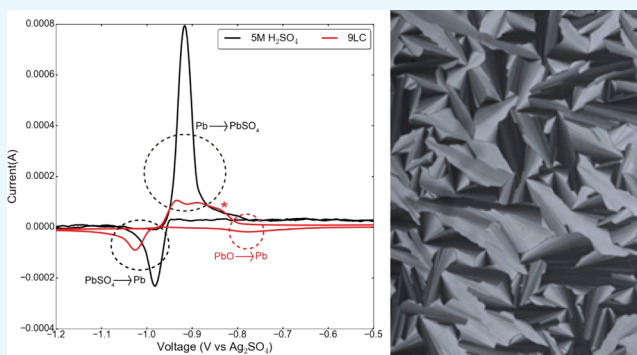


Lyotropic Liquid Crystalline Mesophase of Sulfuric Acid–Nonionic Surfactant Stabilizes Lead(II) Oxide in Sulfuric Acid Concentrations Relevant to Lead Acid Batteries

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ABSTRACT: Concentrated sulfuric acid (SA) and nonionic surfactant ($C_{12}H_{25}(OCH_2CH_2)_{10}OH$, $C_{12}E_{10}$) form lyotropic liquid crystalline (LLC) mesophases in a broad range of SA concentrations; the SA/ $C_{12}E_{10}$ mole ratio may vary from 2 to 11 in the LLC mesophases in the presence of a small amount of water. The mesophase is hexagonal at low SA concentration and cubic at higher concentrations. Three different compositions were prepared (one hexagonal and two cubic) with the SA/ $C_{12}E_{10}$ mole ratio of 2.5, 6, and 9, denoted as 2.5LC, 6LC, and 9LC, respectively. They all display electrochemical SA activity in Pt and Pb systems. Most interestingly, they show the electrochemical formation of stable PbO species in a deeply acidic medium as evidenced by the X-ray diffraction, cyclic voltammetry, and linear sweep voltammetry experiments. The preferable properties of PbO over $PbSO_4$ for lead acid batteries (LABs) make it uniquely positioned as a superior gel electrolyte for the LABs that would mitigate sulfation.



INTRODUCTION

Hygroscopic species, such as salts and acids, can be used as solvents in the assembly of surfactants into the lyotropic liquid crystalline (LLC) mesophases.^{1–4} Acids, such as H_3PO_4 and H_2SO_4 (sulfuric acid, SA), which have very low vapor pressure, form stable LLC mesophases in a very broad range of acid concentrations.^{3,4} Notice also that SA is a strong acid and forms mesophases upon addition of a small amount of water. The mixture of concentrated SA and surfactant is a liquid that flows when it is spread over a substrate. However, in a short period, it absorbs enough ambient water to form a rigid and stable LLC mesophase. The typical water uptake of a stable SA/ $C_{12}E_{10}$ LLC phase changes with the SA concentration and varies from 2.3 to 4.3 water/SA when the SA/ $C_{12}E_{10}$ ratio increases from 2 to 12.⁴ Note also that the mesophase is two-dimensional (2D) hexagonal up to around 3.5 SA/ $C_{12}E_{10}$ and becomes micelle cubic above this ratio. The cubic mesophases are always better conductors than the 2D hexagonal ones due to the three-dimensional (3D) connectivity of the ionic species in the mesophase.⁵ The SA/ $C_{12}E_{10}$ mesophases have excellent proton conductivities that increase with increasing SA concentration in the media. As the SA/ $C_{12}E_{10}$ ratio increases from 2 to 11, the proton conductivity increases from 2 to 19 mS/cm.⁴

Acid–surfactant and salt–surfactant mesophases have great potential to be used as gel electrolytes in various electrochemical devices such as batteries, solar cells, supercapacitors, electrochromic devices, and so on.^{6–11} The LLC mesophases have already displayed excellent conductivity that finds application in many electrochemical systems.^{2,3,6–11} In this

report, this phase and its application in lead acid batteries (LABs) will be investigated.

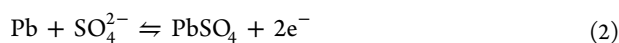
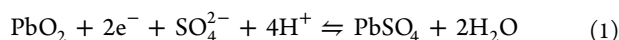
Recent industry projections predict that LABs will hold their dominant position in the energy storage market in the near future, owing to their current standing applications such as starter batteries in cars and uninterrupted power supplies at various institutions where low cost, reliability, and relative safety of the chemistry make them stand out.¹² Applications where safety is the most critical component, as is the case in military applications, LABs' proven safety record makes them indispensable. For example, LABs are used to buffer the output of the reactor in nuclear submarines. The U.S. Department of Defense has recently awarded a contract worth U.S.\$31 million to improve the batteries for this application.¹³ On the flip side, SA industry is thriving, owing to the established market in LABs, which use SA as an electrolyte. The total annual consumption is estimated to be U.S.\$85 billion.¹⁴ Therefore, new developments that improve LABs would have substantial industrial and commercial impact, even though the chemistry is mature and well-known.

