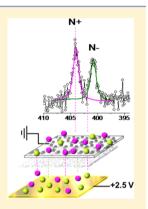
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In Situ XPS Reveals Voltage Driven Asymmetric Ion Movement of an Ionic Liquid through the Pores of a Multilayer Graphene Electrode

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ABSTRACT: Under application of a voltage bias, asymmetric ion movement of an ionic liquid (IL) through a multilayered graphene (MLG) electrode has been detected by X-ray photoelectron spectroscopy, via recording the intensity of the two nitrogen peaks. Accordingly, we observe that upon increasing the bias, the two peaks representing the cationic and anionic fragments of the IL start appearing with increasing intensity, together with an asymmetry in their ratio, differing from unity by about 10%. Bias-dependent binding energy shifts followed through atomic features of the IL (F 1s, N 1s, and C 1s) and the graphene electrode (C 1s) indicate that a distinct solid—liquid interface develops throughout the entire intercalation process with an additional and pertinent evidence for finite potential drops across the two electrical double layers. This evidence is bolstered by the fact that the measured binding energy difference between the F 1s of the liquid and C 1s peak of the semisolid MLG electrode is only about half of the applied bias, where the rest of the applied voltage is screened by the two electrical double layers at the solid—liquid interfaces between (i) the MLG–IL and (ii) IL—bottom metal electrode. A simple electrostatic estimation indicates that even this small 10% ion imbalance would lead to 4 orders of magnitude larger voltage development between the IL and the



MLG phases and suggests the need for amendment(s) to the current understanding of the dielectric description of ILs.

■ INTRODUCTION

Charge distribution, storage, and movement in solutions and soft matter are of paramount importance for understanding and rational use of various electrochemical concepts toward advanced applications in energy and chemical, biochemical, microfluidics, and sensing applications. Over the last 3 decades, the vast research efforts using advanced measurement and computational and simulation techniques have changed the concept of electroneutrality in many physicochemical processes taking place in solution surfaces or in pores of solid materials.¹

Being a chemically sensitive and quantitative surface analysis technique, X-ray photoelectron spectroscopy (XPS) has been the pivotal method to extract information about distribution of cations and anions on various surface structures once difficulties in handling liquids had been overcome through multiple advancements in experimental techniques. The pioneering XPS investigation of ion enrichment on surfaces of viscous liquids by H. Siegbahn^{2–5} was later successfully extended to aqueous solutions in the form of continuous jets using synchrotron-based photoemission spectroscopy by Faubel and co-workers.⁶ In parallel, developments in ambient pressure XPS have enabled investigation of a large variety of critical materials and chemical processes.^{7,8} All of these giant leaps and developments have also been successfully supported and guided by extensive molecular dynamics simulations.^{9,10}

Another important scientific advancement has emerged from a completely different focus area, investigation of porous materials by NMR, another quantitative spectroscopic technique for investigation of electrochemical energy generation and storage systems.^{11–16} In all of these studies, it was explicitly shown that the anion/cation ratio strongly deviated from stoichiometry, depending on their chemical nature (i.e., specific ion effects) and was also controlled by the polarity of the applied electrical fields. However, although implied, local breakdown of the electroneutrality had not been mentioned explicitly, except in the work of Luo et al.¹³ Different aspects of these experimental and theoretical developments have recently been reviewed.^{17–19}

Nonvolatile room-temperature ionic liquid (IL) electrolytes have allowed us and others to utilize lab-based XPS instruments for investigating various electrochemical processes under ultrahigh vacuum conditions, without the need for extensive pumping techniques or synchrotron facilities.²⁰⁻³⁰ In one of our recent studies, we have used a multilayered graphene (MLG) electrode on paper to fabricate an optically reconfigurable device using an ionic liquid as the electrolyte,³ which was also adopted for terahertz radiation³² or flexible electrochromic devices³³ and tunable fractal metasurfaces.³⁴ In this contribution, we report on using a similar multilayered graphene as the top electrode and utilize XPS to monitor in situ (i) the changes in the anion/cation intensity ratio under applied electric fields and (ii) the electrical potential developments on different surface structures, which are derived from the shifts in the binding energies of the corresponding atomic core levels in a chemically resolved fashion.^{35,3}

