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Substrate temperature influence on the properties of GaN thin films grown by hollow-cathode plasma-assisted atomic layer deposition

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Gallium nitride films were grown by hollow cathode plasma-assisted atomic layer deposition using triethylgallium and N_2/H_2 plasma. An optimized recipe for GaN film was developed, and the effect of substrate temperature was studied in both self-limiting growth window and thermal decomposition-limited growth region. With increased substrate temperature, film crystallinity improved, and the optical band edge decreased from 3.60 to 3.52 eV. The refractive index and reflectivity in Reststrahlen band increased with the substrate temperature. Compressive strain is observed for both samples, and the surface roughness is observed to increase with the substrate temperature. Despite these temperature dependent material properties, the chemical composition, E_1(TO), phonon position, and crystalline phases present in the GaN film were relatively independent from growth temperature. © 2015 American Vacuum Society.
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I. INTRODUCTION

GaN is among the most notable III–V compounds with its excellent properties, such as wide direct bandgap, high thermal stability, and high electron velocities, which make it an attractive candidate for short wavelength optoelectronics and high-power and high-frequency electronics.[1,2] Although high-quality GaN films grown by metal-organic chemical vapor deposition[3,4] and molecular beam epitaxy[5] at high growth temperatures ($>800\,°C$), which is necessary to increase the cracking efficiency of ammonia (NH_3) and mobility of precursor species at the substrate surface,[6] high growth temperatures limit the integration of GaN-based optoelectronic devices with Si CMOS technology as well as not being suitable for deposition of GaN on flexible polymeric substrates. Moreover, the integration of narrow band gap (In-rich) with wide band gap (Ga-rich) III-nitride layers is not possible due to the low vapor pressure and disassociation temperature of InN.[7] This restricts the development of highly efficient III-nitride multijunction solar cells and spectral tunable light sources. In order to overcome the limitations due to high substrate temperatures, most prominent alternative approaches for low-temperature growth of GaN include physical vapor deposition techniques (sputtering and pulsed laser deposition) as well as a chemical vapor deposition technique: plasma-assisted atomic layer deposition (PA-ALD).[8–10] When compared with PVD techniques, PA-ALD stands out in terms of large area uniformity, submonolayer precision thickness control, and ultimate conformality.[11]

There are only a few recent reports on the growth of GaN films using atomic layer deposition at low temperatures ($<300\,°C$) with improved crystalline quality and electrical properties.[6,12,13] In these reports, PA-ALD of GaN using trimethylgallium,[12] triethylgallium (TEG),[6,13] and N_2/H_2 plasma as the precursors were reported at relatively low substrate temperatures. However, yet there is no report on the effect of the substrate temperature on the structural and optical properties of GaN films grown by using hollow-cathode plasma-assisted atomic layer deposition (HCPA-ALD). An optimized ALD process window was developed in the self-limiting growth regime ($185–385\,°C$),[10] in which saturation for both precursors are satisfied along with constant growth rate per cycle. Details of the optimization results will be published elsewhere. Following the growth of GaN film within the ALD window, GaN films were deposited at higher substrate temperatures within the decomposition limited region in which CVD type of deposition takes place and growth rate increased with substrate temperature.[14]

In this work, we have reported the effect of the substrate temperatures (200 and 450 °C) on the structural and optical properties of HCPA-ALD grown GaN films on Si (100) and quartz substrates. A comparative analysis of structural, compositional, refractive index, optical phonon modes, and optical band edge is demonstrated.

II. EXPERIMENT

GaN samples were grown using a modified Fiji F200-LL ALD reactor (Ultratech/CambridgeNanotech, Inc.) in which quartz-based inductively coupled RF-plasma source was replaced with a stainless steel hollow-cathode plasma source (Meaglow Inc.). GaN films were grown with identical parameters except substrate temperatures of 200 and 450 °C. TEG was kept at room temperature and used as the gallium precursor. The TEG exposure doses were adjusted in order to obtain the self-limiting growth scheme which is an evidence of ideal ALD growth mechanism. Three-hundred watt

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remote rf-plasma assisted decomposition of nitrogen/hydrogen (N_2/H_2) mixture was utilized as a nitrogen precursor. In addition, the total numbers of deposition cycle were adjusted to achieve similar film thickness (1750 cycles for sample grown at 200 °C and 900 cycles for sample grown at 450 °C). Based on the optimization studies, each ALD cycle consisted of four steps: 0.5 s TEG, 10 s Ar purge, 40 s 50 + 50 sccm N_2/H_2 plasma, and 10 s Ar purge.