In the charged state, the positive electrode of a LAB consists of Pb(IV) species and the negative electrode consists of Pb(0) species. At the end of discharge, both electrodes attain Pb(II) species. The main electrochemical reactions are as follows (forward reaction indicates discharge)¹⁵

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The formation of large PbSO_4 crystals as the discharge products, which cannot be reduced back to lead during subsequent charge cycles, causes battery death. This process, known as sulfation, is known as one of the leading causes of LAB failure. To battle sulfation, LABs are designed not to stay in the discharged state for extended periods. Although discharge products are always PbSO_4 , short stays in the discharge state only allow small and reducible/oxidizable PbSO_4 crystals to form, thus preventing compromise of the operation of the battery. Extended stays in the discharge state, however, cause formation of large nonconducting PbSO_4 particles, resulting in the complete breakdown of the battery.¹⁶ Attempted solutions to combat sulfation in the literature involve the addition of conductive additives to the electrode material or involve composite approaches, where current demands that would otherwise cause deep discharge are drawn from an ultracapacitor instead of the LAB.^{17,18}

At the end of the discharge cycle, if the Pb(II) species formed were oxides and not sulfates, it would be advantageous both in terms of the kinetics of electrode reactions and conductivity. The Pb(II) oxide species is known to be more advantageous both in terms of conductivity and kinetics than sulfate.¹⁵ Therefore, during the industrial production of LABs, the active material production starts from PbO and is mixed with enough water to produce the tribasic PbSO_4 ($\text{PbSO}_4 \cdot 3\text{PbO}$), which is more stable in acid solutions compared with PbO . During the actual battery operation, the Pb(II) species generated on both positive and negative electrodes are the sulfate because that is the stable form according to the well-known Pourbaix diagram at any pH below 8, as shown in Figure 1.^{15,19} In 6 M H_2SO_4 , which is typically used in LABs, the oxide species is not expected to be stable.

LABs using gel electrolytes are already commonplace.^{20,21} The gels employed are made up of colloidal silica that is cross-linked using the acid-based dehydration.^{20,21} The major advantage of using a gel-based electrolyte is the mitigation of

stratification. The gel keeps the electrolyte homogeneous along the height of the active material plates, which keeps the entire active material area operational throughout the battery life. A LLC that has electrochemically active SA would also potentially be a substituent for the colloidal silica gel, thus eliminating the stratification in a similar fashion.

In the LLCs, we will show that the oxide species are more stable than in the aqueous SA solution; it is verified through cyclic voltammetry (CV), X-ray diffraction (XRD), and visibly by the color of the samples as they come out of the electrochemical oxidation. This stability has consequences regarding mitigation of sulfation in the LABs.

RESULTS AND DISCUSSION

Three different compositions were prepared by mixing and homogenizing quantitative amounts of $\text{C}_{12}\text{E}_{10}$, SA, and water (1:2.5:7.5, 1:6:18, and 1:9:27 mole ratios, respectively) to obtain the liquid crystalline gel phases. The 2.5 SA/ $\text{C}_{12}\text{E}_{10}$ sample (denoted 2.5LC) displays a focal conic fan texture, characteristic of the 2D hexagonal columnar phase (Figure 2a). The other two samples of 6 and 9 SA/ $\text{C}_{12}\text{E}_{10}$ mole ratios (denoted as 6LC and 9LC, respectively) are dark between the crossed polarizers. All of the three samples diffract at small angles, with a slight shift to the lower angles with increasing SA amount in the samples; see Figure 2b. Even though there is a single line in the diffraction patterns, the mesophase is micellar cubic in 6LC and 9LC.⁴ Notice also that the surfactant molecules are not stable in concentrated SA solutions and slowly undergo dehydration to form carbon nanoparticles. A mixture of concentrated SA and surfactant is a transparent colorless liquid, which slowly turns into dark brown over time. The carbonization process is halted in the presence of a small and stoichiometric amount of water in the LLC media. For instance, if a solution of concentrated SA and $\text{C}_{12}\text{E}_{10}$ (undergoes slow carbonization) is exposed to ambient laboratory condition, the mixture slowly absorbs enough water to transform to a stable gel phase, where the $\text{C}_{12}\text{E}_{10}$ molecules are stable.

Two separate studies have been performed to investigate the proton activity or the apparent pH of the LC systems. Both tests are picked from the electrochemistry literature on well-established experiments using SA solutions as electrolytes. The two experiments picked are the well-known polycrystalline Pt in SA CV experiment and the standard polarization resistance experiment on stainless steel. The CV of a 5 M aqueous SA solution, 6LC, and 9LC using Pt as the working electrode (WE) are shown in Figure 3a (2.5LC data not shown for clarity). CVs clearly show that both 6LC and 9LC have electrochemically active SA and display characteristic peaks of Pt in the SA systems.^{22,23} This is a clear indication that the SA inside the LLC mesophases is electrochemically active, and that the cyclic voltammetric features of hydrogen adsorption/desorption and the oxide formation/reduction are feasible inside the LLC mesophases. The peak onsets shift to more positive potentials on the hydrogen desorption/adsorption in case of LLC phases than in the 5 M SA aqueous solution. This shift, roughly 40 mV, indicates that the activities of H^+ are slightly higher in the LLC mesophases than in the 5 M aqueous SA solution. The corrosion rate measurements of stainless steel in the LLC phases indicate a similar level of activity. In the literature,²⁴ it is well-established that the rate of corrosion increases with increasing concentration until 6 M and decreases with further increase from 6 M SA to a concentrated SA. As

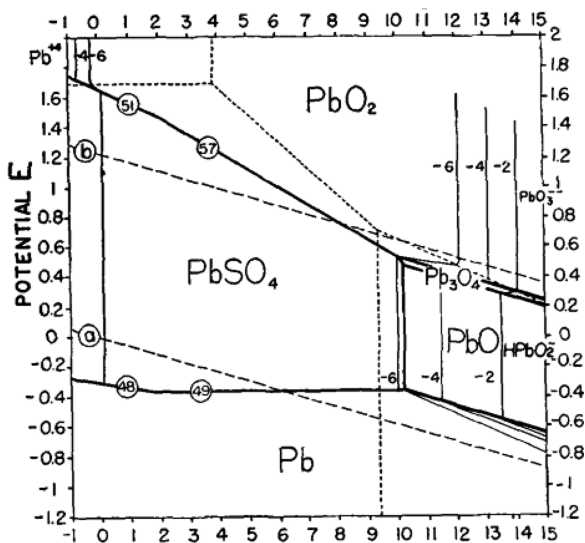


Figure 1. Well-known Pourbaix diagram for Pb species in SA solution. Reprinted from ref 19 with permission.