EXPERIMENTAL SECTION

A porous polyethylene membrane (PEM) separator is used to allow the IL to pass but prevent shorting of the top MLG and

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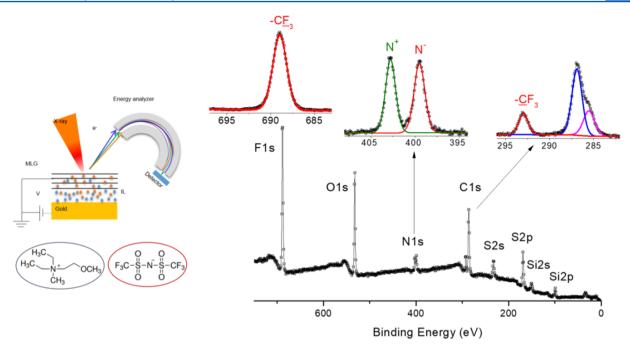


Figure 1. Experimental setup used for the operando-XPS in the back-gate geometry. Chemical structure of the ionic liquid and its survey and region's XP spectra.

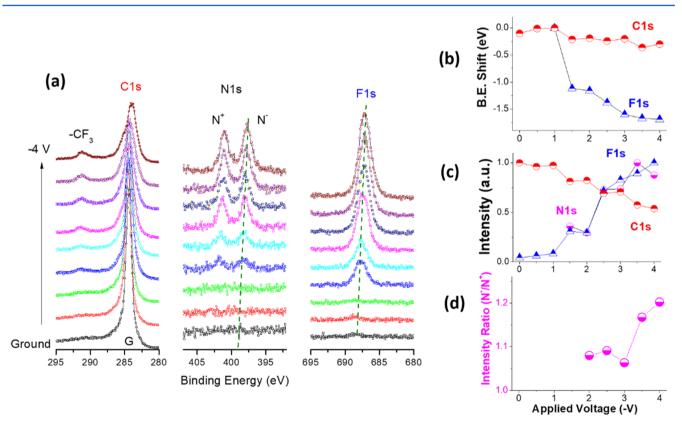


Figure 2. (a) XP spectra of the surface of the device under bias voltages changing from 0 to -4 V. (b) Normalized intensities of C 1s, N 1s, and F 1s. (c) Variation in the N⁻/N⁺ intensity ratio. (d) Variation in the binding energy of C 1s, representing the MLG and F 1s and N 1s representing the IL.

bottom gold electrodes. Multilayer graphene samples are grown on nickel foils (Alfa Aesar) using chemical vapor deposition at temperatures ranging from 850 to 1000 $^{\circ}$ C at ambient pressure. A mixture of H₂, Ar, and CH₄ gases is used during the growth, with flow rates set as 100, 100, and 30 sccm, respectively, and the growth time is 5 min. This procedure enables fabricating multilayer graphene samples having 100–250 layers.³³ The MLG electrode is transferred onto the PEM, which is directly placed on the gold electrode, and the ionic liquid is initially introduced in between the PEM and the bottom electrode. A Thermo Fisher K-Alpha X-ray photoelectron spectrometer with a monochromatized photon energy of 1486.6 eV has been used

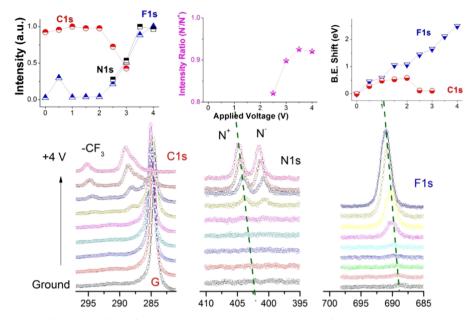


Figure 3. XP spectra recorded from the surface of the device under bias voltages changing from 0 to +4 V.

to collect data for all measurements. Direct current external bias is applied to either the MLG or the bottom electrode using a Keithley 2400 source-meter electrical signal. Data processing is carried out using the Avantage software package provided by the manufacturer.

