrating semiconductors, metals and insulators featuring micro- and nanosizes for advanced functionality^{21–24}, and silica nanofibres obtained by tapering²⁵. Direct attempts to produce very long metallic²⁶ and semiconducting^{27,28} micro- and nanowires inside viscoelastic matrices using thermal size reduction similar to the Taylor wire process²⁹ could not produce ordered structures. Arbitrarily distributed filaments are obtained, making it impossible to identify/address each filament as single nanostructures.

Here we report the first successful fabrication of arrays of millions of ordered indefinitely long nanowires and nanotubes in a flexible polymer fibre. Using a new iterative size-reduction process, semiconducting nanowires and hollow-core piezoelectric polymer nanotubes, with diameters below 15 nm, are obtained as well-ordered high-density arrays. The nanostructure sizes are uniform for hundreds of metres along the fibre and radially homogeneous in the cross-section. Unique multimaterial core-shell nanowire arrays were obtained and nanowires and nanotubes can be extracted to obtain extremely long free-standing slivers.

To fabricate nanowires that bridge the macroscopic and the nanoscopic world, we designed a multistep iterative thermal size-reduction process inspired from composite-fibre drawing from polymer reels. We start with a macroscopic polymer rod with an annular hole which is tightly fitted with a thermoelastically compatible material that is to become nanowires on multiple axial elongation and radial reduction. Millimetric fibres obtained from the first thermal size-reduction step are cut and arranged in hexagonal lattices inside a protective jacket, vacuum consolidated and redrawn. This second step reduces the wire size to a few micrometres. The drawing step is repeated a third time with the fibres obtained from the previous step to obtain nanometre-sized wires (Fig. 1, see also Supplementary Fig. S1). Typically, macroscopic rods are reduced by 25- to 300-fold within each step. Using a reduction factor of 50–100 for three iterative steps, a 10 mm initial rod is reduced to hierarchically ordered 10 nm

¹UNAM-National Nanotechnology Research Center, Bilkent University, 06800 Ankara, Turkey, ²Institute of Materials Science and Nanotechnology, Bilkent University, 06800 Ankara, Turkey, ³Department of Physics, Bilkent University, 06800 Ankara, Turkey. *e-mail: bayindir@nano.org.tr.