Structural properties of the GaN films grown on Si(100) substrates were studied by using a PanAnalytical X'Pert PRO MRD system using Cu Kα radiation in Θ-2Θ mode. Grazing-incidence x-ray diffraction (GIXRD) has been performed in the range of 25°–75°. Incidence angle (ω) was 0.3° for GIXRD measurements. GIXRD patterns were obtained by performing scans with a step size of 0.1°. Peak positions were obtained by fitting the GIXRD data using PANALYTICAL X'PERT DATA VIEWER software. X-ray photoelectron spectroscopy (XPS) (Thermo Scientific) measurements were carried out using Al Kα radiation. The pass energy, step size, and spot size were 30 eV, 0.1 eV, and 400 μm, respectively. Peak deconvolution was performed using the AVANTAGE software, without applying any restrictions to spectral location and full width at half maximum values. The surface morphology was investigated by using atomic force microscopy (AFM, Park systems Corp., XE-100). The film thicknesses and optical constants were measured by spectroscopic ellipsometer (SE) (V-VASE, J.A. Woollam Co., Inc.) in the wavelength range of 300–1000 nm at various incidence angles. In order to evaluate the film thickness and optical band gap, the UV-visible room temperature optical transmission measurements were carried out in the 215–1000 nm wavelength range on GaN films deposited on quartz substrates with a resolution of 1 nm. Fourier transform infrared reflectance (FTIR) spectra were taken at room temperature and at normal incidence of IR radiation in the range of 450–700 cm^{-1} (22.22–14.28 μm) with a spectral resolution of 2 cm^{-1} by using a FTIR spectrometer (Bruker Vertex 70) with a mirror optics microscope (hyperion microscope) and a liquid nitrogen cooled MCT detector. All spectra were normalized to the reflection of a gold mirror.

III. RESULTS AND DISCUSSION

A. Structural, compositional, and surface analyses

Grazing incidence x-ray diffraction was used to analyze the structure of HCPA-ALD grown GaN films. Increasing the substrate temperature affected neither peak positions nor the diffraction planes. Figure 1 shows the crystalline phases of the ~60 nm thick GaN films grown at different temperatures. The patterns include three main Bragg peaks corresponding to (002), (103) reflections, and a peak that encloses the (200), (112), and (201) reflections of the hexagonal GaN phase. The sharp diffraction peaks showed no other phase or impurity related diffraction peak, which validates the relatively high purity and crystalline structure of both GaN films. Similar XRD patterns were obtained for GaN films deposited at different substrate temperatures. It can be seen that the GIXRD peak intensities increased with substrate temperature,
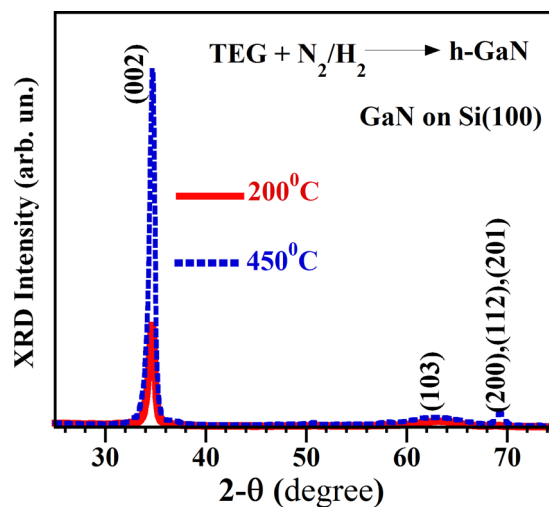
FIG. 1. (Color online) GIXRD patterns of GaN films as a function of 2Θ deposited at 200 and 450 °C on Si (100) with the growth parameters obtained in the ALD window.