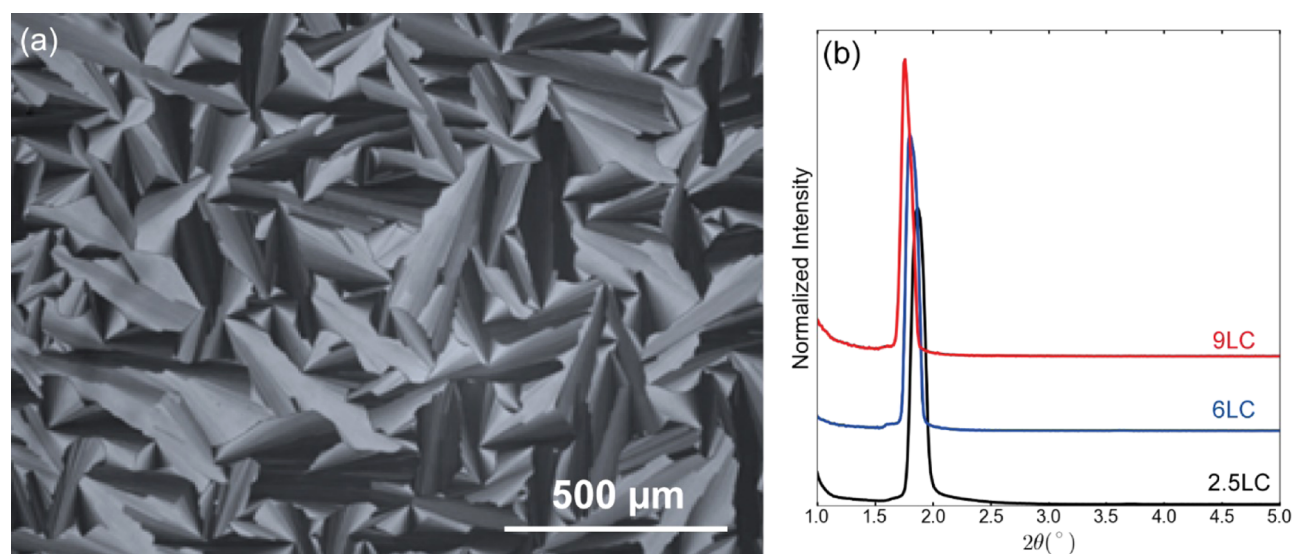


Figure 2. Polarized optical microscope (POM) image of 2.5LC (a) and small-angle XRD patterns of 9LC, 6LC, and 2.5LC, top to bottom (b). XRD patterns are offset for clarity.

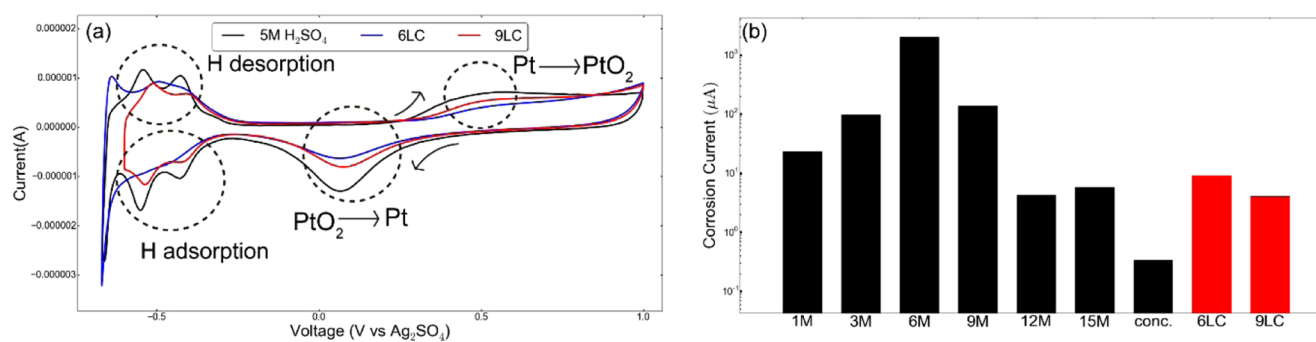


Figure 3. Pt electrode experiments (a) and the corrosion rate experiments (b) to confirm the electrochemical activity and to establish an analogy to aqueous solutions of SA.