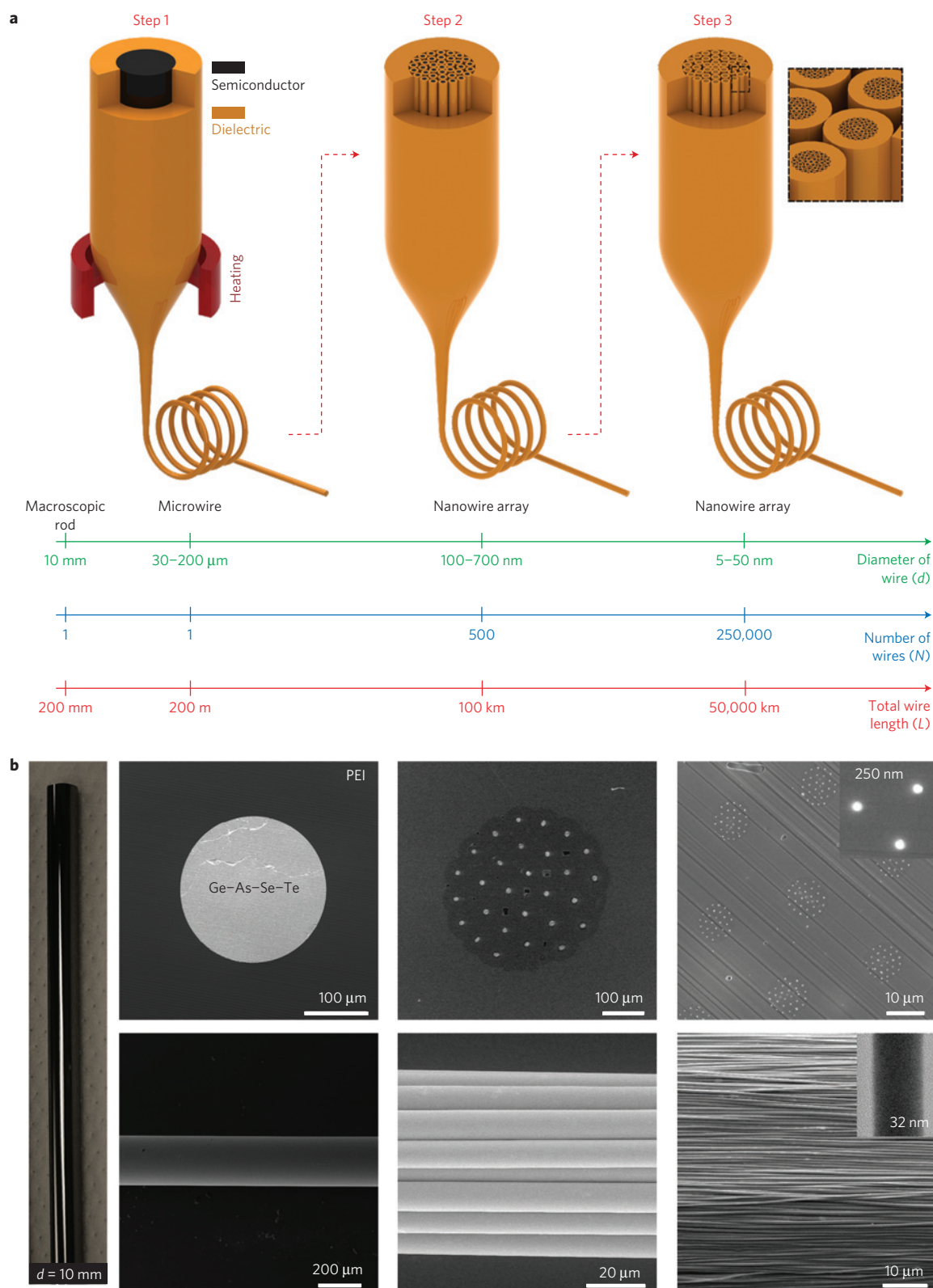


Figure 1 | A new nanofabrication technique, based on iterative size reduction, to produce ordered, indefinitely long nanowire and nanotube arrays. a, A macroscopic multimaterial rod is reduced to ordered arrays of nanowires by thermal size reduction in a protective polymer matrix in successive steps. Each step starts with structures obtained from a previous step, resulting in geometrical size reduction and increment in wire number and length. Using the technique, we produced millions of kilometre-long nanowires with sub-10 nm diameter and an aspect ratio of 10^{11} . **b,** As an example, a 10 mm amorphous semiconducting rod (Ge-As-Se-Te) is reduced to hundreds of metres of single 200 μm wire (reduction factor $\times 50$), ~ 30 wires of 5 μm diameter (reduction factor $\times 50$) and $\sim 1,000$ wires of 250 nm (reduction factor $\times 50$). Nanowire arrays are obtained as embedded in a dielectric polyetherimide encapsulation. Wire-array cross-sections with well-ordered wires and extracted, globally oriented slivers are shown after each step. Inset: Transmission electron microscopy image of a single 32-nm-thick nanowire obtained by further scaling down the third-step nanowires.

