

# Morphological Control of Mesoporosity and Nanoparticles within $\text{Co}_3\text{O}_4$ –CuO Electrospun Nanofibers: Quantum Confinement and Visible Light Photocatalysis Performance

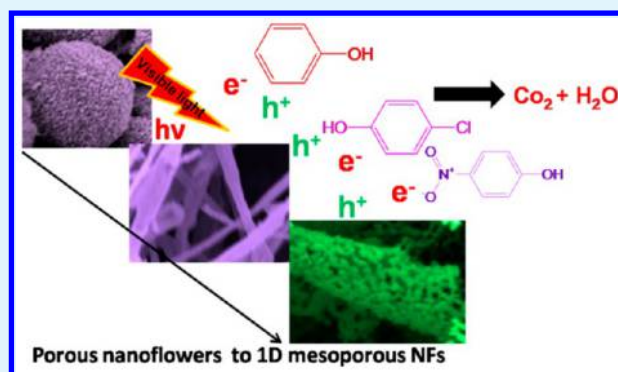
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## S Supporting Information

**ABSTRACT:** The one-dimensional (1D) mesoporous and interconnected nanoparticles (NPs) enriched composite  $\text{Co}_3\text{O}_4$ –CuO nanofibers (NFs) in the ratio Co:Cu = 1/4 ( $\text{Co}_3\text{O}_4$ –CuO NFs) composite have been synthesized by electrospinning and calcination of mixed polymeric template. Not merely the mesoporous composite  $\text{Co}_3\text{O}_4$ –CuO NFs but also single mesoporous  $\text{Co}_3\text{O}_4$  NFs and CuO NFs have been produced for comparison. The choice of mixed polymer templates such as polyvinylpyrrolidone (PVP) and polyethylene glycol (PEG) for electrospinning is responsible for the formation of 1D mesoporous NFs. The HR-TEM result showed evolution of interconnected nanoparticles (NPs) and creation of mesoporosity in all electrospun NFs. The quantum confinement is due to NPs within NFs and has been proved by the surface-enhanced Raman scattering (SERS) study and the UV–vis–NRI diffuse reflectance spectra (DRS). The high intense photoluminescence (PL) spectra showing blue shift of all NFs also confirmed the quantum confinement phenomena. The lowering of PL spectrum after mixing of CuO in  $\text{Co}_3\text{O}_4$  nanofibers framework ( $\text{Co}_3\text{O}_4$ –CuO NFs) proved CuO as an efficient visible light response low cost cocatalyst/charge separator. The red shifting of the band gap in composite  $\text{Co}_3\text{O}_4$ –CuO NFs is due to the internal charge transfer between  $\text{Co}^{2+}$  to  $\text{Co}^{3+}$  and  $\text{Cu}^{2+}$ , proved by UV–vis absorption spectroscopy. Creation of oxygen vacancies by mixing of CuO and  $\text{Co}_3\text{O}_4$  also prevents the electron–hole recombination and enhances the photocatalytic activity in composite  $\text{Co}_3\text{O}_4$ –CuO NFs. The photocurrent density, Mott–Schottky (MS), and electrochemical impedance spectroscopy (EIS) studies of all NFs favor the high photocatalytic performance. The mesoporous composite  $\text{Co}_3\text{O}_4$ –CuO NFs exhibits high photocatalytic activity toward phenolic compounds degradation as compared to the other two NFs ( $\text{Co}_3\text{O}_4$  NFs and CuO NFs). The kinetic study of phenolic compounds followed first order rate equation. The high photocatalytic activity of composite  $\text{Co}_3\text{O}_4$ –CuO NFs is attributed to the formation of mesoporosity and interconnected NPs within NFs framework, quantum confinement, extended light absorption property, internal charge transfer, and effective photogenerated charge separations.

**KEYWORDS:** electrospinning, nanofibers, quantum confinement, mesoporous, phenolic compounds, visible light



## 1. INTRODUCTION

Semiconductor photocatalysis, a promising green technology, has attracted significant attention because of its high performance in environmental pollution control.<sup>1,2</sup> Phenolic compounds are toxic organic byproducts produced during the manufacturing process of agrochemicals, dyes, and pharmaceuticals.<sup>3,4</sup> Most of these compounds have been listed as toxic priority pollutants by the European community and U.S. (Environmental Protection Agency).<sup>5</sup> Nanostructured materials, especially metal and metal oxide nanoparticles, nanoplates, nanorods, nanotubes, and nanofibers, have attracted particular interest in catalytic applications because of their unique optical properties, versatile catalytic activities, and high surface areas.<sup>6–8</sup> Among them, one-dimensional (1D) structures have a great deal of attention due to their superior charge transport

properties, few grain boundaries, and the quick ion diffusion at the semiconductor–phenolic compounds solutions interface.<sup>9,10</sup> 1D nanostructures such as metal oxide nanofibers (NFs) by electrospinning have remarkable characteristics, e.g., high porosity, a large surface area to volume ratio, and excellent substrates for secondary nanostructures.<sup>11,12</sup> The metal oxide  $\text{Co}_3\text{O}_4$  is a visible light active p-type semiconductor (1.48 eV) with attractive electronic and structural properties.<sup>13</sup> But bulk  $\text{Co}_3\text{O}_4$  has limitation in photocatalytic activity. In order to make  $\text{Co}_3\text{O}_4$  an effective photocatalyst, it is necessary to make a heterojunction with other semiconductor, resulting in

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enhanced photocatalytic activity due to lowering of electron–hole recombination.<sup>14,15</sup> Hence, mixture of  $\text{Co}_3\text{O}_4$  with other semiconductor oxides could fulfill the electron–hole separation strategy. The other strategy is morphology and structural framework of  $\text{Co}_3\text{O}_4$ . Recently,  $\text{Co}_3\text{O}_4$  nanofibers fabricated from spiral electrospinning have been utilized as a catalyst for the oxidation of formaldehyde.<sup>16</sup> The best option is to choose semiconductor oxides having cocatalyst property. A cocatalyst can enhance the photocatalytic activity of a semiconductor by (i) suppressing the electron–hole recombination by trapping the photogenerated electrons, (ii) improving the photostability of the catalyst by consumption of photogenerated charges, particularly holes, and (iii) lowering the activation energy and hence catalyzing the reactions.<sup>17</sup> Transition metal oxide semiconductor CuO (1.2 eV) has been reported as cocatalyst to enhance the photocatalytic activity. Shah et al. observed high photocatalytic activity of  $\text{Ag}/\text{AgCl}/\text{TiO}_2$  by using CuO as cocatalyst.<sup>18</sup> Photodegradation activity of organic substance by  $\text{WO}_3$  has been enhanced by using CuO as cocatalyst.<sup>19</sup> Hence, design and combination of semiconductor  $\text{Co}_3\text{O}_4$  and semiconductor CuO cocatalyst NFs could improve the visible light photocatalytic performance. The design and mixed modification of  $\text{Co}_3\text{O}_4$  and CuO include (i) morphological modification such as mesoporosity and NPs within NFs, (ii) quantum confinement effect, and (iii) combination of reactive facets of two oxides NFs.