indicating a slight improvement in crystalline quality of GaN, which might be due to faster reaction and mass transport kinetics.[14] The lattice parameters (*a* and *c*) for GaN films were roughly calculated from the peak positions using the famous Bragg's law. The calculated lattice constants in *c*-axis were 5.176 ± 0.00003 and 5.167 ± 0.00003 Å, and the calculated lattice in *a*-axis were 3.184 ± 0.00003 and 3.185 ± 0.00003 Å for the GaN samples grown at 200 and 450 °C, respectively. These values are in good agreement with unstrained GaN ($a_0 = 3.189$ Å, $c_0 = 5.185$ Å).[15] It should be noted that the lattice constant *a* of GaN films did not change with the substrate temperature while the lattice constant *c* decreased. The disharmony in lattice constants with the increasing substrate temperature was correlated with the compressive strain in the films. The elastic strains parallel ($\varepsilon_{//}$) to the *c*-axis and vertical ($\varepsilon_{\perp}$) to the *c*-axis were calculated by using the following expressions: $\varepsilon_{//} = (a_{epi} - a_0)/a_0$ and $\varepsilon_{\perp} = (c_{epi} - c_0)/c_0$.[15] c_{epi}, a_{epi}, c_0, and a_0 are the lattice constants for epilayer and bulk material, respectively. The calculated parallel strain were -0.15 ± 0.006 and -0.12 ± 0.006, and the vertical strain were -0.17 ± 0.006 and -0.34 ± 0.006 for the films grown at 200 and 450 °C, respectively. The calculated values of in-plane ($\varepsilon_{//}$) strain show the existence of very small compressive strain for both GaN films. On the other hand, the calculated negative values of out plane ($\varepsilon_{\perp}$) strain show the existence of compressive strain along the *c*-axis for both GaN films with increasing substrate temperature. The calculated values of $\varepsilon_{\perp}$ and $\varepsilon_{//}$ indicate the existence of compressive strain in all samples.

Furthermore, the average crystallite sizes of the GaN films were calculated from the (002) reflections using Scherer equation and found to be 12.98 ± 0.0003 nm (200 °C) and 14.26 ± 0.0003 nm (450 °C). It can be concluded that the crystallite size increased with substrate temperature, which confirms the improved GaN crystalline quality.

We explored the influence of substrate temperatures on the compositional properties of GaN films by using the aforementioned ALD growth conditions. Figure 2 shows the
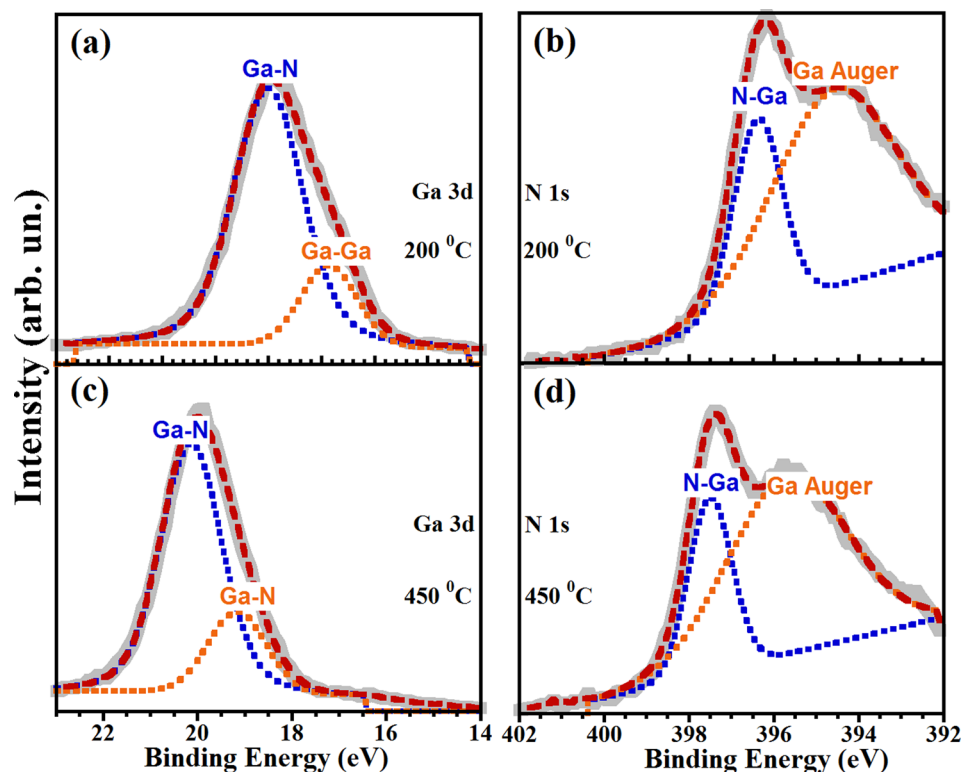
FIG. 2. (Color online) High resolution XPS spectra of (a) Ga 3*d*, and (b) N 1*s* for GaN film grown at 200 °C and (c) Ga 3*d*, and (d) N 1*s* for GaN film grown at 450 °C.