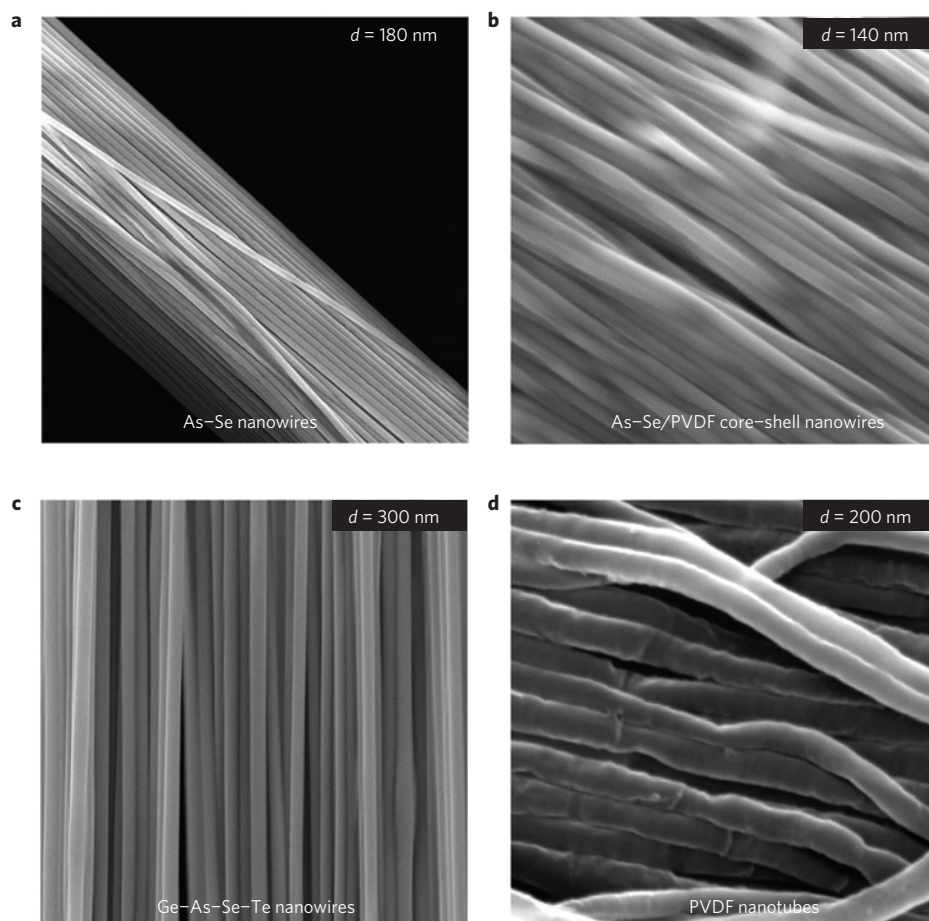


Figure 2 | Globally ordered, multimaterial nanowire, nanotube and cylindrical core-shell arrays. Nanowire and nanotube arrays are extracted from polymer matrix by chemical etching, retaining their global alignment. **a**, As_2Se_3 semiconducting nanowires. **b**, As_2Se_3 nanowire core with PVDF encapsulation forming a glass-polymer cylindrical core-shell structure. **c**, High-refractive-index low-bandgap semiconducting $\text{Ge}_{15}\text{As}_{25}\text{Se}_{15}\text{Te}_{45}$ nanowire slivers. **d**, Hollow-core piezoelectric polymer (PVDF) nanotube slivers with ~ 20 nm wall thickness. Hollow cores of the tubes are evident from the creases in the slivers after extraction by chemical etching.

wires. The total number of nanowires and their ultimate size distribution is determined by the number of packed fibres after each step, the total number of iterative steps and the reduction factor as shown in Fig. 1.

We demonstrate nanowire production using chalcogenide semi-conducting glasses in Fig. 1. Stable glass-making non-oxide chalcogen compounds have low softening points, making them amenable for thermal co-drawing inside polymer matrices owing to their excellent match of thermomechanical properties (Supplementary Fig. S2). We synthesized a 10-mm-diameter glass-making chalcogenide $\text{Ge}_{15}\text{As}_{25}\text{Se}_{15}\text{Te}_{45}$ (GAST, $T_g \sim 190^\circ\text{C}$) from high-purity elemental constituents using the conventional melt-quenching technique³⁰ (see the Methods section). Around the glass rod, a high-temperature engineering polymer (polyetherimide, $T_g \sim 220^\circ\text{C}$) is rolled and thermally consolidated in vacuum. The composite structure is then thermally drawn to reduce radial size (Supplementary Fig. S3). Typically, the process is carried out at 50 cm min^{-1} drawing speed while feeding the macroscopic rod at 1.5 mm min^{-1} into a furnace at 275°C . The ‘fibre wires’ obtained from the first drawing are then packed and vacuum consolidated to prepare a second-step macroscopic rod. Vacuum consolidation ensures structural homogeneity and purity from defects that may adversely affect the subsequent thermal drawing process (see the Methods section). Beginning with a 10-mm-diameter GAST rod, the diameter is successively reduced to $200\text{ }\mu\text{m}$, $5\text{ }\mu\text{m}$ and 250 nm within each drawing step. Further reduction in the third

step results in sub- 50 nm nanowire arrays. In Fig. 1, we show a perfectly ordered array of GAST micro- and nanowires in the polyetherimide polymer cross-section. It is remarkable that the macroscopic structure is undistorted after the drawing steps. We have also demonstrated that the micro- and nanowires can be extracted from the polymer matrix retaining their global alignment simply by exposing the polymer section to organic etching agents (dichloromethane (DCM) or dimethylacetamide; see the Methods section). Resulting free-standing micro- and nanowire slivers are shown in Fig. 1.

We fabricated a variety of multimaterial nanowire and nanotube structures by using the same technique (Fig. 2; Supplementary Figs S4–S7). Regular nanowire arrays are fabricated from the highly nonlinear glass As_2Se_3 (Fig. 2a), ordered core-shell arrays of As_2Se_3 –PVDF (polyvinylidene fluoride; Fig. 2b), high-refractive-index GAST glass (Fig. 2c), and photoconductive and phase-change Se glass (Supplementary Figs S4–S6). Separately, nanotubes of piezoelectric PVDF (melting temperature $T_m = 171^\circ\text{C}$) are obtained using the same process by using PVDF sheets as the filling material and polysulphone (PSU, $T_g \sim 190^\circ\text{C}$) as the jacket layer (Fig. 2d). PVDF sheets take their tubular shape through melting at the drawing temperature ($T_{\text{draw}} = 240^\circ\text{C}$) followed by self-organization due to high surface energy between the fluoride polymer and the protective PSU jacket. Macroscopic PVDF tubes can be seen after the first and second step drawings, and more interestingly tubular shapes are conserved in the