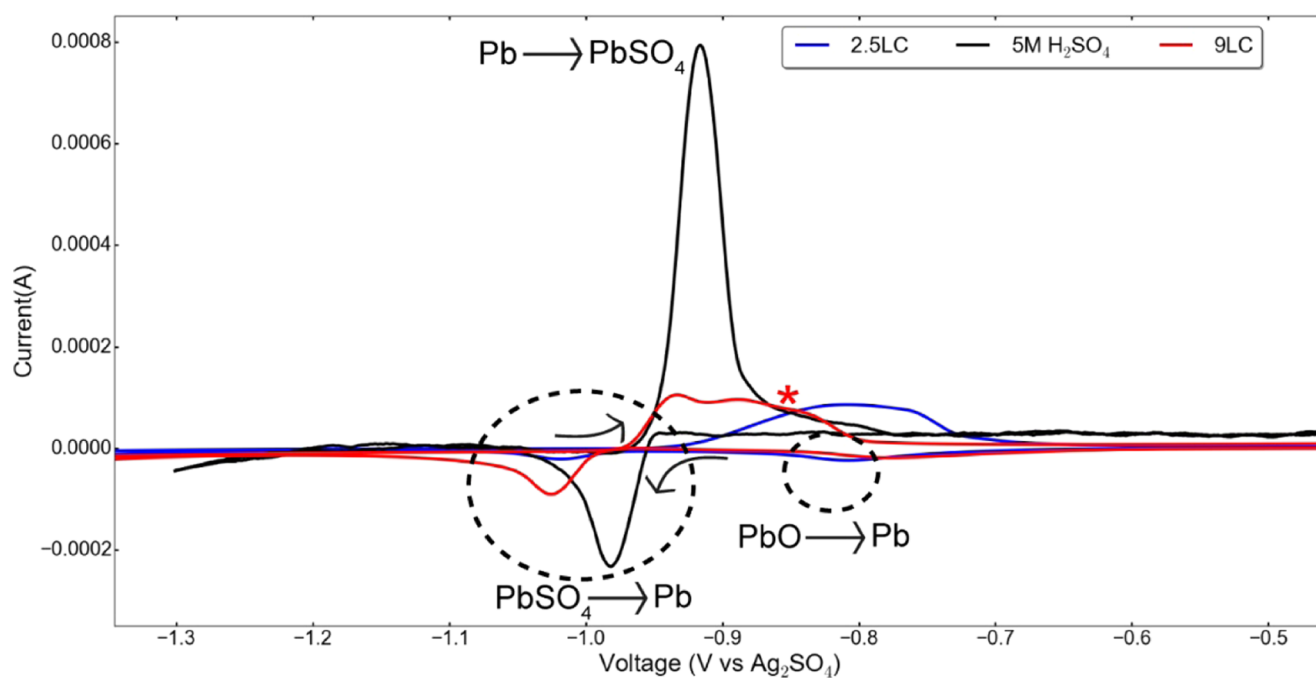


Figure 4. CV profiles for Pb electrodes in an aqueous solution (black) compared with those in two LLC phases (2.5LC in blue and 9LC in red).

shown in Figure 3b, the corrosion rates for 6LC and 9LC not only show a rate of corrosion that is commensurate with the above conclusion, but also that there is a reinforcing trend of decrease in the corrosion rate with an increase in acid amount. Based on the CV experiments of Pt and the corrosion rates of stainless steel, the SA inside the LLC mesophases is electrochemically active, and that the activities of the SA inside the phases correspond to the activities of 5–10 M SA in aqueous solutions.

The electrochemistry of Pb in SA is industrially important in LABs. Therefore, we focus on investigating the LLC mesophases using Pb as the WE. The SA/C₁₂E₁₀ LLC mesophases show not only high conductivity but also relevant electrochemical reactions. This highlights the applicability of LLC mesophases as alternative electrolytes in LABs. Moreover, any change in the properties of the relevant electrode reactions may have an impact on the preparation and preservation of the electrodes in the traditional LABs. As a result, we mainly focus on the oxidation of Pb to PbO/PbSO₄ and subsequent reduction of the latter to Pb metal, which is the negative electrode half-reaction in the LABs.

A typical Pb CV of a 5 M SA is shown in Figure 4. It displays an oxidation peak due to oxidation of Pb to PbSO₄ on the oxidation side and a much smaller reduction peak attributed to the reduction of PbSO₄ back to Pb, as also shown in the literature.^{25–27} The weak reduction peak is attributed to the formation of large nonconducting PbSO₄ crystallites that cannot be reduced.¹⁵ The corresponding CVs in 2.SLC and 9LC mesophases display distinctively different voltammograms. Notice that the single oxidation peak in the 5 M SA becomes two in the SA/C₁₂E₁₀ samples. We focus our attention on 9LC due to its higher SA activity (as confirmed by the polarization resistance and the Pt electrode experiments). Looking at the data more carefully, we notice that the oxidation peak not only has a shoulder but also a new reduction peak, which is distinct from the reduction of PbSO₄ present. This shows the unique effect of SA/C₁₂E₁₀ mesophases on the Pb redox products. There is clearly a new species that gets reduced at a distinctly more positive potential than sulfate. Furthermore, the ratio of PbSO₄ reduction peak current to the Pb oxidation peak current is also markedly larger in the LLC system at 1 mV/s compared with the ratio in the 5 M SA solution, suggesting a more reversible reaction, as shown in Table 1. The LLC system leads to smaller PbSO₄ products that are more readily reduced than their counterparts formed in the aqueous solution.

Table 1. PbSO₄ Oxidation and Reduction Peak Currents at 1 mV/s Sweep Rate

electrolytes	oxidation peak current (μA)	reduction peak current (μA)	ratio (%)
9LC	107.8	101.6	94.2
6LC	225.2	46.08	20.5
5 M H ₂ SO ₄	979.6	270.5	27.6

Shoulder peaks can only be resolved at low sweep rates, such as 1 mV/s, indicating the slow kinetics of the process. To identify the new species formed in the LLC system, we conducted chronoamperometry (CA) experiments at various potentials to grow films over the Pb electrode and characterized them using the XRD and scanning electron microscopy (SEM) (Figure 5). First inspection of the Pb electrodes after this process shows a distinct difference between the electrodes

obtained from 5 M SA and 9LC in terms of the color of the Pb surface. The Pb electrode is yellow-brown in the SA/C₁₂E₁₀ compared with gray-white in the 5 M SA solution (see Figure 7 for pictures). We further analyzed the electrodes by collecting their XRD patterns. Figure 6 displays two sets of XRD patterns of the Pb electrodes after conditioning and CA experiments. The XRD patterns of both electrodes, after conditioning using 5 M SA and 9LC, only displayed Pb(0) diffraction lines. However, the electrode prepared from the CA experiment distinctively displayed lines originating from the PbO and PbSO₄ in the 9LC system (I) compared with predominantly PbSO₄ in the 5 M SA solution (II); see Figure 6b. The PbO lines are relatively broad due to the formation of PbO nanoparticles, preferentially grown along the [200] axis. The average particle size of the PbO particles was determined to be 14 nm, obtained from the Scherrer equation. Both forms of PbO with similar particle sizes (α -PbO and β -PbO with a higher concentration of β -PbO) are observed in the pattern obtained from the 9LC electrode. The SEM images of these two electrodes, prepared using 9LC and 5 M SA, display quite different surface morphologies. Crystalline surface species over the surface of the electrode used in 9LC are more uniform and much smaller than those on the electrode used in 5 M SA, consistent with the electrochemical and XRD results; see Figures 5 and 6. One explanation for this observed result is that the PbO formation on the surface of the Pb electrode is stable when it is in contact with the LLC phase opposed to the high instability in the SA solution.