evolution of Ga 3*d* [Figs. 2(a) and 2(c)] and N 1*s* [Figs. 2(b) and 2(d)] high resolution XPS (HR-XPS) spectra for different substrate temperatures {200 °C [Figs. 2(a) and 2(b)] and 450 °C [Figs. 2(c) and 2(d)]}. The HR-XPS spectrum of Ga 3*d* [Fig. 2(a)] for GaN film grown at 200 °C can be decomposed into two components centered at 19.0 and 17.85 eV and assigned to Ga–N and Ga–Ga bonding states, respectively.[12,16] N 1*s* spectrum [Fig. 2(b)] for GaN film grown at 200 °C can also be similarly decomposed into two components at 396.37 and 394.57 eV, which is assigned to the N–Ga bond and Auger Ga peaks.[12] Although the spectral locations and intensities of Ga 3*d* and N 1*s* peaks were found to be slightly different for GaN samples grown at 450 °C, the atomic composition of GaN film was hardly changed. The HR-XPS spectrum of Ga 3*d* [Fig. 2(c)] for GaN film grown at 450 °C was fitted by two subpeaks located at 20.10 and 19.15 eV, corresponding to Ga–N bonds.[12] The N 1*s* spectrum [Fig. 2(d)] obtained for GaN grown at 450 °C was fitted similarly using two subpeaks located at 397.49 and 395.67 eV, corresponding to N–Ga and Auger Ga bonds, respectively.[12,16] As the deposition temperature increases, the Ga 3*d* and N 1*s* binding energies (BEs) shifted from lower BE to the higher BE. The shift in the Ga 3*d* and N 1*s* core levels was observed to be 1 eV. This shift might be attributed to a change in band bending, which is a result of oxygen reduction in the GaN structure. This explanation agrees well with the atomic percentage of oxygen obtained from the XPS analysis. The atomic percentage of oxygen is 3.24 and 1.71 at. % for GaN films grown at 200 and 450 °C, respectively. Nevertheless, no carbon signals were observed

by the XPS measurement, indicating that ligands consisting of ethyl groups in TEG were almost completely removed during the exposure of the N_2/H_2 plasma.

To further investigate the difference between the GaN films grown on ALD window and decomposition limited region, the surface morphology was characterized by AFM. The surface roughness and grain size of GaN films were obtained from AFM images [Figs. 3(a) and 3(b)]. Root mean square surface roughness values measured from a $1 \times 1 \mu m^2$ scan area of GaN samples grown at 200 and 450 °C were recorded as 0.36 and 0.98 nm, respectively. The surface of the film grown in self-limiting growth region is very smooth and mostly two-dimensional with several bumps compared to the film grown at 450 °C with hillocks larger in diameter. As a result, GaN film grown at higher temperatures exhibited substantially higher surface roughness. The mean grain sizes were obtained from the analysis of AFM images as 5.6 and 5.9 nm for GaN films grown at 200 and 450 °C, respectively. Although there is no significant difference in the AFM-measured grain size, the surface roughness becomes rougher for GaN films grown out of the ALD temperature window due to the effect of the growth temperature in terms of growth speed along different crystallographic directions.[13]