RESULTS AND DISCUSSION

The ionic liquid used, [N,N-diethyl-N-methyl-N-(2methoxyethyl)ammonium bis(trifluoromethanesulfonyl)imide] DEME-TFSI, has two nitrogen moieties, one representing the cation and the other the anion. The corresponding N 1s core levels are conveniently separated in the energy scale by 3.2 eV. We also utilize the unique property of XPS to monitor the electrical potentials developed locally since the relevant data are derived from the measured kinetic (or binding) energy of the emitted photoelectrons, which are directly affected by the developed or externally applied electrical fields.^{29,30} XPS has long been successfully used to monitor various in situ processes, such as oxidation, reduction,^{21-24,37-41} intercalation,⁴²⁻⁴⁸ overpotential developments,⁴⁹ etc. In this work, we report a direct observation of a stoichiometric imbalance or asymmetric ion movement by XPS, complemented with a potential drop measurement, under the influence of an applied electric field.

Deviations from the stoichiometry or charge imbalances in surface structures of materials containing ionic liquids by XPS have long been reported and discussed, mostly in terms of chemical compositions, interactions, or reactions.^{22,23,27,40,49–53} Our strategy is different since we monitor the changes under working conditions (in Operando) in electrochemical device(s). Figure 1 shows the experimental setup and the schematics of the device together with a survey spectrum of the neat IL, where the stoichiometric cation/anion ratio is spectroscopically validated by the two equally intense N 1s peaks representing the cation and anion, respectively.

In Figure 2, we show the spectral evolution of the various components, under increasing imposed bias potentials (Φ_B). At the beginning, only the C 1s peak of the graphene is observed as the major component, with the F 1s as a very weak one, most probably due to contamination. Starting from -1.5 V and

onward, peaks representative of the IL become observable with a continuous increase in their intensity, till the start of electrochemical decomposition of the IL around -4.0 V, easily noticeable with an increase in the spectrometer background pressure. These spectral evolutions provide a wealth of information about the operation of the device and composition of the analyzed regions.

Appearance of N 1s and F 1s peaks after 1.5 V indicates the onset of the intercalation process and the threshold voltage. This onset voltage is comparable to the same value determined using both electrochemical and XPS measurements for perchlorate ion intercalation in highly oriented pyrolytic graphite from aqueous solutions.^{46,47} Since XPS probes the very top surface (~5–6 nm), appearance of F 1s and N 1s peaks shows that the ions can efficiently intercalate the thick active surface (>100 graphene layers). The intensity of C 1s decreases with increasing voltage due to the partial coverage of the top surface with the IL. The C 1s peak of the –CF₃ group associated with IL also appears after the threshold voltage.

Although the graphene surface is grounded, the binding energy of C 1s also experiences small shifts with the applied bias, from 284.37 to 283.67, most likely due to the shift in the Fermi energy of graphene.⁵⁴ Interestingly, we also observe cointercalation of anions and cations of the ionic liquid with a measurable ion/charge imbalance (>10% change in the ratio of N⁻ to N⁺). This charge imbalance (due to mobile and quasiindependent ions) is also responsible for electrostatic doping on graphene layers. When we apply positive bias voltage, the charge imbalance is reversed (Figure 3).

Implicit in our arguments is a picture of the MLG electrode as a loose but conducting network of carbon atoms electrically connected to each other and to the grounding wire and the ionic liquid as the second but less conducting network of electrolyte medium positioned between the two electrodes. The grounding wire brings the Fermi level of the top electrode to that of the spectrometer, but the Fermi level of the bottom electrode is shifted with the applied bias potential (Φ_B).^{55–57} As a result, if we were able to measure the kinetic energy of any core level representative of the bottom electrode (not possible due to the limited XPS probe depth, <6 nm), a complying shift

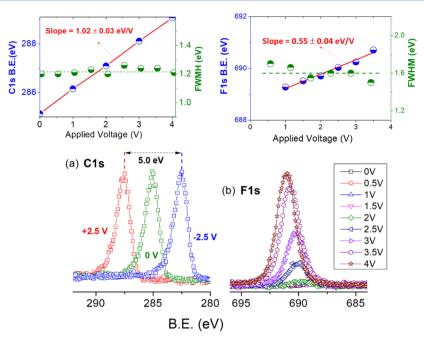


Figure 4. (a) C 1s region of a few-layer graphene on Cu, recorded under +2.5, 0, and -2.5 V bias, where the shifts in the binding energy positions comply exactly with the applied bias. (b) Details of the F 1s region recorded from the surface of the device under bias, changing from 0 to +4 V (same as those in Figure 2). Insets show linear fits (red line) of the position (blue dots) and the full-width half-maximum of the C 1s (graphene on Cu) and F 1s (IL on MLG electrode) peaks (green dots) against the applied bias. For the linear fit of F 1s, points at 0 V (surface impurity) and 4 V (decomposition) are excluded.