Construction of mesoporous structure within NFs can also enhance the photocatalytic performance.<sup>16</sup> The construction of mesoporous structures in single  $\text{Co}_3\text{O}_4$  and CuO and in mixture of both can elevate the surface area of the photocatalyst which results in the increase of a large number of reactive sites. It can also enhance the light absorption efficiency because of more photons being distributed onto the surface of the photocatalyst, using the pores as light transfer paths.<sup>20</sup> Moreover, mesoporous materials themselves can be used as a support of guest materials.<sup>21</sup> The surface texture of  $\text{Co}_3\text{O}_4$ , CuO, and mixture of both NFs can be constructed by adopting electrospinning process. It has been noted the electrospinning process aids the creation of porous and well-ordered NFs by adding suitable optimized block copolymer template and other additives like acid and appropriate solvent.<sup>22,23</sup> Hence, mixing of polyvinylpyrrolidone (PVP) and polyethylene glycol (PEG) solutions as mixed polymeric template may fulfill the porous NFs/porous NPs within NFs strategy. The vital benefit of electrospinning process is the formation of surface hydroxyl ( $-\text{OH}$ ), which also plays a significant role in its photocatalytic activity.<sup>24</sup>

Hence, formation of mesoporosity within the composite  $\text{Co}_3\text{O}_4$ –CuO NFs could enhance the surface reactive sites and light absorption property. Furthermore, quantum confinement effects generated from quantum dots have been investigated intensively.<sup>25</sup> When the crystal size of the semiconductor is close to its exciton Bohr radius, its band gap will be enlarged while the crystal size is decreased.<sup>25</sup> So formation of NPs within NFs could provide the quantum confinement phenomena and will lead to enhanced light absorption property. Moreover, the combination of reactive facets of  $\text{Co}_3\text{O}_4$  and CuO NFs can act as an efficient composite photocatalysts for phenolic compound degradation. It has been noted that the mesoporous  $\text{Co}_3\text{O}_4$  exposed  $\{110\}$  facets showed high activity in oxidation of ethylene.<sup>26</sup> Xie et al.<sup>27</sup> observed that the  $\text{Co}_3\text{O}_4$  nanorods, which predominantly exposed their  $\{110\}$  facets, favoring the presence of active  $\text{Co}^{3+}$  species at the surface, exhibited a much

higher activity for CO oxidation. Therefore, the synthesis design of nanostructured  $\text{Co}_3\text{O}_4$  catalysts with highly reactive crystal facets is a key to exploring different catalytic properties and applications. Thus, fabrication of porous NFs could be the better option for the generation of highly reactive  $\{110\}$  facets. Not merely  $\text{Co}_3\text{O}_4$  facets but also CuO exposed with  $\{001\}$  facets have high reactive sites for gas sensing and Li-ion battery application.<sup>28</sup> So combination of both mesoporous  $\text{Co}_3\text{O}_4$  and CuO NFs with their respective reactive facets could enhance the photocatalytic properties.

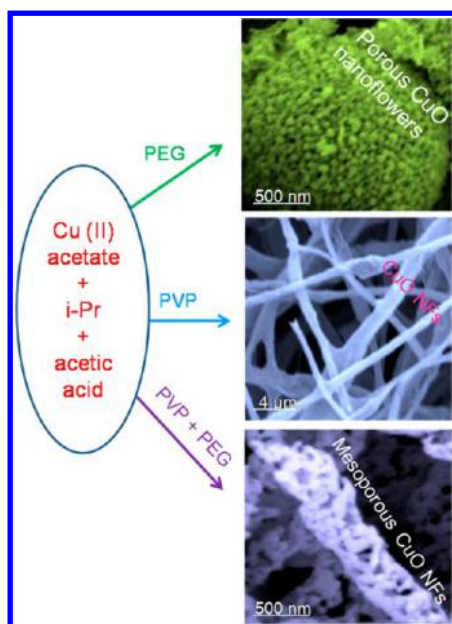
In this article, we report the practical approach for the fabrication of mesoporous  $\text{Co}_3\text{O}_4$  NFs, CuO NFs, and composite  $\text{Co}_3\text{O}_4$ –CuO NFs by in situ electrospinning process. The mixed polymeric templates (PVP and PEG) have taken the key role for the fabrication of above-mentioned electrospun mesoporous NFs ( $\text{Co}_3\text{O}_4$  NFs, CuO NFs) and  $\text{Co}_3\text{O}_4$ –CuO composite NFs. Here, we have explored the mechanistic pathways for the formation of mesoporous and NPs within all fabricated electrospun NFs. It is investigated that the quantum confinement effect arises due to the formation of NPs within NFs, proved by SERS analyses. We have also explored the mechanism of the photocatalytic phenolic compound degradation process and the quantum confinement induced visible light photocatalytic activity of mesoporous composite  $\text{Co}_3\text{O}_4$ –CuO NFs. In the present study, it is also explained that the role of CuO is as a cocatalyst which enhanced the photocatalytic activity in the composite  $\text{Co}_3\text{O}_4$ –CuO NFs by separation of charge carriers and suppress their recombination. Under the combined action of  $\text{Co}_3\text{O}_4$  and CuO, the as-synthesized mesoporous composite  $\text{Co}_3\text{O}_4$ –CuO NFs exhibits the highest photocatalytic activity compared with the other two single mesoporous NFs. This is explained by the enhanced morphology, textural property, and optical properties.

## 2. EXPERIMENTAL SECTION

**2.1. Materials.** All the chemicals and reagents are of analytical grade and used without further purification. Cu(II) acetate (98%, Aldrich), Co(II) acetate (98%, Aldrich), polyethylene glycol (PEG, Mw = 1000 g/mol, Sigma-Aldrich), polyvinylpyrrolidone (PVP, Mw = 1 300 000 g/mol, Aldrich), isopropanol (95.5%, Sigma-Aldrich), acetic acid (100%, Sigma), 4-chlorophenol (99%, Alfa Aesar), 4-nitrophenol (99%, Alfa Aesar), phenol (99%, Alfa Aesar), nitric acid (Sigma), ammonia (Sigma), rhodamine 6G (98%, Sigma), potassium iodide (98%, Aldrich), and *p*-benzoquinone (98.5%, Alfa Aesar) were used in the present study.

**2.2. Electrospinning of Mesoporous CuO NFs, CuO NFs, and Porous CuO Nanoflower.** In order to optimize the mesoporous CuO NFs strategy, the Cu(II) acetate was taken as precursor of CuO. The experimental procedure proceeded by taking 0.5 g of Cu(II) acetate dissolved into the mixture of 1 g of PVP and 0.5 g of PEG (PVP:PEG = 3:1) in 10 mL of isopropanol. The reaction solution was stabilized by adding 1 mL of acetic acid. The whole solution was stirred for 24 h prior to electrospinning. Further, the solution was filled in a 3 mL plastic syringe with needle diameter of 0.4 mm and placed on a syringe pump (KDS101, KD Scientific, USA). The flow rate of the polymer solution was controlled by the syringe pump at fixed rate of 0.5 mL h<sup>-1</sup>. The electric field (15 kV) was applied from a high voltage power supply (Spellman, SL series, USA). The sample was collected on aluminum foil which was grounded by metal collector at a distance 17 cm from the needle tip. Electrospinning procedure was carried out at 23 °C and 18% relative humidity in a Plexiglas box. Finally, the mesoporous CuO NFs were fabricated by calcining the PVP/CuO/PEG composite nanofibers at 450 °C for 4 h in air. The materials and the representative SEM images are summarized in the Scheme 1. Similarly, CuO NFs were prepared by taking only PVP as polymer template. The other parameters like concentration of

**Scheme 1. Electrospayed Porous CuO Nanoflowers (Polymeric Matrix, PEG Only), Electrospun CuO NFs (Polymeric Matrix, PVP Only), and Mesoporous CuO NFs (Polymeric Matrix, PEG/PVP Mixture) Obtained after Calcination at 450 °C for 4 h**



isopropanol, acetic acid and amount of Cu(II) acetate were similar to those of mesoporous CuO NFs. The CuO NFs were prepared by calcining PVP/CuO at 450 °C for 4 h. Moreover the porous CuO nanoflowers were prepared only by using the PEG as a polymeric template. Since PEG is a low molecular weight matrix, electrospaying was occurred instead of electrospinning in which particles were formed instead of fibers. The concentrations of isopropanol, acetic acid, and amount Cu precursor were similar to those for mesoporous CuO NFs and CuO NFs. Finally, porous CuO nanoflowers were produced by calcining PEG/CuO at 450 °C for 4 h (Scheme 1). Thus, it was understood that the mixture of PVP and PEG was responsible for the formation of mesoporous CuO NFs whereas the use of only PVP and PEG yielded the formation of CuO NFs, and porous CuO nanoflowers, respectively. In order to adopt the mesoporous NFs strategy, the other oxides such as mesoporous  $\text{Co}_3\text{O}_4$  NFs and mesoporous composite  $\text{Co}_3\text{O}_4$ -CuO NFs have been prepared by using the mixture of PVP/PEG matrix, and the details are given in the following section.