B. Optical properties

The refractive index (*n*), the extinction coefficient (*k*), and optical band edge values (E_g) were determined from spectroscopic ellipsometry measurements and are shown in Fig. 4. In order to model the spectroscopic ellipsometry spectra, we followed two steps: Cauchy dispersion function
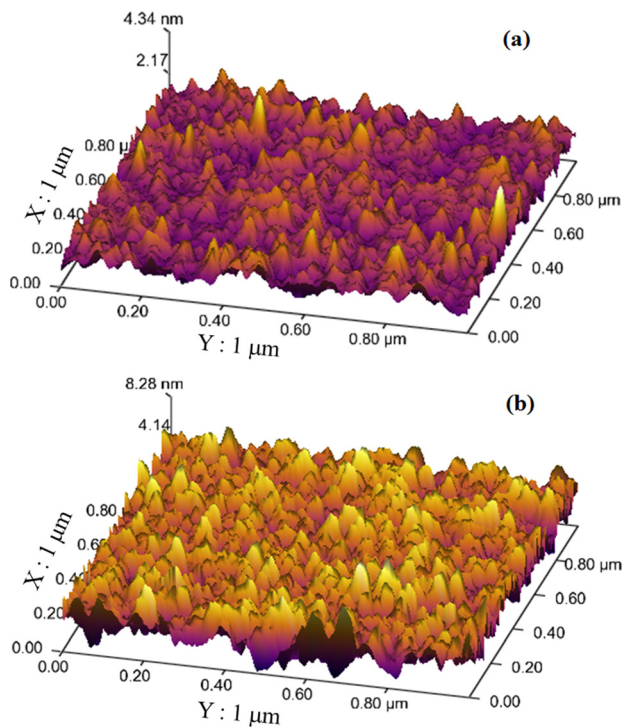
FIG. 3. (Color online) AFM images of GaN films grown at (a) 200 °C (b) 450 °C.