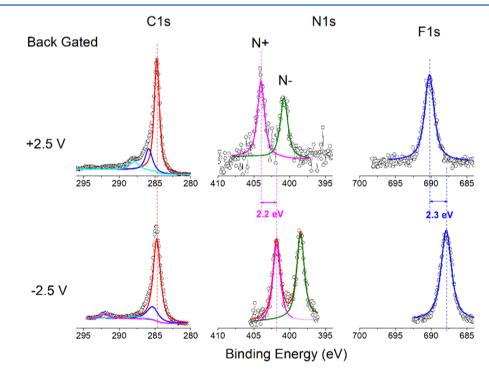


Figure 5. C 1s, F 1s, and N 1s regions of the intercalated thick graphene layer with the IL at +2.5 and -2.5 V back-gate voltages.

of $-e\Phi_{\rm B}$ (in eV) would have been measured, where *e* is the electron's charge. A negative bias accelerates the outgoing photoelectrons, resulting in an apparent decrease (BE = $\hbar\nu$ – KE) in the measured binding energy position, and the positive bias works in the opposite direction, as shown in Figure 4a for a few-layer graphene grown on Cu. On the other hand, peaks belonging to the IL moieties (F 1s, N 1s, and C 1s), intercalated in the grounded graphene layers, but also in

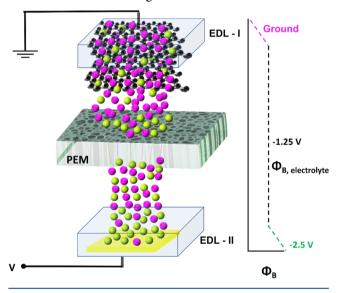
electrical contact with the bottom Au electrode, display shifts with the bias but not fully complying with it $(-e\Phi_{B,electrolyte})$, as depicted in Figure 4b by the recorded F 1s peaks under backgated positive bias.

A linear fit to the data of Figure 4b gives a slope of $0.55 \pm 0.04 \text{ eV/V}$, very close to half of the applied bias, which we assign to the electrical potential of the liquid medium. Furthermore, and as also shown in the same figure, no

additional broadening of the peaks is observed, attesting to the uniformity of the potentials developed on the liquid surface.³⁶ As discussed in detail in our previous work,²⁹ this finding reflects directly the efficient screening through the formation of two electrical double layers (EDLs) at both solid–liquid interfaces.^{54,55} Entirely similar findings were also obtained when XPS data was recorded under application of the bias through the top MLG electrode, reflecting the consistency of our measurements.

As a further proof of the observed noncomplying shifts of the liquid phase, we show in Figure 5 the difference in the N 1s and F 1s peaks under positive and negative 2.5 V biases, which are measured as 2.2 and 2.3 eV, slightly less than half of the bias difference of 5.0 eV [see Figure 4a], between the MLG–IL and IL–Au-bottom electrodes. The voltage drops on the two EDLs are slightly unequal (the average of the two measured shifts, 2.25 eV and 5.0 - 2.25 = 2.75 eV) due to the different capacitances of the two EDLs and reflect that the MLG–IL interface has a slightly larger capacitance (2.75/2.25 = 1.2) compared to that of the IL–Au electrode due most probably to the larger and fractal contact area between the MLG electrode and the IL, as depicted in Scheme 1.