To test this hypothesis, the resulting PbO/PbSO₄ grown on the Pb electrode using 9LC was submerged in the 5 M SA solution. The loss of the initial yellow-brown color hints at the instability of the formed oxide in the solution phase. The linear sweep voltammogram (LSV) of PbO/PbSO₄ after submerging in the 5 M SA solution (for 24 h), showing a single reduction peak attributed to the lead sulfate reduction back to lead, is shown in Figure 7b. Similarly, the LSV of a pristine PbO/PbSO₄ shows a shoulder peak (Figure 7a), thus suggesting the presence of another species, which is not stable in the aqueous SA solution, in the coating. Further cycling the potential for this lead strip did not yield the same shoulder peak as shown in Figure 7c. These results clearly show that the oxide species, as expected, are not stable in the solution phase, whereas they do form in LLC mesophases and are stable.

The reduction peak that is attributed to the reduction of PbO was also observed in the aqueous systems only when the CV was performed to very oxidizing potentials that also lead to PbO₂. This behavior was also shown in the literature.^{26,27} Under these conditions, the two reduction peaks are present, and the peak at the more positive potential is attributed to the reduction of PbO. In the current work, the reduction peak attributed to PbO is visible even when potentials are never swept to as positive potentials.

CONCLUSIONS

SA forms LLC mesophases with a nonionic surfactant (such as C₁₂E₁₀) and displays the electrochemical activity similar to that of aqueous SA. The SA/C₁₂E₁₀ mesophases, however, produce stable PbO in addition to PbSO₄ during the oxidation or discharge of LABs, which can be utilized industrially. It is likely that the PbO always forms during oxidation of lead in the discharge of LABs. However, PbO converts into insoluble, nonconducting PbSO₄ in ordinary LABs due to its high instability in aqueous SA solution. The formation and high

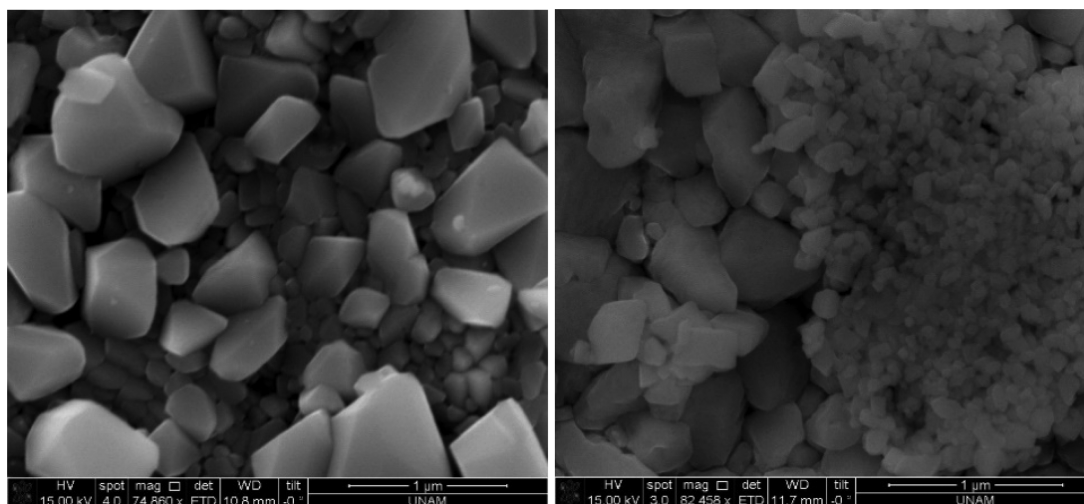


Figure 5. SEM images of a lead strip after 10 h of CA at -0.91 V (vs Ag_2SO_4) (left) in 5 M H_2SO_4 and in (right) 9LC .