was used for transparent wavelength region as the first step. In the following step, by using the optical constants obtained from the first step, Tauc–Lorentz model was used for the absorbing wavelength region of GaN films. The measured ellipsometry data were fitted by assuming two layer structures with a bulklike GaN layer on the Si substrate and a top surface roughness layer for the film deposited at 450 °C. The rough layer is modeled as a mixture of bulk material and void layer. The fitting results provide thickness values for GaN film deposited at 450 °C as the following: 54.1 nm for bulk layer and 14.8 nm for rough GaN layer. On the other hand, the measured ellipsometry data for the GaN film deposited at 200 °C was fitted successfully by assuming merely single GaN bulk layer and the thickness of GaN layer is found to be 49.0 nm. There is a good correlation between the fractional thickness of the GaN film with the top rough layer deposited at 450 °C and the measured surface roughness, which is measured as 0.98 nm. This justifies the two-layer model used for the analysis of ellipsometry data. The formation of the rough layer with the increasing substrate temperature manifests the increase in surface roughness of the GaN films. The value of refractive index at 632 nm is 2.38 and 2.18 for the GaN films deposited at 450 and 200 °C, respectively. The evaluation of the refractive index of GaN is plotted in Fig. 4(a), showing a strong substrate temperature dependence of $n(\lambda)$. This temperature dependence of the refractive index can be due to the variation in crystalline orientation or near-random orientation of crystallites, surface morphology, and improved crystallinity.[17,18] Refractive indices obtained for both GaN films are consistent with the refractive index value of a typical polycrystalline GaN film.[17] The absorption coefficient, $\alpha(\lambda) = 4\pi k(\lambda)/\lambda$, was calculated from
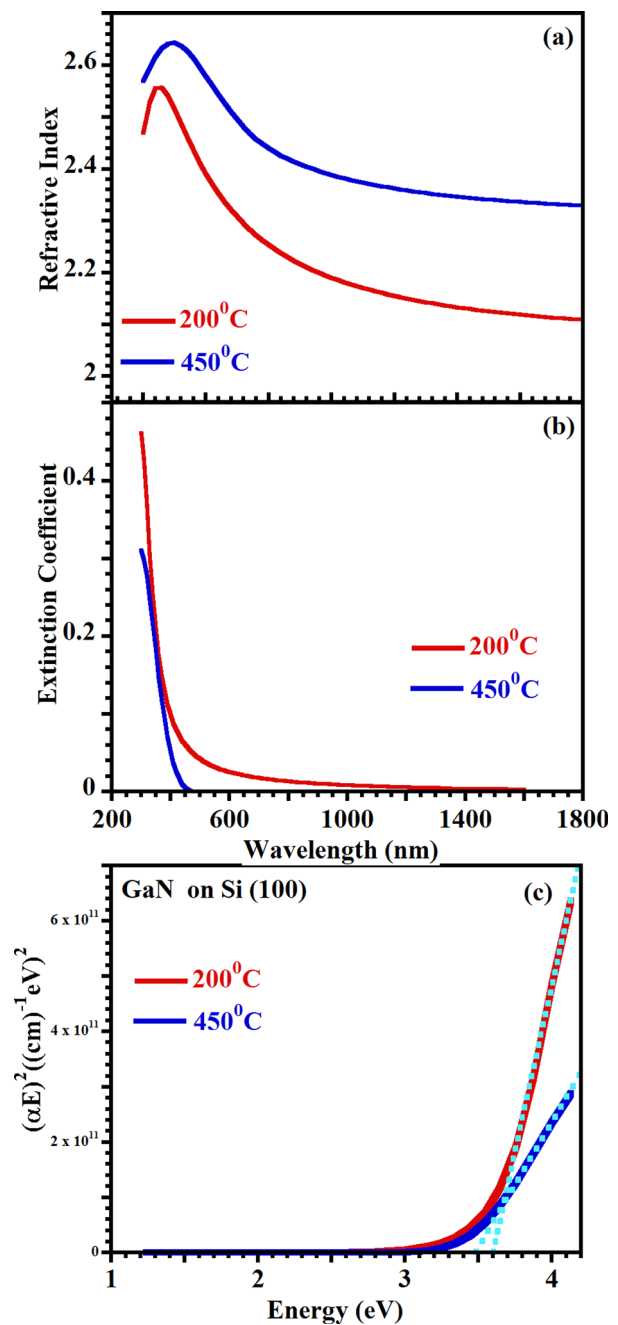

FIG. 4. (Color online) Spectral dispersion of (a) refractive index and (b) extinction coefficient of GaN films deposited at substrate temperatures of 200 and 450 °C. (c) Optical band edge obtained from the spectroscopic ellipsometry data.

the $k(\lambda)$ values determined from ellipsometry. The approximate optical band gap value was obtained via extrapolation of the linear part of the absorption spectrum to $(\alpha E)^2 = 0$. It is found that the band gap decreases with the substrate temperature from 3.60 to 3.52 eV.

The optical transmission measurements of GaN films grown on quartz substrates were carried out within the 215–1000 nm wavelength range. Figure 5 shows the optical transmission spectra of GaN layers together with the calculated spectra. The optical transmission spectra were fitted by applying a model dielectric function (MDF)[7] with additional
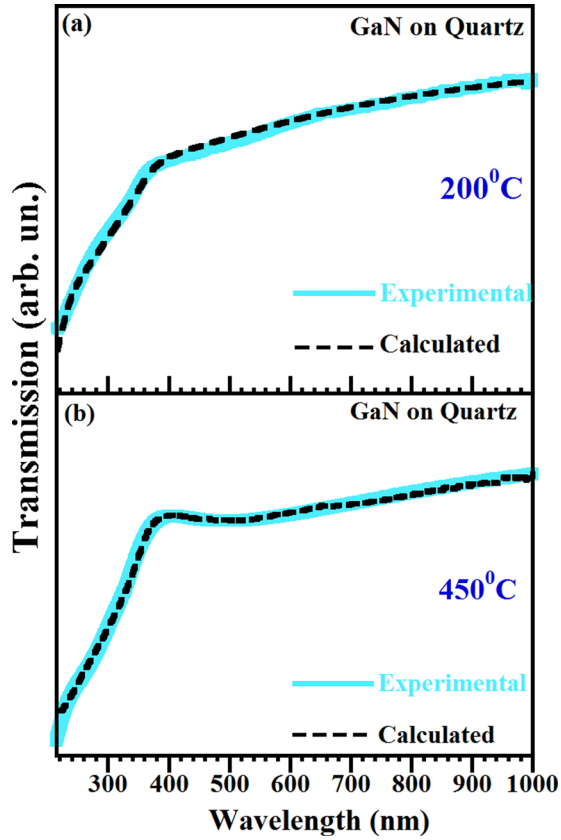
FIG. 5. (Color online) Experimental and best fit calculated transmission spectra of GaN films grown on quartz substrates at different substrate temperatures.