Scheme 1. Schematics of Ion and Potential Distributions under -2.5 V Back Gating



The quantitative nature of XPS enables us to also obtain an order of magnitude estimation of the extent of the intercalation of the ionic liquid in the multilayer graphene at any potential applied. For example, using the intensities of the F 1s and C 1s at +2.5 V, we get an atomic ratio of F/C = 0.053. Taking into account the chemical formula of the IL $(C_{10}F_6)$, one gets approximately one ion pair for ~ 110 C atoms of the graphene. Hence, the $\sim 10\%$ excess positive charge derived from the nitrogen intensity corresponds to one excess positive charge for every ~1100 C atoms of the graphene layers. Using approximately a circular X-ray spot size of 200 μ m and assuming the probe depth of 6 nm, the analyzed volume can be estimated as $[4\pi \times (200 \ \mu m)^2 \times 6 \ nm] = 3 \times 10^{-9} \ cm^3$. Assuming further that density of the MLG layer is close to that of graphite (2 g/cm^3) , we can estimate that the analyzed volume contains approximately 5×10^{-8} mol of carbon atoms to yield 10^{-12} mol or 3×10^{11} excess ions and a surface charge density (SCD) of 0.2 C/m². This number is in the same order of magnitude as that of the SCD derived from the surface potentials measured using XPS of aqueous silica colloidal nanoparticles charged upon changing the pH of the solution from 0.3 to 10.0.^{56,57} However, the same SCD is at odds with the estimation of the electrical potential development on the layer, as shown below. Since the device geometry adopted is similar to that of a parallel plate capacitor of a thickness d (~ 1 mm), containing mostly the IL as the dielectric (with a dielectric constant of $\varepsilon \sim 20$),⁵⁸ the electrical potential Φ can be obtained as

$$\Phi = \frac{q}{4\pi\varepsilon\varepsilon_0 d} = \frac{3 \times 10^{11} \times 1.6 \times 10^{-19} \text{ C}}{4\pi \times 20 \times 8.85 \times 10^{-12} \frac{F}{m} \times 10^{-3} \text{ m}}$$

\$\approx 2 \times 10^4 V\$

This simple order of magnitude estimation gives an unreasonably large potential. As discussed above, the potential difference developed between the IL medium and the graphene layers (near ground) is measured as ~ 1.2 V due to the doublelayer screening²⁹ and is 4 orders of magnitude lower than the estimated one. Although the 10% extra charge we are reporting is in agreement with the findings of MacFarlane et al. through conductivity/viscosity correlations,⁵⁹ for the same number to yield a reasonable voltage on the order of few volts, this simple dielectric picture needs to be amended. Either the local dielectric constant is much higher or there are other effects changing the understanding of the system. Ionicity and overcrowding have been heavily discussed in the recent literature, which can be contributing factors to the low voltage that is induced by a high charge excess.¹⁷ In that, the structure is such that most of the excess ions are screened by the rest and do not induce extra voltage. This violates the assumption made in the above order of magnitude estimate where excess charges are homogeneously distributed within the volume.

CONCLUSIONS

In summary, two important assets of XPS have been utilized: (i) quantification of the nonstoichiometric ion movement/ intercalation through the pores of a multilayer graphene top electrode in response to voltage bias and (ii) mapping of the electrical potential developments on the graphene electrode intercalated with the IL, in a totally noninvasive and chemically specific fashion. Although only the potential difference between the grounded top graphene electrode and the surface IL moieties is measured, as functions of the voltage bias, the outcome, guided also by our earlier findings, is claimed to reflect the voltage drops across the two electrical double layers between the ionic liquid and the top and bottom electrodes. Furthermore, using the measured ion enrichment of about 10%, it is estimated that the electrical potential developed would be 4 orders of magnitude larger than the measured value of ~ 1.2 V, which casts doubts on the traditional approach of treating ILs as simple dielectric media. All of these findings are new and are expected to impact both the fundamental understanding of certain electrochemical concepts and more importantly better usage of electrode/electrolyte systems for applications in energy harvesting and storage, since the geometry of the device used in this work has a close similarity, in both chemical makeup and geometry, to the electrical double-layered supercapacitors.

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Notes

The authors declare no competing financial interest.

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ABBREVIATIONS

DEME-TFSI, *N*,*N*-diethyl-*N*-methyl-*N*-(2-methoxyethyl)ammonium bis(trifluoromethanesulfonyl)imide; EDL, electrical double layer; RTIL, room-temperature ionic liquid; MLG, multilayered graphene; SCD, surface charge density; XPS, X-ray photoelectron spectroscopy

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