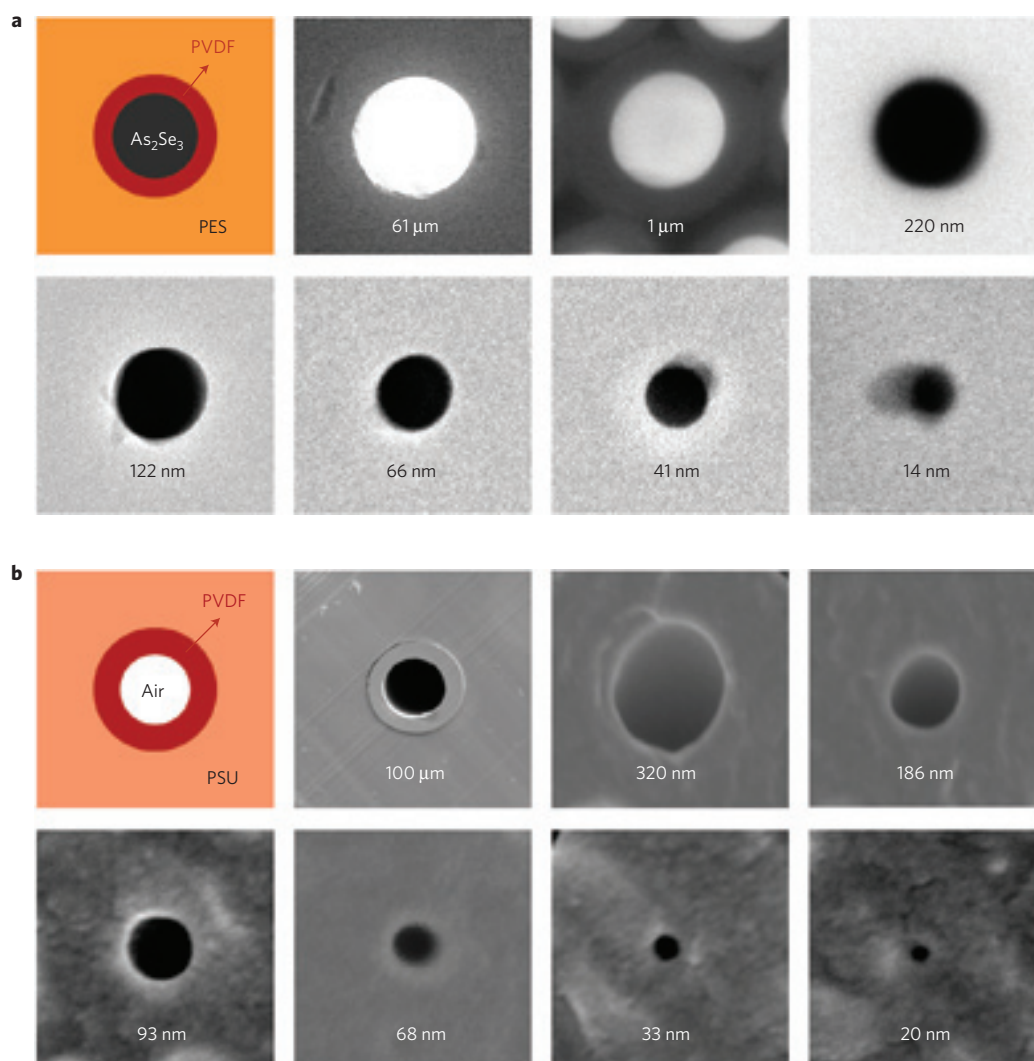


Figure 3 | Regular size reduction and ultimate achievable limit with multimaterial iterative size-reduction technique. We produced kilometre-long nanowires and nanotubes with any diameters within the 100 μm –10 nm range by preserving ordered geometry. **a**, As_2Se_3 –PVDF core-shell nanowires scaled regularly from 200 μm to 14 nm. **b**, Hollow-core PVDF nanotubes scaled regularly from 200 μm to 20 nm without losing structural integrity. Smearing around the nanowires is caused during sample preparation with ultramicrotomy.

nanoscale, resulting in very long arrays of nanofluidic channels in a flexible polymer fibre. Thermal melt-drawing of globally aligned PVDF nanotubes is an interesting alternative to existing templating and sol-gel methods for producing high-density nanotube channels³¹.

Nanowire fabrication with low-melting-temperature metals and metal alloys (Sn, SnPb, SnAg) was also explored using the same process. Fibre drawing is ideally designed for glassy materials; however, if it is possible to melt the crystalline substance without oxidizing inside a protective glassy matrix during thermal draw, crystalline rods will melt and recrystallize as elongated wires conserving the structural integrity^{21,22,30}. Recently, it was shown that other phases (such as metals and semimetals) can be drawn in an encapsulating glassy matrix³². Here, drawing occurs in a regime dominated by viscous forces, allowing for internal low-viscosity metallic domains to be arranged in cross-sections confined by viscous glassy boundary layers. However, this latter type of composite drawing becomes especially challenging with submicrometre filament sizes. The Taylor process, other attempts and the authors' own previous work suggests that crystalline materials can be co-drawn in glassy matrices when the drawing conditions are just right. To demonstrate that it is indeed feasible to

fabricate ordered arrays of micrometre- and sub-micrometre-scale metallic wires in polymer fibres, we selected tin and tin alloys owing to their low melting temperature ($T_m = 223^\circ\text{C}$). The drawing process is carried out with similar parameters to those of chalcogenide glasses. We started with a 1.6-mm-diameter soldering alloy $\text{Sn}_{96.5}\text{Ag}_{3.5}$ tightly fitted inside a hollow-core polyethersulphone rod, and scaled successively to 60 and 4 μm (Supplementary Fig. S7).