**2.3. Electrospinning of Mesoporous  $\text{Co}_3\text{O}_4$  NFs and Mesoporous Composite  $\text{Co}_3\text{O}_4$ -CuO NFs.** For the synthesis of mesoporous  $\text{Co}_3\text{O}_4$  NFs and mesoporous composite  $\text{Co}_3\text{O}_4$ -CuO NFs, the optimal conditions/parameters were applied, like the electrospinning of mesoporous CuO NFs, i.e., using mixed PVP/PEG polymeric matrix. The mesoporous  $\text{Co}_3\text{O}_4$  NFs are fabricated by taking 0.5 g of cobalt(II) acetate as a precursor. The synthesis process was followed in a similar way as mesoporous CuO NFs. Finally, the  $\text{Co}_3\text{O}_4$  NFs were formed by calcined the PVP/ $\text{Co}_3\text{O}_4$ /PEG composite at 450 °C for 4 h in air. The mesoporous composite  $\text{Co}_3\text{O}_4$ -CuO NFs were fabricated by electrospinning method by taking a mixture of Co(II) acetate and Cu(II) acetate as a precursor, respectively. Composite  $\text{Co}_3\text{O}_4$ -CuO NFs (Co:Cu = 1/4) were synthesized by taking a calculated amount of Co(II) and Cu(II) acetate in the mixture of PVP:PEG = 3:1, in isopropanol and acetic acid. The whole solution was stirred for 24 h before electrospinning. Lastly, the mesoporous composite  $\text{Co}_3\text{O}_4$ -CuO NFs were fabricated by the calcinations of PVP/ $\text{Co}_3\text{O}_4$ -CuO/PEG composite at 450 °C for 4 h in air. The mesoporous composite  $\text{Co}_3\text{O}_4$ -CuO NFs in the ratio Co:Cu = 1/4 are denoted as  $\text{Co}_3\text{O}_4$ -CuO NFs. In a similar way,  $\text{Co}_3\text{O}_4$ -CuO NFs

(Co:Cu = 1/3) and  $\text{Co}_3\text{O}_4$ -CuO NFs (Co:Cu = 1/2) have been synthesized for comparison of photocatalytic performance.

**2.4. Characterization.** The specific surface area, pore size, and pore volume of mesoporous NFs were measured by  $\text{N}_2$  sorption method at liquid nitrogen temperature (-196 °C) using Quantachrome Instrument Autosorb (iQ<sub>2</sub>). The specific surface area and pore size distribution were estimated based on Brunauer-Emmett-Teller (BET) and Barrett-Joyner-Halenda (BJH) methods, respectively. The crystal structures of the nanofibers were characterized using a PANalytical X'Pert Pro multipurpose X-ray diffractometer (XRD) in the range of  $2\theta = 10$ –80° with Cu  $K\alpha$  (1.5418 Å) radiation. The surface morphology and elemental composition of mesoporous NFs were analyzed by field emission scanning electron microscopy (FE-SEM, Quanta 200 FEG, FEI). The morphology and mesoporosity of the nanofibers were examined by high resolution transmission electron microscope (HRTEM, Tecnai G2 F30, FEI). The scanning transmission electron microscopy-energy dispersive X-ray (STEM-EDX) analysis was used to study composition of the metal oxides within the nanofibers. The diameter of the NFs was calculated by ImageJ software. The UV-vis absorption spectra of the mesoporous NFs were recorded by UV-vis spectrophotometer (Varian Cary 100). The UV-visible-near-infrared diffuse reflectance spectra (UV-vis-NIR DRS) of the NFs were recorded by UV-vis-NIR spectrophotometer (Cary 5000). The photoluminescence spectra were measured on a fluorescence spectrometer (FL-1057 TCSPC) with an excitation wavelength at 390 nm light. Raman spectra measurement was conducted on a scanning near-field optical spectroscopy (SNOM) confocal Raman spectroscope (WITec) with a laser light wavelength of 532 nm. The Fourier transform infrared (FTIR) spectra of the samples were recorded with a Bruker VERTEX 70 in the range of 400–4000  $\text{cm}^{-1}$  at room temperature using KBr pellet.

**2.5. Visible Light Photocatalytic Degradation of Phenolic Compounds.** The visible light photocatalytic degradation of phenolic compounds was examined by taking phenol, 4-chlorophenol (4-CP), and 4-nitrophenol (4-NP) solution. A stock solution of 20  $\text{mg L}^{-1}$  for phenol, 4-CP, and 4-NP was prepared and suitably aqueous diluted to the required initial concentration. Prior to photocatalysis, the phenolic compounds solution was stirred for 20 min in order to reach equilibrium. Photocatalytic degradation of phenolic compounds (phenol, 4-CP, and 4-NP) was carried out by taking 10 mL of each solution (20  $\text{mg L}^{-1}$ ), 1  $\text{mg}\cdot\text{mL}^{-1}$  of catalyst dose at pH 6 for 70 min. The pH of the solution was monitored by Mettler Toledo pH meter with proper addition of 0.01 M  $\text{HNO}_3$  and/or  $\text{NH}_4\text{OH}$ . The phenolic compounds solution with catalyst was placed for visible light irradiation (300 W, Osram, sunlight simulation) at room temperature. The separated reaction solution from catalyst was analyzed UV-vis spectrophotometer (Varian Cary 100). The maximum absorbance of phenol, 4-CP, and 4-NP were at 210, 224, and 316 nm, respectively.

**2.6. Surface-Enhanced Raman Spectroscopy.** For surface-enhance Raman spectroscopy (SERS) measurements,<sup>29</sup> 500  $\mu\text{L}$  of Rh 6G water solution of 20  $\mu\text{M}$  was kept under ultrasound with 1 mg of all respective NFs. Then the Rh 6G-absorbed NFs were dispersed in 50  $\mu\text{L}$  of water. The suspension of 10  $\mu\text{L}$  was dropped onto a glass slide. The dropped solution spread evenly into a circle. After evaporation of the water, the sample was subjected to Raman measurement. All experiments were carried out at room temperature in the same conditions.

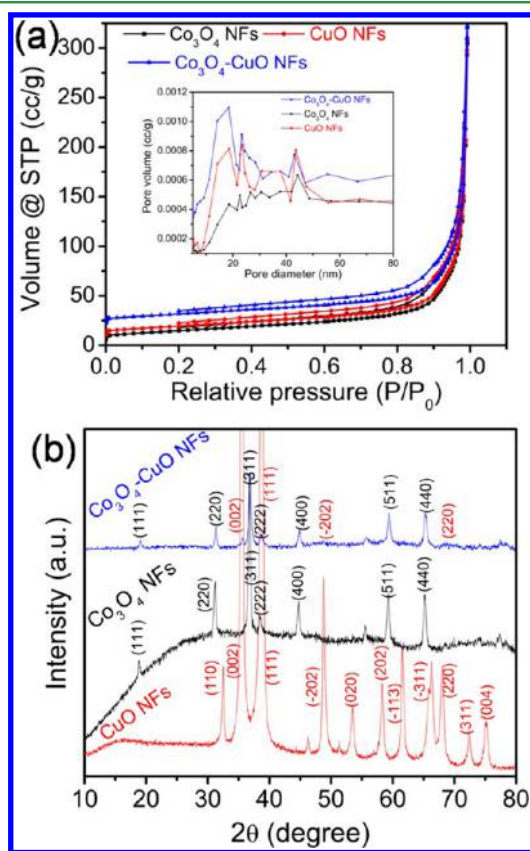
**2.7. Active Species Trapping Experiments.** The holes and radicals trapping experiments were conducted by adding various scavengers to the mesoporous composite  $\text{Co}_3\text{O}_4$ -CuO NFs in order to know the main reactive radical species involved in the photocatalytic process. Three different scavengers such as potassium iodide (KI; 0.1 mmol), isopropyl alcohol (IPA; 0.1 mmol), and *p*-benzoquinone (BQ; 0.1 mmol) have been used for the detection of holes ( $\text{h}^+$ ), hydroxyl radicals ( $\cdot\text{OH}$ ), and superoxide radical anions ( $\text{O}_2^{\cdot-}$ ) species in the photocatalytic phenolic compound degradation, respectively.<sup>30</sup> The trapping experiment is similar to that of the photocatalytic degradation of phenolic compounds. The scavengers are added to the phenolic compounds solutions prior to the photocatalyst.