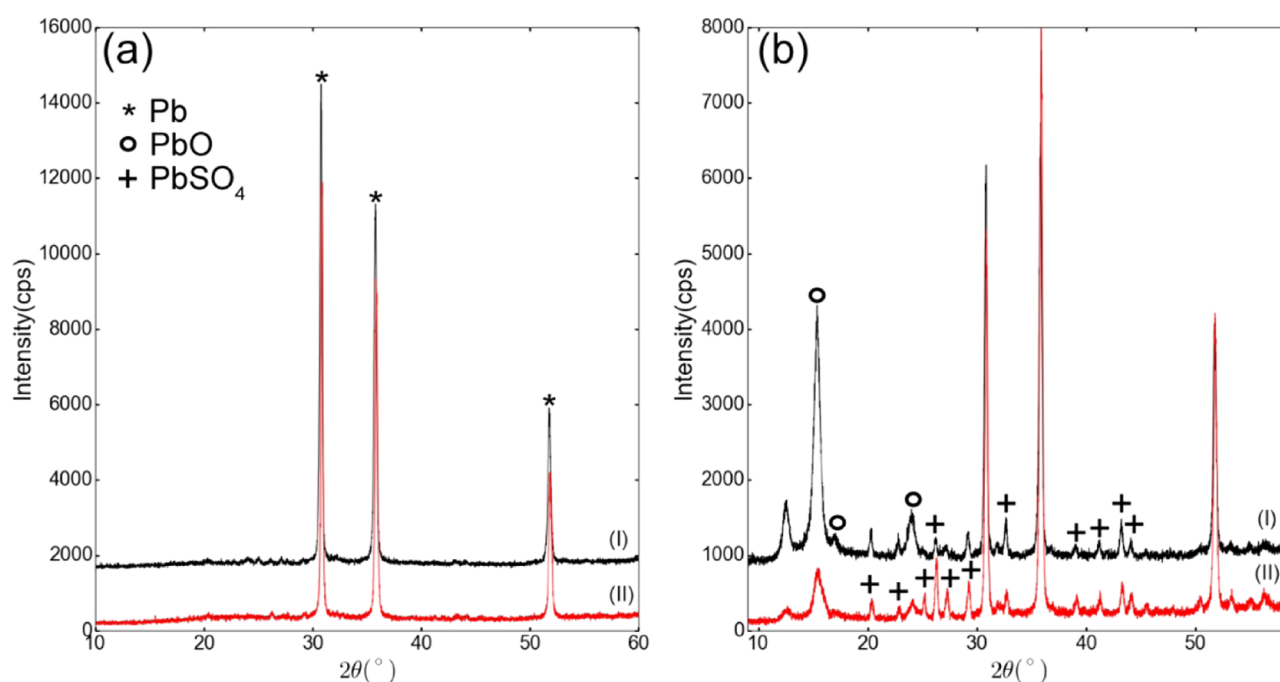


Figure 6. XRD patterns of as-conditioned Pb electrodes (a) and after oxidation (b). The black curves (I) show the results in the LC and the red curves (II) show the aqueous solution. XRD patterns are offset for clarity.

stability of PbO in the case of LLC as the electrolyte would be a vital improvement for the development of gel electrolyte LABs over traditional LABs.

One possible explanation for this effect is that the contact between the electrode and electrolyte is hindered by the surfactant molecules. This stabilizes the PbO formed over the electrode surface. Otherwise, a direct contact of the SA solution with the electrode surface is detrimental for the PbO particles and decomposes them into PbSO_4 . The mesophases are electrochemically active but chemically protective for the formation and stability of PbO .

EXPERIMENTAL SECTION

Preparation of LLC Gels. In a general procedure, the LLC gels were prepared by varying $\text{SA}/\text{C}_{12}\text{E}_{10}$ (SA is H_2SO_4 and $\text{C}_{12}\text{E}_{10}$ is 10-lauryl ether, $\text{C}_{12}\text{H}_{25}(\text{OCH}_2\text{CH}_2)_{10}\text{OH}$) mole ratio

from 2.5 to 9 (2.5 low, 6 intermediate, and 9 high) and keeping the $\text{SA}/\text{H}_2\text{O}$ mole ratio constant at 3.0. For example, the sample with a 9 $\text{SA}/\text{C}_{12}\text{E}_{10}$ mole ratio was prepared as follows: first, 9 mmol of SA (1.038 g, 85% SA) was added to 27 mmol of water (0.486 g) and stirred a few seconds by magnetic stirrer. Afterward, 1 mmol of $\text{C}_{12}\text{E}_{10}$ (0.626 g) was added to the above clear solution. The resulting mixture was homogenized by using a vortex mixer and then heated in the heat bath at 80°C for 48 h to obtain the clear gel (denoted as 9LC). The other two compositions, 2.5LC and 6LC , were prepared using 0.288 g SA, 0.135 g water, and 0.626 g $\text{C}_{12}\text{E}_{10}$ and 0.692 g SA, 0.314 g water, and 0.626 g $\text{C}_{12}\text{E}_{10}$, respectively, following the same procedure. For characterization, the gels were dissolved in water and then coated on glass slides via spin coating at 1000 rpm. For electrochemistry, the gel formed was used directly in a vial.