We demonstrate the utility of the top-down size-reduction technique by producing wires with sizes spanning the whole micro- and nanoscale (Fig. 3). Beginning with the same macroscopic rod, structures of six orders of magnitude in size can be obtained. We demonstrate this facility with core-shell As_2Se_3 –PVDF nanowires and hollow-core PVDF tubes. As_2Se_3 –PVDF core-shell nanowires were regularly scaled down to 61 μm , 1 μm , 220, 122, 66, 41 and 14 nm (Fig. 3a). Hollow-core PVDF nanotubes were scaled regularly, conserving their structural integrity, from 100 μm down to 20 nm (Fig. 3b). It is remarkable that we produce extended lengths of micro- and nanotubes that are highly sought materials in micro- and nanofluidics research. The wires and tubes are obtained as ordered arrays in the surrounding polymer matrix. In the micrographs, smearing around the nanowires is caused during

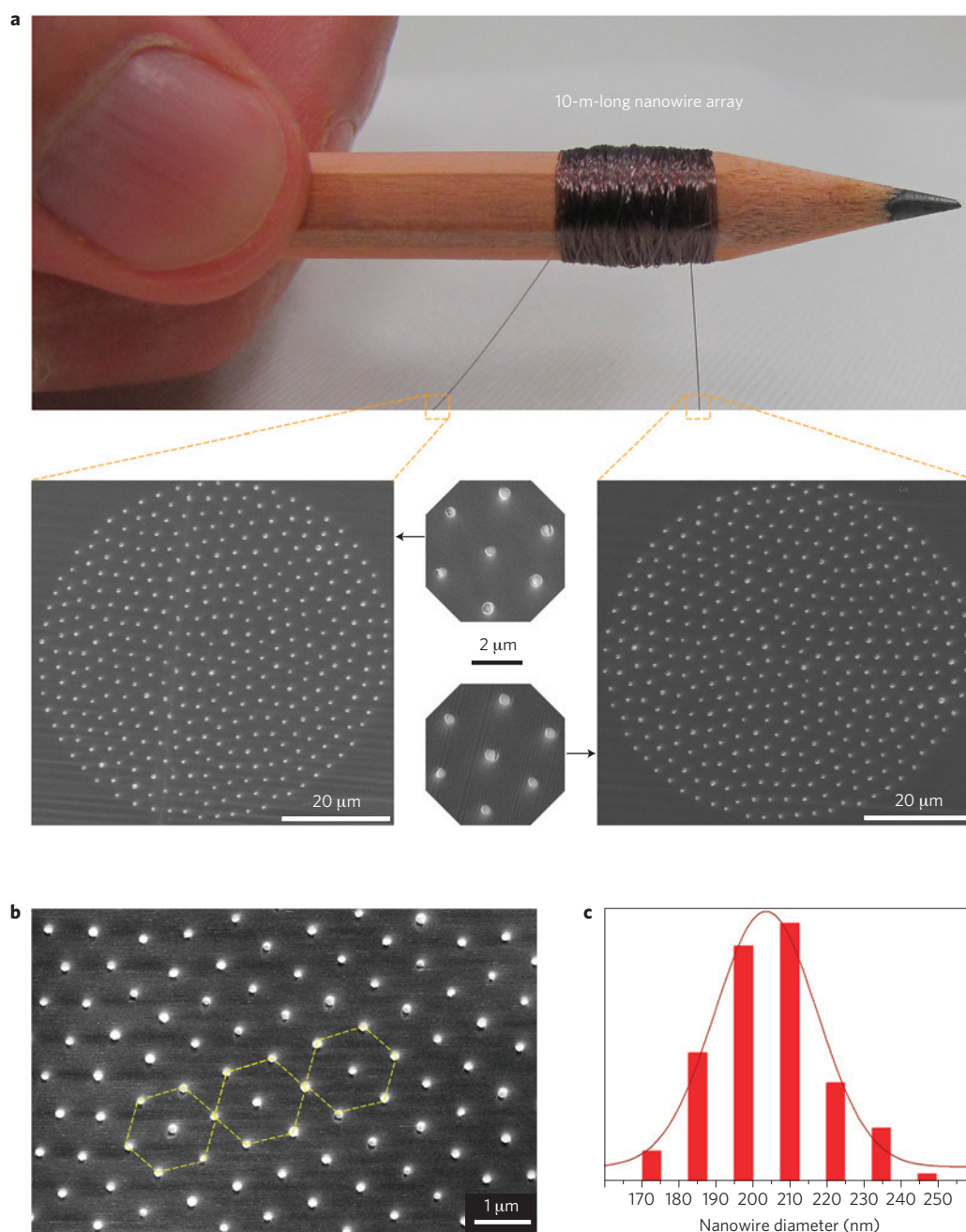


Figure 4 | Radial and axial uniformity of the nanowire arrays. **a**, A polymer-embedded nanowire array rolled around a pencil truly spans macroscopic and nanoscale worlds. Cross-sectional SEM micrographs from both sides of a 10-m-long polymer fibre that contains hundreds of As_2Se_3 -PVDF core-shell nanowires prove that nanowire arrays are axially uniform to less than 1% for macroscopic distances. **b**, High-precision hexagonal packing of core-shell nanowires in the polymer matrix. **c**, Radial size distribution of the nanowires, shown as a histogram, is uniform with a standard deviation of 6.5%.

sample preparation with microtomy, making it challenging to image embedded nanowires in the polymer matrix. For glassy filling materials, the ultimate size reduction limit depends on the drawing dynamics and viscoelastic properties of the matrix. Amorphous glasses and polymers that soften during drawing can be easily reduced to 10 nm radial size without axial breakdown. We infer that, with feed-in and draw speeds tuned and controlled carefully, the drawing process can yield molecular wires; the ultimate size reduction limit is still an enticing question.

We demonstrated radial and axial uniformity of the As_2Se_3 -PVDF core-shell nanowires by taking cross-sectional scanning electron microscopy (SEM) images (FEI, Nova NanoSEM 600) from both sides of a 10-m-long nanowire-embedded fibre, rolled

conveniently on a pencil (Fig. 4). It is extraordinary that the conformal surface coverage of a macroscopic object with nanowires can be achieved trivially. Roughly 400 of the As_2Se_3 -PVDF core-shell nanowires with 200 nm diameter are obtained with two-step iterative drawing. The two-dimensional array structure and integrity of the embedded nanowire arrays are retained for macroscopic lengths. The size distribution along the fibre length is uniform within 1% (Fig. 4a). High-precision hexagonal packing of core-shell nanowires in the matrix is shown in Fig. 4b. The size distribution of the nanowires from the second step is shown as a histogram and found to be uniform within 6.5% (Fig. 4c).

Finally, we investigated the size-dependent photoconductivity of selenium micro- and nanowire arrays fabricated using the described