**2.8. Photoelectrochemical Measurements.** Electrochemical measurements were conducted using a three-electrode electrochemical cell consisting of a platinum wire as the counter electrode, Ag/AgCl electrode as reference electrode, and a prepared electrode as the working electrode filled with an aqueous solution of 0.1 M  $\text{Na}_2\text{SO}_4$ . The electrolyte was saturated with nitrogen gas prior to electrochemical measurements. The photoelectrochemical measurements were performed using an Ivum multichannel potentiostat under illumination conditions ( $\lambda = 400$  nm). Irradiation was performed using a 300 W Xe lamp with a 400 nm cutoff filter.

### 3. RESULTS AND DISCUSSION

**3.1. Porosity of the Nanofibers.**  $\text{N}_2$  sorption isotherms and the pore diameter distributions of all the NFs are shown in Figure 1a, and a summary of the specific surface area, pore



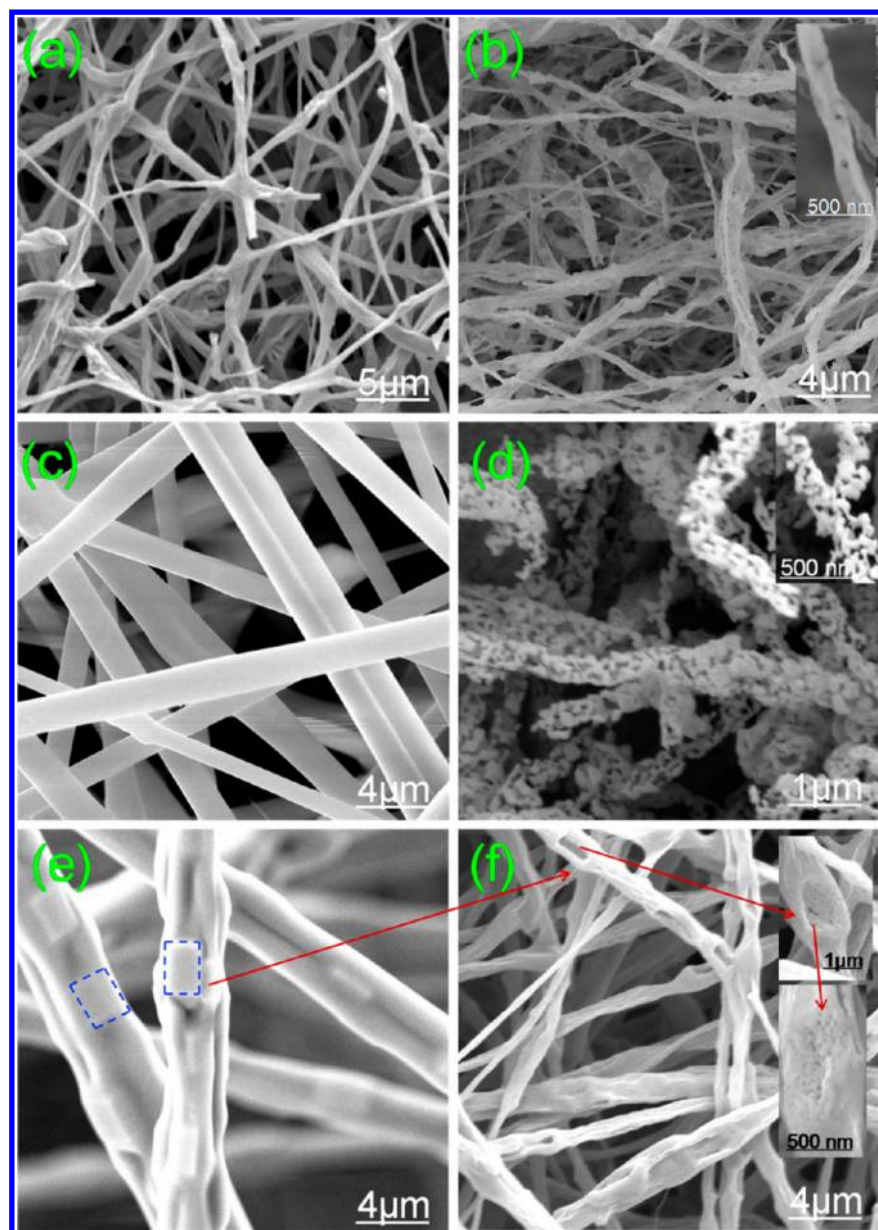
**Figure 1.** (a)  $\text{N}_2$  sorption isotherm and pore diameter (inset) and (b) wide-angle X-ray diffractograms of mesoporous  $\text{Co}_3\text{O}_4$ , CuO, and composite  $\text{Co}_3\text{O}_4$ -CuO NFs.

volume, and average pore diameter data is shown in Table S1 in Supporting Information. All the NFs are of typical IV isotherms with type H3 hysteresis loop according to the Brunauer–Deming–Deming–Teller (BDDT) classification, suggesting the presence of mesoporous structure.<sup>31</sup> The most important thing is that the materials showed intraparticle mesoporosity or “framework porosity”.<sup>32</sup> This is because the hysteresis loops of  $\text{Co}_3\text{O}_4$  NFs, CuO NFs, and composite  $\text{Co}_3\text{O}_4$ -CuO NFs start within a relative pressure ( $P/P_0$ ) region of 0.1–0.5.<sup>33</sup> The formation of mesoporous structure within NFs is due to the removal of mixed PVP and PEG polymeric matrix at optimum calcinations temperature (450 °C in 4 h). The pore size distribution curve of all NFs is shown in the Figure 1a (inset). The intense peak of all NFs is at 2–50 nm, the resulting mesoporous range. The BET specific surface area, pore

diameter, and pore volume of the NFs are listed in Table S1. The specific surface areas of  $\text{Co}_3\text{O}_4$  NFs, CuO NFs, and composite  $\text{Co}_3\text{O}_4$ -CuO NFs are 28, 35, and 39  $\text{m}^2/\text{g}$ , respectively. The high surface area of NFs will provide more reactive sites for an interaction with phenolic compounds. The pore diameter (Table S1) are in the range of the mesoporous. Hence from the BET isotherm and pore diameter data, it is concluded that all NFs have mesoporous character within framework.

The wide angle XRD patterns of mesoporous  $\text{Co}_3\text{O}_4$  NFs, CuO NFs, and composite  $\text{Co}_3\text{O}_4$ -CuO NFs are shown in the Figure 1b. In the XRD pattern of porous  $\text{Co}_3\text{O}_4$  NFs, the Bragg reflections at angles of 18.86°, 31.16°, 36.92°, 38.56°, 44.67°, 59.28°, and 65.26° correspond to (111) (220), (311), (222), (400), (511), and (440) main crystal planes indicating the formation of pure cubic crystalline  $\text{Co}_3\text{O}_4$  (JCDPS, card no. 42-1467). In the XRD pattern of mesoporous CuO NFs, the peaks at angles of 32.49°, 35.59°, 38.80°, 48.87°, 53.52°, 58.39°, 61.49°, 66.25°, 68.14°, 72.46°, and 75.23° were purely assigned to the (110), (002), (111), (−202), (020), (202), (−113), (−311), (220), (311), and (004) crystal planes, respectively. It can be seen that all the diffraction peaks belong to the CuO monoclinic phase (JCPDS 48-1548). The porous composite  $\text{Co}_3\text{O}_4$ -CuO NFs consisting of reflection planes (111), (220), (400), (222) (511), and (440) are due to cubic crystalline  $\text{Co}_3\text{O}_4$ , and crystal planes (002), (111), (−202), and (220) are due to monoclinic CuO. Hence, the presence of both  $\text{Co}_3\text{O}_4$  and CuO reflection planes in mesoporous composite  $\text{Co}_3\text{O}_4$ -CuO NFs indicates the formation of  $\text{Co}_3\text{O}_4$ -CuO composite.