oscillators. This approach provided a good fit of experimental transmission spectra, where the MDF is defined as

$$\varepsilon'_{\text{GaN}}(\omega) = \varepsilon_{\text{GaN}} + \sum_{n=1}^{m} \frac{S_{ab}}{\left[E_{ab}^2 - (h\omega)^2\right] - ih\omega\Gamma_{ab}}. \quad (1)$$

Here, m is the number of additional absorption centers, ε_{GaN} is the dielectric function of GaN,[19] S_{ab} is the oscillator strength, E_{ab} is the energetic position of the oscillator, and Γ_{ab} is the damping of the Lorentzian. From Fig. 5, it is seen that there is a good agreement between the simulated and experimental data (solid line). The optical band gap values were found to be 3.55 and 3.50 eV for films grown at 200 and 450 °C, respectively. The two numbers are both higher than the bulk value of 3.4 eV and consistent with the optical band gap values obtained from SE analysis. The difference in the optical band gap may be attributed crystal disorder, dislocations, and grain boundary misorientations.[17,18]

Figure 6 shows the room temperature experimental (solid line) and calculated infrared reflectance spectrum for GaN films grown on Si(100) at 200 °C [Fig. 6(a)] and 450 °C [Fig. 6(b)]. Since no additional information was obtained in the high-frequency range above 700 cm^{-1}, we will focus on the Reststrahlen band between 500 and 700 cm^{-1}, which is centered about 570 cm^{-1}. The theoretical analysis of the FTIR reflectance from this GaN/Si sample can be applied to obtain further information on the optical phonon. Theoretical IR

spectra were calculated using the conventional transfer matrix method where we used $n = 3.42$ for the refractive index of Si.

The dielectric functional GaN layer is modeled using Eq. (2) assuming two contributions, two Lorentz oscillators for lattice contribution[20]

$$\varepsilon(\omega) = \varepsilon_{\infty}\left(\prod_{i=1}^{2} \frac{\omega^2 + i\gamma_{\text{LO},i}\omega - \omega_{\text{LO},i}^2}{\omega^2 + i\gamma_{\text{TO},i}\omega - \omega_{\text{TO},i}^2} + \sum_{j=1}^{5}\left(\frac{S_J\omega_{\text{TO},j}^2}{\omega_{\text{TO},j}^2 - \omega^2 - i\omega\Gamma_j}\right)\right). \quad (2)$$