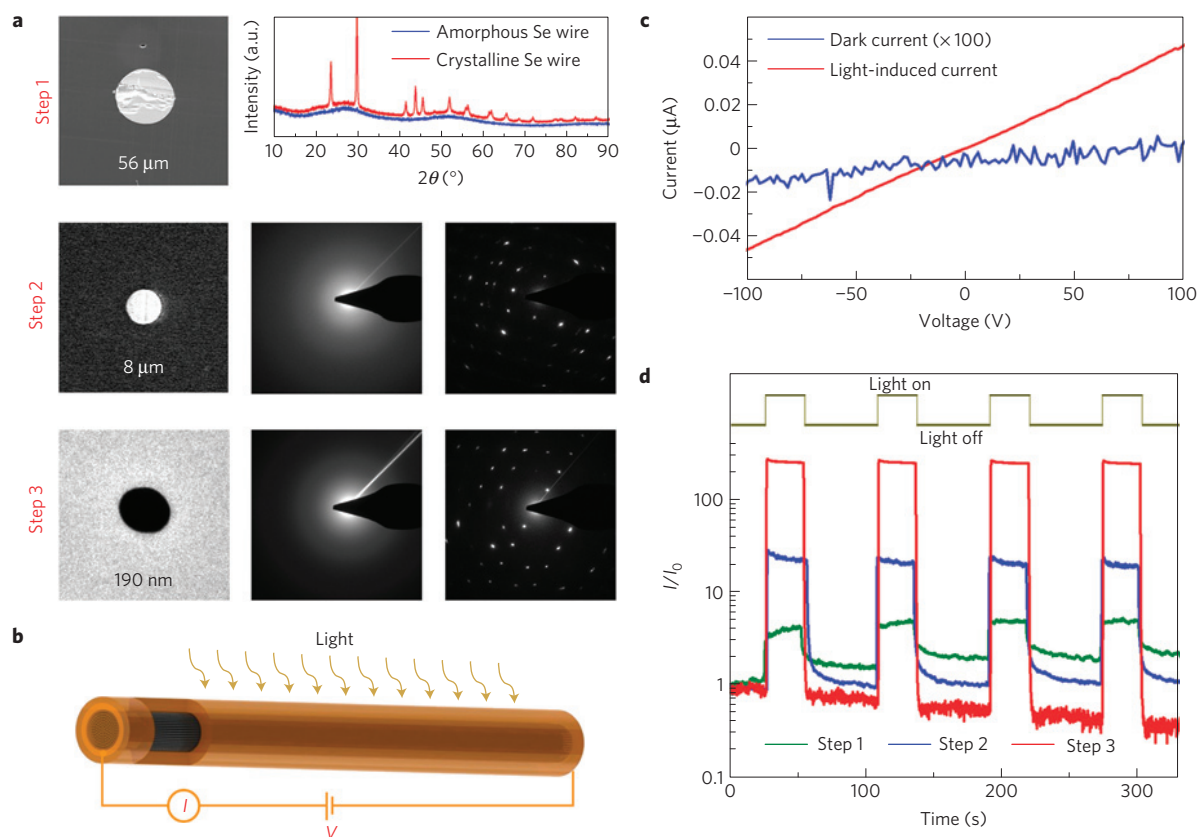


Figure 5 | Photoconductivity of selenium microwire and nanowire arrays. **a**, Amorphous selenium wires are crystallized through heat treatment or by exposing to the organic reagent pyridine. SEM images of regularly scaled-down individual selenium wires. The electrically conductive hexagonal crystallographic phase is obtained by X-ray and electron diffraction. **b**, A schematic representation of the photoconductance measurement geometry. **c**, The photoconductance from a selenium nanowire in the dark and on white-light illumination. **d**, A logarithmic increase in the ratio of photocurrent versus dark current (I/I_0) and a reduction in switching time (τ) with reducing nanowire diameter are observed. The performance increase is attributed to the high optical density of selenium, increased surface-area-to-volume ratio and enhanced scattering in the polymer matrix.

technique. Elemental selenium has phase-dependent electrical conductivity and photoconductivity properties, making it suitable for phase-change memory, optoelectronics and photodetection applications. In a three-step drawing process, we obtained a $56\ \mu\text{m}$ wire, an array of $100 \times 8\ \mu\text{m}$ wires and an array of $10,000 \times 190\ \text{nm}$ nanowires. The structure of the nanowires inside the polyethersulphone polymer matrix is conserved after the drawing steps (Fig. 5). The uniformities of the selenium nanowires from the second and third steps are found to be $\pm 3\%$ and $\pm 11\%$ respectively (Supplementary Fig. S5). Factors contributing to the overall size distribution are slight variations in the fibres that are used to make the macroscopic composite, and the total accumulated size variation from each previous step. Amorphous selenium wires are obtained after drawing, as revealed by X-ray diffraction and transmission electron microscope diffraction (FEI, Tecnai G2 F30). The wires are crystallized through heat treatment, by annealing above the crystallization ($T_x = 110^\circ\text{C}$) but below the melting temperature^{27,33} and by exposing to the organic reagent pyridine (Fig. 5a, see Methods section). After