**3.2. Morphological Analyses of Mesoporous Nanofibers.** The morphologies of the mesoporous nanofibers were investigated by SEM images. Figure 2 describes the morphology, and Table S1 depicts the diameter of different mesoporous NFs before and after calcination. The PVP/PEG is used as mixed polymeric carrier matrix for the electrospinning of nanofibers. The nanofibrous materials before calcinations (dried 70 °C for 6 h) are denoted as, for example, PVP/ $\text{Co}_3\text{O}_4$ /PEG, whereas after calcination (removal of PVP and PEG at 450 °C, 4 h) they are denoted as mesoporous  $\text{Co}_3\text{O}_4$  NFs. Similarly, the other NFs can be symbolized in order to describe the SEM morphology. The average diameter of mesoporous NFs is mentioned for all samples. The bead-free nanofibers with fiber diameter of  $280 \pm 50$  nm for PVP/ $\text{Co}_3\text{O}_4$ /PEG NFs are obtained before calcination (Figure 2a). The rough and mesoporous nanofibers with narrow fiber diameter of  $160 \pm 30$  nm for mesoporous  $\text{Co}_3\text{O}_4$  NFs are obtained (Figure 2b) after calcinations of PVP/ $\text{Co}_3\text{O}_4$ /PEG. Similarly, bead-free and smooth fibrous morphology having diameter of  $400 \pm 40$  nm is also observed for the sample of PVP/CuO/PEG in Figure 2c, whereas a rough, mesoporous nature and narrow diameter ( $130 \pm 20$  nm) for CuO NFs are obtained after calcination as depicted in Figure 2d. A clear mesoporous nature of CuO NFs framework is observed which is consistent with BET result. Moreover, Figure 2e shows the bead-free PVP/ $\text{Co}_3\text{O}_4$ -CuO/PEG NFs having diameter of  $600 \pm 70$  nm with irregular and inhomogeneous structure having rectangular moieties which is marked in arrows. These rectangular moieties disappeared, and creation of mesoporosity appeared within blank space of rectangular voids (Figure 2f) as well as the diameter reduction of the NFs ( $250 \pm 30$  nm) which is due to the combined effect and removal of PVP and PEG in the optimum calcinations temperature, which resulted in mesoporous  $\text{Co}_3\text{O}_4$ -CuO NFs. Furthermore, in order to examine the role of PEG for



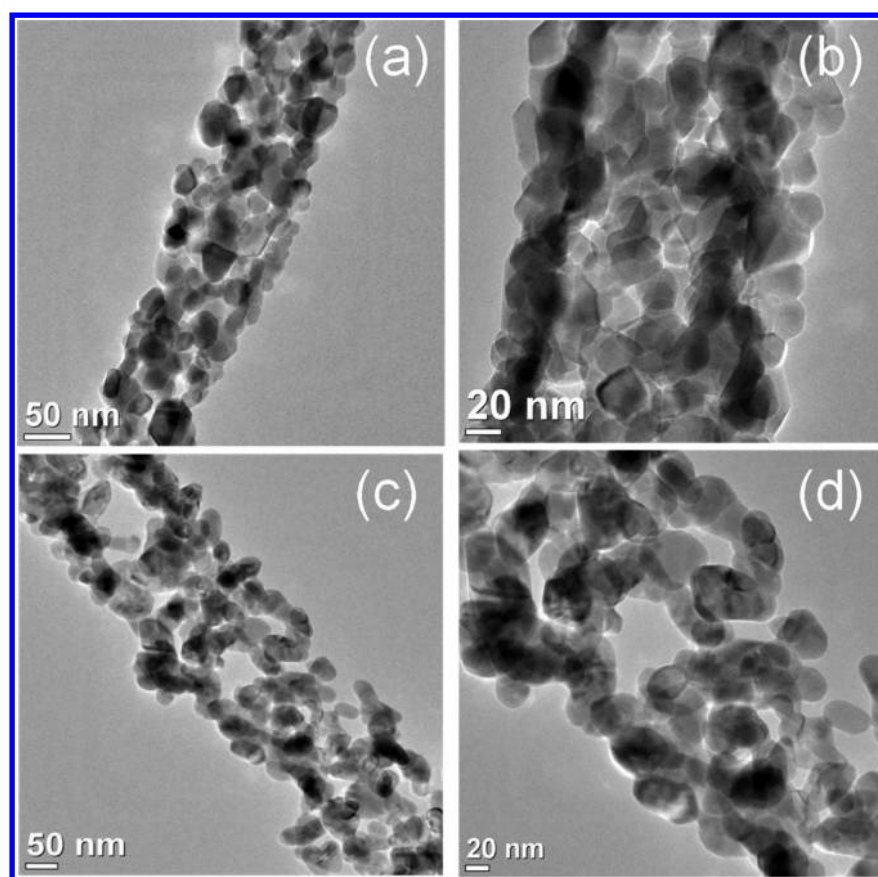
**Figure 2.** Representative SEM images of (a) PVP/Co<sub>3</sub>O<sub>4</sub>/PEG, (b) mesoporous Co<sub>3</sub>O<sub>4</sub> NFs, (c) PVP/CuO/PEG, (d) mesoporous CuO NFs, (e) PVP/Co<sub>3</sub>O<sub>4</sub>-CuO/PEG, and (f) mesoporous Co<sub>3</sub>O<sub>4</sub>-CuO NFs.

generating mesoporosity, SEM image of porous CuO nanoflower is incorporated in the Supporting Information (Figure S1). The CuO nanoflowers enriched with porous morphology have been obtained from the neat PEG matrix. From Scheme 1, Figure 2, and Figure S1, it is understood that PEG is responsible for generation of porosity whereas PVP aids to create NFs and mixed PVP/PEG produces mesoporous NFs. The decrease of fiber diameter for all NFs and creation of mesoporosity after calcination is due to the removal/decomposition of PVP matrix and quick decomposition of PEG, respectively. In short, a combination of high molecular weight polymeric matrix (i.e., PVP) helps the electrospinning of the nanofibers, and low molecular weight polymeric template (i.e., PEG) helps to create mesoporosity once the NFs are calcined in order to obtain inorganic NFs by removal of polymeric components.

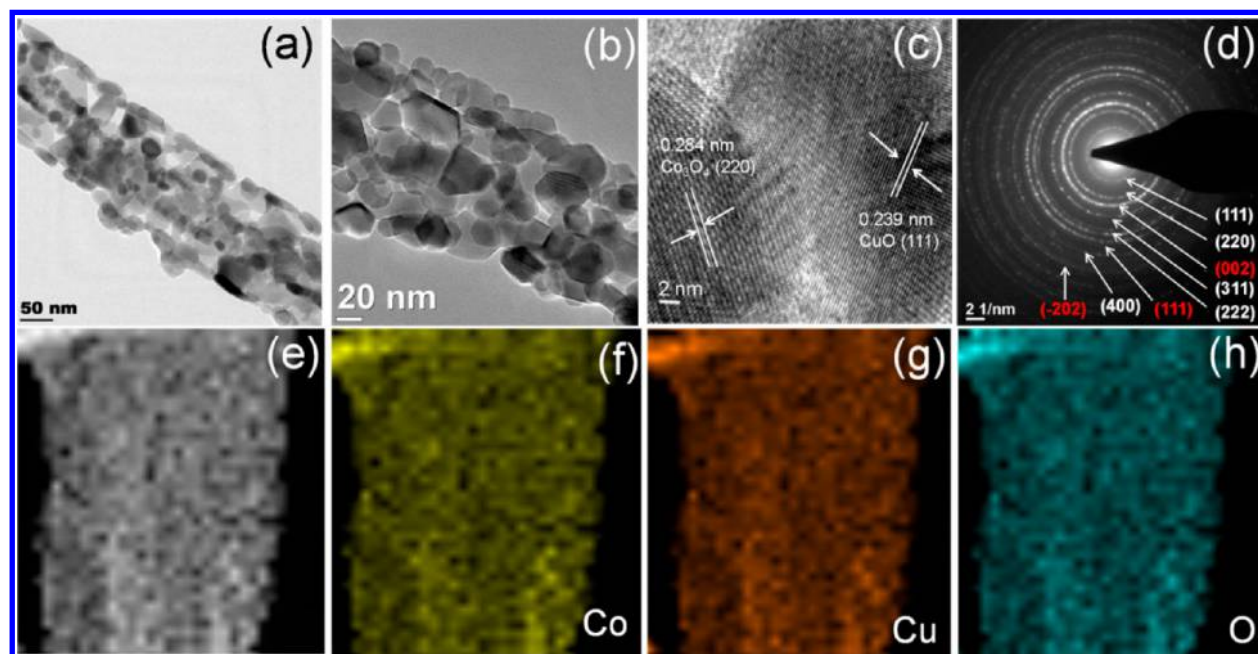
The TEM and HR-TEM analyses of the mesoporous NFs have been performed in order to study the morphology and