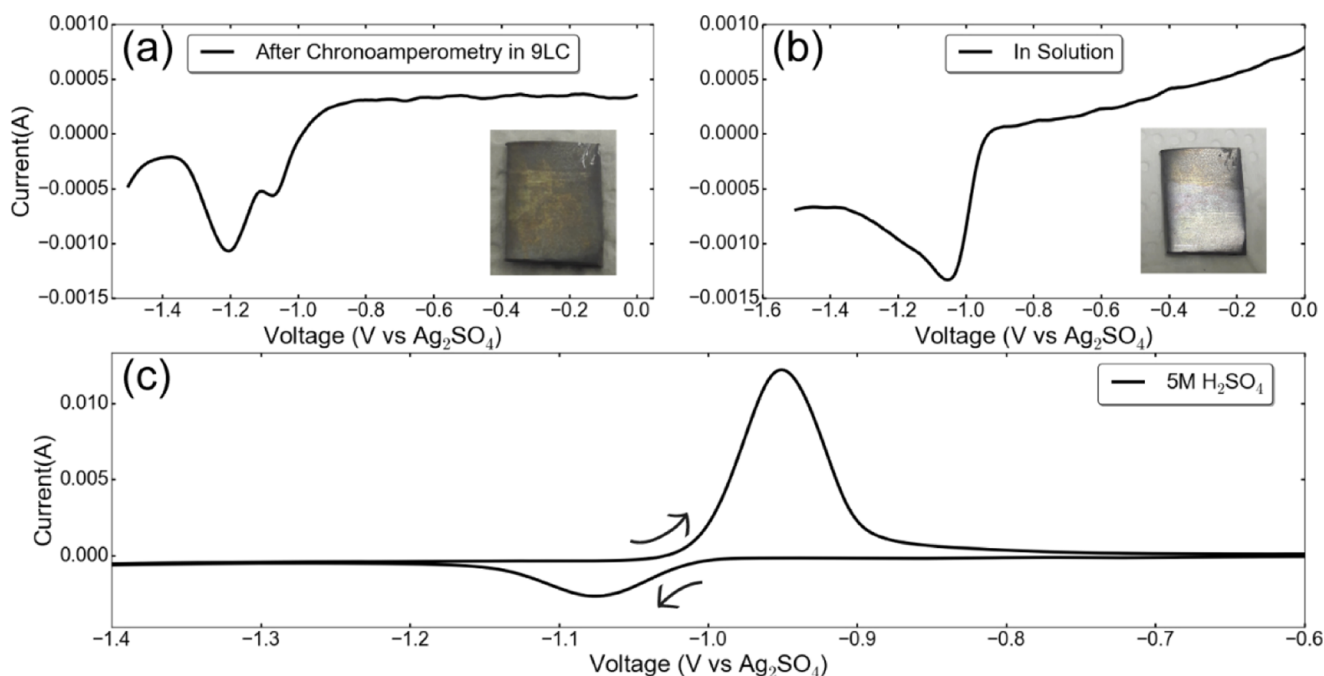


Figure 7. PbO–PbSO₄ film is developed on top of a lead strip in 9LC by 10 h CA, followed by LSV in aqueous H₂SO₄ (a). Another similar film is kept in aqueous H₂SO₄ for 24 h, followed by LSV (b). The CV is done after submerging in aqueous H₂SO₄ for 24 h (c).

Measurements. The XRD patterns were recorded using a Rigaku Miniflex diffractometer equipped with a high-power Cu K α source operating at 30 kV/15 mA and a wavelength of 1.5405 Å. The POM images were obtained in the transmittance mode by using a Zeiss Axio Scope.A1 polarizing optical microscope. The SEM images were recorded using a ZEISS EVO-40 SEM operated at 15 kV.

All of the electrochemical experiments were conducted on a Gamry Interface 5000E potentiostat. The Pt CVs were conducted in a three-electrode cell with a 6 mm diameter Pt WE, a graphite counter electrode (CE), and Ag/Ag₂SO₄ as a reference electrode (RE).

Polarization resistance experiments were conducted in a two-electrode geometry, using stainless steel electrodes fitted through a Teflon frit. Each stainless steel electrode had an area of 0.48 cm². Voltage sweeps were chosen to be ± 0.015 V versus open-circuit potential (E_{oc}) with a sweep rate of 0.10 mV/s. A 10 min equilibration time followed by 5 h experiment was done. The value of the self-exchange current was obtained by a linear fit to the data near 0 V (vs E_{oc}).

The Pb CVs were collected in a three-electrode cell with a Pb WE strip of 2 cm \times 6 cm, a graphite CE, and an Ag/Ag₂SO₄ as RE. The electrode area of the Pb WE was normalized by an epoxy resin such that only a 6 mm diameter circular region of Pb was exposed to the solution/liquid crystal. Moreover, the surface of the Pb WE was conditioned before each sweep experiment at sufficiently negative potentials to reduce all of the electrode surface. Reported Pb in H₂SO₄ CVs were, in all of the cases, in a steady state; at least 12 cycles were done and no appreciable change was observed in subsequent cycles.

Ag/Ag₂SO₄ RE was fabricated following a literature report.²⁸ Briefly, an Ag wire 1 mm in diameter and 3.2 mm long was oxidized under a constant current of 80 μ A for 3 h in a degassed 0.1 M Na₂SO₄. The formation of gray-white Ag₂SO₄ as opposed to the brown oxide was observed. Facile ion transport across the ion-exchange membrane of RE and liquid

crystal was confirmed by impedance measurements (measured $Z = 346.5 \Omega$). The stability of the fabricated electrode was confirmed by Pt in H₂SO₄ CVs over a 24 h period. The potential calibration was done by measuring and monitoring the open-circuit potential of a two-electrode system, in which the working side is a Ag/AgCl RE and the reference side is the Ag/Ag₂SO₄ RE ($E_{oc} = -0.407$ V).

Chronoamperometry experiments were done at various potentials for 1, 5, and 10 h for both 9LC and 5 M H₂SO₄ solutions after conditioning at appropriate negative potential and linear sweep (at 1 mV/s) to the potential of interest. A three-electrode setup consisting of Pb WE, graphite CE, and the previously prepared Ag/Ag₂SO₄ RE was used. The lead strips were washed gently with deionized water and the SEM images were taken.

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Notes

The authors declare no competing financial interest.

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REFERENCES

- (1) Çelik, Ö.; Dag, Ö. A new lyotropic liquid crystalline system: oligo(ethylene oxide) surfactants with $[M(H_2O)_n]X_m$ transition metal complexes. *Angew. Chem., Int. Ed.* **2001**, *40*, 3799–3803.
- (2) Albayrak, C.; Cihaner, A.; Dag, Ö. A new, highly conductive, lithium salt/nanoionics surfactant, lyotropic liquid crystalline meso-phase and its application. *Chem. – Eur. J.* **2012**, *18*, 4190–4194.