crystallization we observe that the conductivity increases to $10^{-4}\ \text{S cm}^{-1}$ consistent with bulk selenium conductivities³⁴. Electrical measurements were carried out with nanowire arrays by cleaving fibre ends with a razor blade and electrical contacting was achieved using silver paint on the facets of the fibre. Step 3 nanowires were chemically crystallized after extraction from the polymer matrix. Electrical gain due to photoconductivity is measured by shining white light (50 W) while applying voltage bias (Fig. 5b). Step-3 wires make Ohmic contact and feature photoconductivity after

crystallization (Fig. 5c). The photoconductance of the nanowires is observed to be a function of the nanowire number and size in the array. We observed that the I/I_0 gain (photocurrent versus dark current) was an order of magnitude larger for step-2 wires and two orders of magnitude larger for step-3 wires with respect to the single selenium microwire from the first drawing (Fig. 5d). The switching between on/off states is also faster with the smaller nanowire sizes. We attribute the logarithmic increase in photoconductance to the high absorption of selenium, increased surface-area-to-volume ratio and enhanced scattering of the incoming beam.

We have demonstrated a unique, high-throughput, lithography-free nanofabrication method to produce macroscopically large nanowire and nanotube arrays. The nanostructures are indefinitely long, ordered, uniform, high-density arrays embedded in a flexible polymer fibre matrix. Diverse multimaterial sets and disparate combinations are used to obtain crystalline semiconducting nanowire arrays, chalcogenide nanowire array waveguides, piezoelectric polymer nanotubes and cylindrical core-shell structures. It is an open question if molecular wires can be obtained using the method. The ultimate size reduction for the nanowires needs to be studied by optimizing drawing parameters for thermoviscous materials and materials that melt during drawing separately. On the other hand, the material set can also be widened to include semiconductors, such as Si and Ge, that soften/melt at higher temperatures using suitable glassy matrices.