formation of nanoparticles (NPs) within NFs (Figure 3). Parts a and b of Figure 3 describe the TEM images of the mesoporous Co<sub>3</sub>O<sub>4</sub> NFs. The highly porous nature is clearly observed, i.e., mesoporous within the Co<sub>3</sub>O<sub>4</sub> NFs which was consistent with the BET analyses. The vital aspect of the present study is that Co<sub>3</sub>O<sub>4</sub> NPs are interconnected in such a way that also aids formation of an empty void (mesoporosity). The particle size of the Co<sub>3</sub>O<sub>4</sub> NPs is  $16.5 \pm 4$  nm (Table S1). The formation of the mesoporous and NPs within NFs can have an architecture as “mesoporous–NPs–Co<sub>3</sub>O<sub>4</sub> NFs”. The similar micrographs were observed in the case of the mesoporous CuO NFs (Figure 3c,d) and also followed the similar architecture, i.e., “mesoporous–NPs–CuO NFs”. The size of the interconnected CuO NPs is  $20.0 \pm 2.0$  within CuO NFs. Hence, mesoporosity and interconnected NPs within both mesoporous CO<sub>3</sub>O<sub>4</sub> and CuO NFs will enhance the surface reactive site and visible light absorption property through quantum confinement phenomena, respectively. These phe-





**Figure 3.** Representative TEM micrographs of mesoporous  $\text{Co}_3\text{O}_4$  NFs (a, b) and mesoporous CuO NFs (c, d).



**Figure 4.** Representative micrographs of mesoporous composite  $\text{Co}_3\text{O}_4$ -CuO NFs: (a, b) TEM images; (c) HR-TEM image (lattice fringes); (d) SAED pattern; (e-h) EDAX mapping patterns.

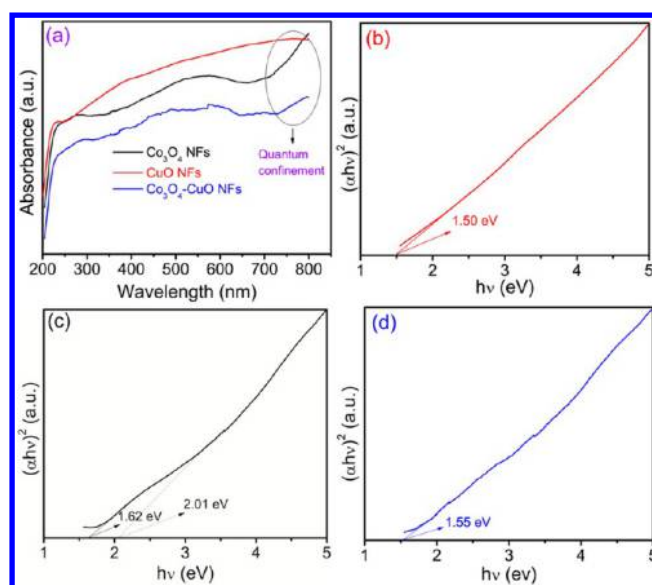
nomena will be helpful for an efficient photocatalytic degradation of phenolic compounds in the visible light. The lattice fringes from HR-TEM and selected area electron diffraction (SAED) pattern of the  $\text{Co}_3\text{O}_4$  NFs are displayed in the Figure S2a,b. The lattice spacing 0.286 nm is due the

presence of  $\text{Co}_3\text{O}_4$  with the plane (220) within the mesoporous  $\text{Co}_3\text{O}_4$  NFs.<sup>34</sup> The SAED pattern exhibits the diffraction rings (111), (220), (311), and (222) of a spinel structure of  $\text{Co}_3\text{O}_4$  in mesoporous  $\text{Co}_3\text{O}_4$  NFs. The inset image in Figure S2b shows the fast Fourier transform (FFT) diffractogram from the

mesoporous  $\text{Co}_3\text{O}_4$  NFs with  $\{110\}$  direction.<sup>21</sup> The exposed surface active facets  $\{110\}$  are more reactive in the mesoporous  $\text{Co}_3\text{O}_4$  NFs, which could absorb visible light efficiently. Figure S2c shows the HR-TEM image along with lattice spacing, and Figure S2d shows the SAED pattern of mesoporous CuO NFs. The lattice spacing 0.240 nm for (111) plane is due to the CuO monoclinic phase.<sup>35</sup> The SAED pattern exhibits the diffraction rings (110), (002), and (111), corresponding to the presence of CuO in the mesoporous CuO NFs. The facets  $\{001\}$  are observed from FFT diffractogram Figure S2d (inset), which is the highly active facets of the CuO.<sup>28</sup> The reactive facets  $\{110\}$  from  $\text{Co}_3\text{O}_4$  and  $\{001\}$  from CuO will enhance the photocatalytic properties through light absorption phenomena.

The TEM and HR-TEM images of the mesoporous composite  $\text{Co}_3\text{O}_4$ -CuO NFs are shown in Figure 4, and particle size within NFs is listed in the Table S1. Figure 4a,b shows that the porous nature, i.e., mesoporous, is confirmed from BET analyses. The TEM imaging (Figure 4a,b) also indicated that the enriched  $\text{Co}_3\text{O}_4$  and CuO NPs are interconnected with each other within composite  $\text{Co}_3\text{O}_4$ -CuO NFs. The particle size of NPs within composite  $\text{Co}_3\text{O}_4$ -CuO NFs is measured as  $15.2 \pm 3.0$  nm (Table S1). Hence, the architecture of “mesoporous-NPs- $\text{Co}_3\text{O}_4$ -CuO NFs” is formed which is similar to those of the other two NFs ( $\text{Co}_3\text{O}_4$  NFs and CuO NFs). From the HR-TEM imaging (Figure 4c), the lattice spacing of 0.284 nm for  $\text{Co}_3\text{O}_4$  (220) and 0.239 nm for CuO (111) was measured within the mesoporous composite  $\text{Co}_3\text{O}_4$ -CuO NFs.<sup>34,35</sup> The decrease of the lattice spacing in  $\text{Co}_3\text{O}_4$  and CuO within composite as compared to the neat  $\text{Co}_3\text{O}_4$  and CuO NFs is due to the formation of the composite  $\text{Co}_3\text{O}_4$ -CuO NFs. The lattice fringes showing the polycrystalline nature are due to the presence of  $\text{Co}_3\text{O}_4$  and CuO. The SAED pattern (Figure 4d) exhibits the diffraction rings (111), (220), (311), (222), and (400) corresponding to the crystalline nature of the spinel structure of  $\text{Co}_3\text{O}_4$ , and the plane corresponding to (002), (111), and (-202) is due to the monoclinic structure of CuO (marked in red) within mesoporous composite  $\text{Co}_3\text{O}_4$ -CuO NFs. The polycrystalline nature of mesoporous  $\text{Co}_3\text{O}_4$ -CuO NFs is also helpful for enhancement of the light absorption property. In order to confirm the distribution of elements and reactive sites on the surface of mesoporous composite  $\text{Co}_3\text{O}_4$ -CuO NFs, EDAX elemental mapping analysis is employed (Figure 4e-h). The EDAX mapping results revealed the high content of Co as compared to Cu in  $\text{Co}_3\text{O}_4$ -CuO NFs, which is consistent with the experimental condition. The EDAX elemental analyses of mesoporous  $\text{Co}_3\text{O}_4$  NFs, CuO NFs, and composite  $\text{Co}_3\text{O}_4$ -CuO NFs reveal the presence of respective elements in each sample of NFs (Figure S3).