Here, ε_{∞}, $\omega_{\text{LO},ij}$, $\gamma_{\text{LO},ij}$, $\omega_{\text{TO},ij}$, and $\gamma_{\text{TO},ij}$ are, in order, high frequency dielectric frequency, the frequency and the broadening value of the ith LO and TO phonon, S_{ij} denotes the strength, and Γ_{ij} the damping constant of the ith lattice mode. The free-carrier effects are neglected since the GaN layers are undoped. The experimental IR spectra for both GaN samples agree well with the calculated Reststrahlen band for a single crystalline wurtzite GaN film grown on Si(100). There is no significant discrepancy between the spectra of GaN samples. The reflectance intensity of Reststrahlen band increases with the deposition temperature, which is a result of improved crystallinity. Two main peaks (A,B) and one dip (C) in the IR spectra are the main features observed in the literature for epitaxial GaN films grown on
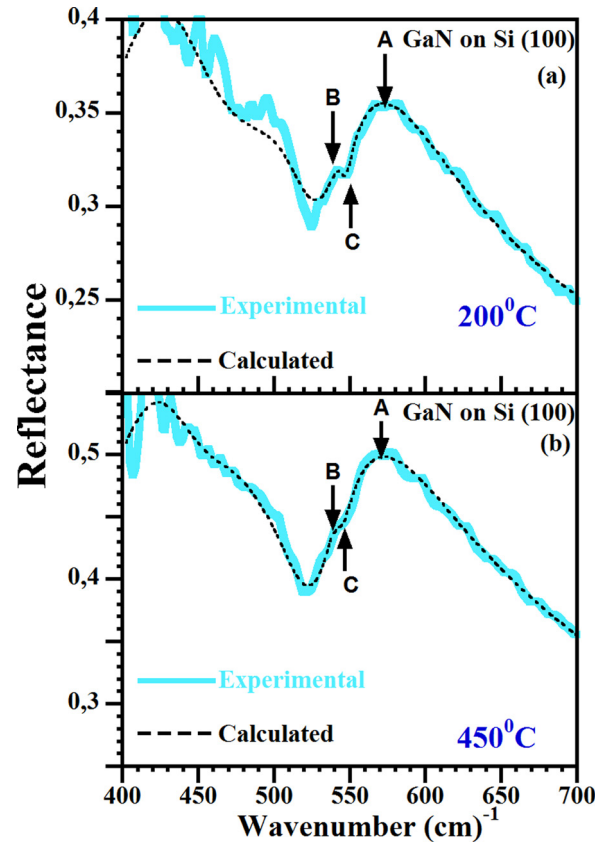

FIG. 6. (Color online) Experimental and calculated spectra IR reflectance spectra for GaN films deposited at substrate temperatures (a) 200 °C and (b) 450 °C.

Si.[21] The main peak A at $\sim$570 cm^{-1} is due to the reflection from the E_1(TO) phonon mode of wurtzite GaN. The E_1(TO) frequencies obtained from the theoretical calculations are 555 and 553 cm^{-1} for GaN films deposited at 200 and 450 °C, respectively. These assignments are close to the reported value for E_1(TO), which is at 557 cm^{-1} in the literature.[21] There is a peak located at point B, which is seen clearly for GaN films deposited at 200 °C, and the peak is very weak for samples deposited at 450 °C due to the rough surface morphology.[21] The calculated wavenumber positions for peak B are 539 and 537 cm^{-1} for substrate temperatures 200 and 450 °C, respectively. The calculated value for peak B is quite close to the determined position for A$_1$(TO) phonon mode at 533 cm^{-1}.[22] From the theoretical calculations, the wavenumber position obtained for dip C is 547 and 546 cm^{-1} for 200 and 450 °C, respectively. The calculated wavenumber positions are not dependent on the substrate temperature, and the shift toward lower wave numbers is consistent with the compressive strain present in the GaN films grown in both self-limiting ALD and decomposition limited CVD growth regime.

IV. SUMMARY AND CONCLUSIONS

We have studied hollow-cathode plasma-assisted atomic layer deposited crystalline GaN films in two main growth regimes: self-limited growth regime where growth rate per cycle stays constant with temperature (ALD region) and thermal decomposition-limited growth regime where growth rate increases with the substrate temperature (CVD region). As mentioned in GIXRD, AFM, spectroscopic ellipsometry, and optical transmission, crystallinity, microstructure, and surface roughnesses of these films grown with optimal growth parameters critically depend on the substrate temperatures. The increase in the substrate temperature leads to an increase in the growth rate, larger crystallite size, higher surface roughness, higher refractive index, and lower optical band gap value. It is found that there is a surface roughness layer including voids form with the increase in substrate temperature and correlates well with surface roughness of the films measured from the AFM images. The increase in optical band gap values calculated from spectroscopic ellipsometer and optical transmission data might be attributed to the film quality and preferred orientation of crystallites.[17] Furthermore, the behavior of E_1(TO) phonon modes were successfully described by a model using Lorentz oscillators for the dielectric function. The position of the E_1(TO) phonon mode is independent of the substrate temperature, and the shift of E_1(TO) phonon mode toward lower wavenumber was correlated and agrees well with the biaxial compressive strain present in both GaN films.

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