We expect that a whole new family of radically novel nanowire applications will be enabled owing to the unique

geometry and material set used. We identify a number of enticing applications that are of current and rigorous scientific and technical interest. For example, these nanostructures can be exploited in nanowire electronics, as large-area conformal photodetectors, large-area flexible nanowire sensors¹⁵, scalable high-density nanowire-based phase-change memory^{35,36} and in high-speed reconfigurable field-effect transistors³³; in energy harvesting, large-area cylindrical heterostructure nanowires can be used as active cells^{37,38} but also for passive light enhancement in resonance-enhanced third-generation photovoltaics^{39,40}; in semiconductor nanophotonics, polymer-embedded chalcogenide nanowires can be used as high-refractive-index dielectric structures for size-dependent absorption⁴¹, structural colouring⁴² and biomimicry, and as dielectric metamaterials⁴³; in nonlinear photonics, ordered nanowires can be used as high-power zero-dispersion optical arrays for new frequency generation⁴⁴ and in discrete optics⁴⁵; in mechanics, nanowire-embedded high-strength composites²⁹; in acoustics, flexible piezoelectric nanowires and nanotubes can be used as sensors and actuators⁴⁶, as pumps, in energy harvesting⁴⁷ and as nanochannels in nanofluidics.

Methods

Synthesis of chalcogenide-glass rods. As₂Se₃ and Se rods, 10 mm in diameter and 15 cm in length, are prepared by melting commercially bought glasses (Amorphous Materials) and pellets (Alfa Aesar) in a vacuum-sealed quartz tube followed by water quenching. Ge₁₅As₂₅Se₁₅Te₄₅ glass rods (the same diameter) are prepared from high-purity elements (Alfa Aesar) using the sealed-ampoule melt-quenching technique. The weighted materials are placed in a quartz tube under nitrogen atmosphere in a glove box (H₂O, O₂ below 0.1 ppm). The quartz tube is kept at 330 °C for an hour under vacuum to remove surface oxides and sealed (10^{−6} torr). The sealed ampoule is then heated to 950 °C, rocked for 18 h and then quenched in iced water.

Macroscopic-composite preparation, consolidation and thermal size reduction.

Macroscopic rods that will undergo axial elongation and radial size reduction are structures approximately 30 mm in diameter. These structures are prepared in a clean pressure flow room, by tightly rolling polyethersulphone or polyetherimide films around chalcogenide glass rods (10 mm diameter) until the diameter is 30 mm (see Supplementary Fig. S1). Then, the macroscopic structure is thermally consolidated under vacuum (10^{−3} torr), above the glass transition temperature of the polymer and glasses to fuse (260 °C for 30 min). Step-2 and higher-step structures are prepared with previously obtained polymer-wire fibres and a hollow-core polymer rod fabricated by rolling polymer sheets on a Teflon rod, consolidation and removing the Teflon rod. About 100 fibres with 0.5 mm diameters are cut as 10 cm fibres, tightly packed and placed inside the hollow-core polymer rod. The structure is consolidated once more to fuse packed fibres to the outer polymer jacket (250 °C for 15 min). The second consolidation step ensures structural integrity during drawing. The macroscopic structure is heated and drawn into hundreds of metres of micro- and nanowire in a vertical two-zone furnace, with a top-zone temperature of 275 °C and bottom zone 200 °C (see Supplementary Fig. S3). The nanowire diameter is controlled using a laser micrometer by monitoring the fibre diameter.

Nanowire and nanotube extraction. The polymer encapsulating the nanowires and nanotubes was extracted for electron microscopy imaging in the longitudinal dimension by DCM (Carlo Erba). Nanowire or nanotube arrays embedded in polymer were fixed on a glass substrate by means of aluminium foil, and then half soaked in bottles containing DCM. The extracted nanostructures were further rinsed gently with DCM to remove residual polymer. A final O₂ plasma treatment (Nanoplas DSB6000) for several hours was applied for chalcogenide and metal nanowires to remove organic residues.

Nanowire crystallization by thermal and chemical methods. Step-1 selenium nanowires were crystallized by annealing at 150 °C for 1 h. Step-2 nanowire crystallization occurred after keeping nanowire arrays at 220 °C for 15 min and cooling to room temperature in 1.5 h. For crystallization of step-3 selenium nanowire arrays, extracted nanowire arrays were soaked overnight in a 10% pyridine (Riedel-de Haën) solution in isopropanol.

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Author contributions

M.Y. and M.B. designed and carried out research, analysed data and wrote the paper. M.K., T.K., M.Y., and M.B. carried out fabrication of nanowires and nanotubes. E.O. and O.A. made photoconduction measurements, E.O.O. and H.D. took SEM and transmission electron microscope micrographs and E.K. and M.B. drew schematic representations.

Additional information

The authors declare no competing financial interests. Supplementary information accompanies this paper on www.nature.com/naturematerials. Reprints and permissions information is available online at <http://www.nature.com/reprints>. Correspondence and requests for materials should be addressed to M.B.

ERRATUM

Arrays of indefinitely long uniform nanowires and nanotubes

Mecit Yaman, Tural Khudiyev, Erol Ozgur, Mehmet Kanik, Ozan Aktas, Ekin O. Ozgur, Hakan Deniz, Enes Korkut and Mehmet Bayindir

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In the version of this Letter previously published, the key for Fig. 5c was incorrect. This error has now been corrected in the HTML and PDF versions.