**3.3. Optical Analyses of Mesoporous Nanofibers.** The optical properties such as absorption edges, effect of composite on the electronic structure, and band gap energy of the mesoporous  $\text{Co}_3\text{O}_4$  NFs, CuO NFs, and composite  $\text{Co}_3\text{O}_4$ -CuO NFs are studied by using UV-visible absorption spectra (Figure 5a). The entire mesoporous NFs showed strong absorption band at 200–300 nm, which is due to the ligand to metal charge transfer (LMCT) between oxygen and metal center within the NFs. The mesoporous CuO NFs exhibit continuous and enhanced light absorption in the range of 400–800 nm. This is due to the efficient harvesting of visible light. Moreover, the octahedrally coordinated  $\text{Cu}^{2+}$  species is observed at 600–800 nm which is due to the d-d transition band.<sup>36</sup> A broad peak starting from 400–670 nm is due to the



**Figure 5.** (a) UV-visible absorption spectra of mesoporous NFs samples, showing plots of  $(\alpha h\nu)^2$  vs photon energy ( $h\nu$ ) for the band gap energy of (b) mesoporous CuO NFs, (c) mesoporous  $\text{Co}_3\text{O}_4$  NFs, and (d) mesoporous composite  $\text{Co}_3\text{O}_4$ -CuO NFs.

d-d transitions of  $\text{Co}^{3+}$  in octahedral sites and  $\text{Co}^{2+}$  in tetrahedral sites in mesoporous  $\text{Co}_3\text{O}_4$  NFs.<sup>37</sup> The d-d transition of octahedral and tetrahedral sites helps light absorption in the visible region. The mesoporous composite  $\text{Co}_3\text{O}_4$ -CuO NFs shows the dual behavior of d-d transitions of  $\text{Co}^{3+}$  and  $\text{Co}^{2+}$  along with  $\text{Cu}^{2+}$  charge transfer in the peak region 400–712 nm. The dominant nature  $\text{Co}_3\text{O}_4$  absorption peak in composite  $\text{Co}_3\text{O}_4$ -CuO NFs is due to the composition of high amount of  $\text{Co}_3\text{O}_4$  with less amount of CuO. The most interesting finding of the present investigation is the rising or red shifting of peaks from 720 to 800 nm in all mesoporous NFs. This phenomenon may be due to the creation of quantum confinement phenomena in mesoporous  $\text{Co}_3\text{O}_4$  NFs, CuO NFs, and composite  $\text{Co}_3\text{O}_4$ -CuO NFs. All presently fabricated NFs showing red shifts which favor high photocatalytic applications in the visible region. The band gap energy of all NFs materials can be calculated by using the following equation.<sup>38</sup>

$$\alpha h\nu = A(h\nu - E_g)^n$$

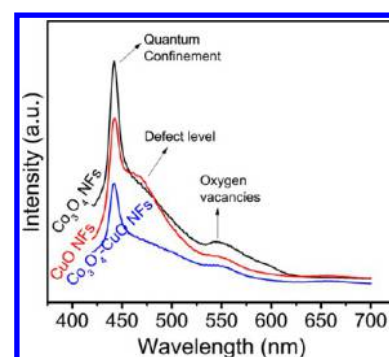
where  $\alpha$ ,  $\nu$ ,  $A$ , and  $E_g$  are the absorption coefficient, light frequency, proportionality constant, and band gap, respectively. The band transition depends upon the value of  $n = 1/2$  for direct transition and  $n = 2$  for indirect transition. It is examined that all the mesoporous NFs show direct allowed transitions. The band gap energies of the all NFs can be estimated from the plots of  $(\alpha h\nu)^2$  versus photon energy ( $h\nu$ ). The intercept of the tangent to the X axis would give a good approximation of the band gap energies for the NFs samples, as shown in Figure 5b,c,d. The band gap of mesoporous CuO NFs (Figure 5b) is found to be 1.50 eV, which belongs to the visible light response semiconductor. The observed band gap value is wider than the bulk CuO (1.2 eV), which is due to the quantum confinement effect for mesoporous CuO NFs. The wider band gap of mesoporous CuO NFs as compared to bulk is due to efficient light absorption by mesoporous NPs and crystalline CuO NFs, confirmed from TEM and HR-TEM analyses. It is well-understood that the reduction in particle size results in



increased surface/volume ratio.<sup>39</sup> The increase of band gap energy in NFs as compared to bulk material is due to the lowering of the coordination number and atomic interaction of surface atom which leads to increases of the highest valence band energy and decreases of the lowest unoccupied conduction band energy.<sup>40</sup> The mesoporous  $\text{Co}_3\text{O}_4$  NFs (Figure 5c) has two  $E_g$  with values of 1.62 ( $E_{g1}$ ) and 2.01 eV ( $E_{g2}$ ), which agreed with the band structure of  $\text{Co}_3\text{O}_4$  with  $\text{Co}^{3+} t_{2g} \rightarrow \text{Co}^{2+} t_{2g}$  ( $E_{g1}$ ) and  $\text{O}^{2-} \rightarrow \text{Co}^{2+} t_{2g}$  ( $E_{g2}$ ) charge-transfer transition, respectively.<sup>41</sup> It is evaluated that the band gap values for bulk  $\text{Co}_3\text{O}_4$  are 1.48 eV ( $E_{g1}$ ) and 2.19 eV ( $E_{g2}$ ).<sup>13</sup> The  $E_{g1}$  value  $\text{Co}_3\text{O}_4$  NFs is 1.62 eV which is wider than the  $E_{g1}$  value of bulk  $\text{Co}_3\text{O}_4$  (1.48 eV). This is due to the quantum confinement effect and lattice contraction. Furthermore, the  $E_{g2}$  value of  $\text{Co}_3\text{O}_4$  NFs is 2.01 eV which is smaller than the  $E_{g2}$  value of bulk  $\text{Co}_3\text{O}_4$  (2.19 eV). This can be attributed to several factors such as lattice expansion and/or the presence of defects (mainly vacancies) in the intergranular regions which forms the donor and acceptor bands.<sup>42</sup> The mesoporous composite  $\text{Co}_3\text{O}_4$ -CuO NFs (Figure 5d) has band gap of 1.55 eV. The narrow band gap as compared to  $\text{Co}_3\text{O}_4$  NFs is due to the formation of localized state by intermixing of Co 2p and Cu 2p. The most interesting finding is that the band gap of  $\text{Co}_3\text{O}_4$ -CuO NFs is wider than the bulk CuO and  $\text{Co}_3\text{O}_4$ , which proves the availability of quantum confinement effect and defect/vacancies sites. Hence, the optical properties of the fabricated mesoporous NFs have quantum confinement, defect/vacancies, and suitable band gap which could enhance the light absorption property in the visible region for an efficient photocatalytic degradation of organic pollutants.

In order to prove the quantum confinement phenomena more clearly within mesoporous NFs, the UV-vis-NRI DRS studies have been performed and the obtained spectra were given in the Supporting Information (Figure S4). In the UV-visible absorption spectra (Figure 5a), the optical behavior of the mesoporous NFs from 200 to 720 nm is already discussed. Hence, in the UV-vis-NRI DRS, the discussion regarding optical properties of the fabricated NFs was focused from 720 to 2000 nm. The interesting and rare finding observed is that all mesoporous NFs have the quantum confinement effect. The high red shifting peaks starting from nearly 780 to 1000 nm is due to the quantum confinement effect for all NFs. The d-d transition band for tetrahedrally coordinated  $\text{Cu}^{2+}$  species appears around 1300–1600 nm for CuO NFs.<sup>43</sup> In both  $\text{Co}_3\text{O}_4$  NFs and  $\text{Co}_3\text{O}_4$ -CuO NFs, the shoulder peak starting from 1020–1210 nm is due to the “intervalence” charge-transfer  $\text{Co}^{2+} \rightarrow \text{Co}^{3+}$  and the peak at 1700 nm is attributed to the  ${}^4\text{A}_2(\text{F}) \rightarrow {}^4\text{T}_1(\text{F})$  transition in the  $\text{Co}_3\text{O}_4$  structure. The charge transfer transition represents an internal oxidation-reduction process which facilitates photocatalytic process in the visible region. Hence, from UV-visible absorption and UV-vis-NRI DRS spectra it is concluded that high red shifting of all mesoporous NFs is due to the quantum confinement effect and the internal charge transfer within  $\text{Co}^{2+}$  and  $\text{Co}^{3+}$  and  $\text{Cu}^{2+}$ .