- (3) Tunkara, E.; Albayrak, C.; Polat, E. O.; Kocabas, C.; Dag, Ö. Highly proton conductive phosphoric acid-nonionic surfactant lyotropic liquid crystalline mesophases and application in graphene optical modulators. *ACS Nano* **2014**, *8*, 11007–11012.
- (4) Olutaş, E. B.; Balci, F. M.; Dag, Ö. Strong acid-nonionic surfactant lyotropic liquid crystalline mesophases as media for the synthesis of carbon quantum dots and highly proton conducting mesostructured silica thin films and monoliths. *Langmuir* **2015**, *31*, 10265–10271.
- (5) Ichikawa, T.; Yoshio, M.; Hamasaki, A.; Kagimoto, J.; Ohno, H.; Kato, T. 3D connected ionic nano-channels formed in polymer films: self-organization and polymerization of thermotropic cubic liquid crystals. *J. Am. Chem. Soc.* **2011**, *133*, 2163–2169.
- (6) Armand, M.; Endres, F.; Macfarlane, D. R.; Ohno, H.; Scrosati, B. Ionic-liquid materials for electrochemical challenges of the future. *Nat. Mater.* **2009**, *8*, 621–629.
- (7) Yazaki, S.; Funashashi, M.; Kagimoto, J.; Ohno, H.; Kato, T. Nanostructured liquid crystals combining ionic and electronic functions. *J. Am. Chem. Soc.* **2010**, *132*, 7702–7708.
- (8) Sakuda, J.; Hosono, E.; Yoshio, M.; Ichikawa, T.; Matsumoto, T.; Ohno, H.; Zhou, H.; Kato, T. Liquid-crystalline electrolytes for lithium-ion batteries: ordered assemblies of a mesogen-containing carbonate and a lithium salt. *Adv. Funct. Mater.* **2015**, *25*, 1206–1212.
- (9) Kobayashi, T.; Ichikawa, T.; Kato, T.; Ohno, H. Development of glassy bicontinuous cubic liquid crystals for solid proton-conductive materials. *Adv. Mater.* **2017**, *29*, No. 1604429.
- (10) Yilmaz, E.; Olutaş, E. B.; Barım, G.; Bandara, J.; Dag, Ö. Lithium salt-nonionic surfactant lyotropic liquid crystalline gel-electrolytes with redox couple for dye sensitized solar cells. *RSC Adv.* **2016**, *6*, 97430–97437.
- (11) Högberg, D.; Soberats, B.; Uchida, S.; Yoshio, M.; Kloo, L.; Segawa, H.; Kato, T. Nanostructured two-component liquid-crystalline electrolytes for high temperature dye-sensitized solar cells. *Chem. Mater.* **2014**, *26*, 6496–6502.
- (12) Garche, J.; Karden, E.; Moseley, P.; Rand, D. *Lead-Acid Batteries for Future Automobiles*, 1st ed.; Elsevier Science: Amsterdam, 2017; pp 3–25.
- (13) Defence Protection Act Title III, Submarine Valve Regulated Lead Acid Batteries. http://www.dpatitle3.com/dpa_db/project.php?id=198 (accessed Aug 5, 2017).
- (14) Davenport, W. G.; King, J. M. *Sulfuric Acid Manufacture*, 1st ed.; Elsevier Science: Amsterdam, 2005; pp 7–15.
- (15) Pavlov, D. *Lead-Acid Batteries: Science and Technology: A Handbook of Lead-Acid Battery Technology and Its Influence on the Product*, 1st ed.; Elsevier Science: Amsterdam, 2011; pp 29–47.
- (16) Culpin, B.; Rand, D. A. J. Failure modes of lead/acid batteries. *J. Power Sources* **1991**, *36*, 415–438.
- (17) Sawai, K.; Funato, T.; Watanabe, M.; Wada, H.; Nakamura, K.; Shiomi, M.; Osumi, S. Development of additives in negative active-material to suppress sulfation during high-rate partial-state-of-charge operation of lead-acid batteries. *J. Power Sources* **2006**, *158*, 1084–1090.
- (18) Stienecker, A. W.; Stuart, T.; Ashtiani, C. An ultracapacitor circuit for reducing sulfation in lead acid batteries for mild hybrid electric vehicles. *J. Power Sources* **2006**, *156*, 755–762.
- (19) Delahay, P.; Pourbaix, M.; Van Rysselberghe, P. Potential-pH diagram of lead and its applications to the study of lead corrosion. *J. Electrochem. Soc.* **1951**, *98*, 57–64.
- (20) Varma, B. P. Lead Acid Battery with Gel Electrolyte. U.S. Patent 6,143,967, March 2, 1982.
- (21) Lambert, D. W. H.; Greenwood, P. H. J.; Reed, M. C. Advances in gelled-electrolyte technology for valve-regulated lead-acid batteries. *J. Power Sources* **2002**, *107*, 173–179.
- (22) Zhan, D.; Velmurugan, J.; Mirkin, M. V. Adsorption/desorption of hydrogen on Pt nanoelectrodes: evidence of surface diffusion and spillover. *J. Am. Chem. Soc.* **2009**, *131*, 14756–14760.
- (23) Attard, G. S.; Bartlett, P. N.; Coleman, N. R. B.; Elliott, J. M.; Owen, J. R.; Wang, J. H. Mesoporous platinum films from lyotropic liquid crystalline phases. *Science* **1997**, *278*, 838–840.
- (24) Panossian, Z.; de Almeida, N. L.; de Sousa, R. M. F.; Pimenta, G. d. S.; Marques, L. B. S. Corrosion of carbon steel pipes and tanks by concentrated sulfuric acid: a review. *Corros. Sci.* **2012**, *58*, 1–11.
- (25) Rocca, E.; Steinmetz, J. Mechanism of passivation of Pb(Ca)-Sn alloys in sulfuric acid: role of tin. *Electrochim. Acta* **1999**, *44*, 4611–4618.
- (26) Hirasawa, T.; Sasaki, K.; Taguchi, M.; Kaneko, H. Electrochemical characteristics of Pb-Sb alloys in sulphuric acid solutions. *J. Power Sources* **2000**, *85*, 44–48.
- (27) Czerwiński, A.; Żelazowska, M.; Grdeń, M.; Kuc, K.; Milewski, J. D.; Nowacki, A.; Wójcik, G.; Kopczyk, M. Electrochemical behavior of lead in sulfuric acid solutions. *J. Power Sources* **2000**, *85*, 49–55.
- (28) Velický, M.; Tam, K. Y.; Dryfe, R. A. W. On the stability of the silver/silver sulfate reference electrode. *Anal. Methods* **2012**, *4*, 1207–1211.