The photoluminescence (PL) emission spectra describe the effectiveness of the charge carrier trapping, immigration, and transfer behaviors of the photoexcited electron-hole pairs in semiconductors. The PL emission ( $\lambda_{\text{exc}} = 390$  nm) spectra of semiconductor mesoporous  $\text{Co}_3\text{O}_4$  NFs, CuO NFs, and composite  $\text{Co}_3\text{O}_4$ -CuO NFs are shown in Figure 6. For all mesoporous NFs the peaks centered at 442, 470, and 550 nm correspond to the blue, green, and yellow-orange bands, respectively. The blue-shift behavior of the intense peak



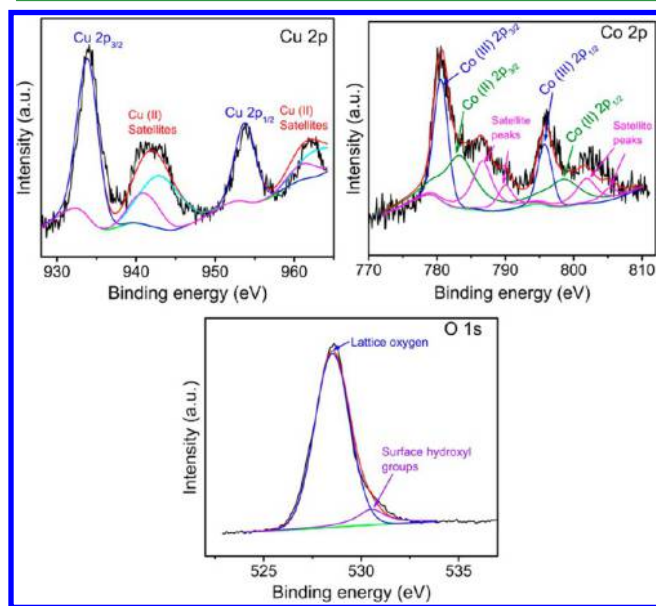
**Figure 6.** Photoluminescence spectra of mesoporous  $\text{Co}_3\text{O}_4$ , CuO, and composite  $\text{Co}_3\text{O}_4$ -CuO NFs.

position at 442 nm for all NFs is due to the Burstein–Moss effect resulting from nanostructure materials, which is provided with the enhanced quantum confinement effect.<sup>44</sup> The intense peak for all NFs at blue region is due to the electron excitation, and the weak green emission (470 nm) is due to the deep level defect.<sup>45</sup> Moreover, the yellow-orange shift (550 nm) is due to the oxygen vacancies ( $V_o$ ). For mesoporous  $\text{Co}_3\text{O}_4$  NFs, the emission peaks 442, 470, and 550 nm correspond to charge transfer  $\text{O}^{2-} \rightarrow \text{Co}^{2+}$  process,  $\text{O}^{2-} \rightarrow \text{Co}^{3+}$  charge transfer, and oxygen vacancies ( $V_o$ ), respectively.<sup>46</sup> Likewise, mesoporous CuO NFs exhibit the copper vacancy ( $V_{\text{Cu}}$ ), Cu interstitial ( $I_{\text{Cu}}$ ), oxygen vacancy ( $V_o$ ) corresponding to the emission peaks at 442, 470, and 550 nm, respectively.<sup>47</sup> The mesoporous composite  $\text{Co}_3\text{O}_4$ -CuO NFs also show the blue and yellow-orange emission peaks, resulting quantum confinement effect, and oxygen vacancies. From the above investigation, it is well-understood that the all mesoporous NFs exhibit quantum confinement effect (blue region), crystal defect (green region), and oxygen vacancies (yellow-orange region). Quantum confined effect provides high light absorption property of nanofibers due presence of NPs within all NFs, which provides high surface to volume ratio. According to the Marcus theory, quantum confined effect favors higher rates of interfacial charge transfer due to the increased thermodynamic energy of the confined electron-hole pairs.<sup>48</sup> Hence, formation of interfacial charge transfers in all NFs increases the generation of electron-hole for photocatalytic application in visible region. The defect level indicates the radiative recombination resulted in increase in the PL intensity.<sup>49</sup> The green emission peak (470 nm) completely vanished in the case of mesoporous composite  $\text{Co}_3\text{O}_4$ -CuO NFs as compared to  $\text{Co}_3\text{O}_4$  and CuO NFs, indicating the prevention of electron-hole recombination. This phenomenon proves the role of CuO as cocatalyst in  $\text{Co}_3\text{O}_4$ . Moreover, the oxygen vacancies of all NFs are situated at 550 nm. Oxygen vacancies aid to trap the photoemission electron from conduction band of fabricated NFs via nonradiative transition. This phenomenon prevents the electron-hole recombination and resulting swift photocatalytic degradation of phenolic compounds in visible light. Furthermore, the Schottky barrier appears to be due to the mixing of the CuO into the  $\text{Co}_3\text{O}_4$ . This phenomenon also helps to create superficial space charge layer between valence and conduction band of  $\text{Co}_3\text{O}_4$  NFs, which could act as an electron sink to efficiently prevent the electron-hole recombination.<sup>50</sup> That is why PL intensity gradually decreases from  $\text{Co}_3\text{O}_4$  NFs > CuO NFs >  $\text{Co}_3\text{O}_4$ -CuO NFs.

**3.4. Chemical Structure Analyses of Nanofibers.** X-ray photoelectron spectroscopy (XPS) analysis has been performed



to investigate the electronic environment and oxidation state of mesoporous composite  $\text{Co}_3\text{O}_4$ -CuO NFs (Figure 7). It has



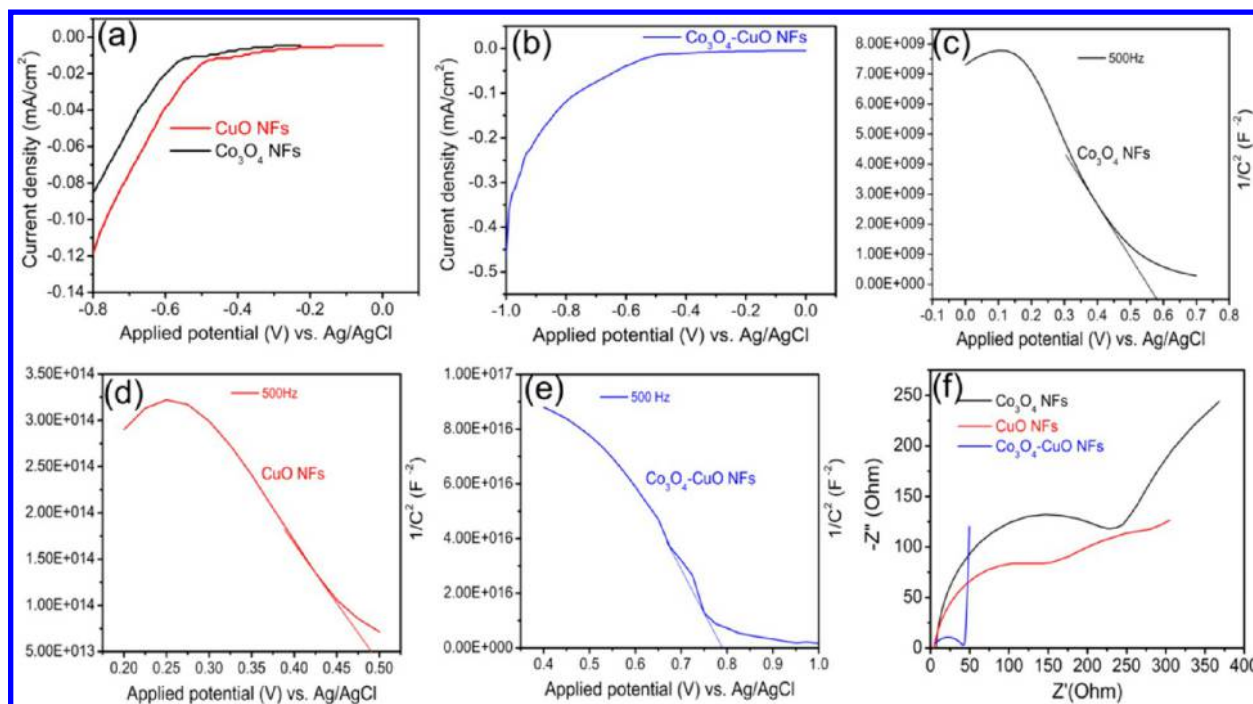
**Figure 7.** Cu 2p core level spectrum, Co 2p spectrum, and O 1s spectrum in the mesoporous composite  $\text{Co}_3\text{O}_4$ -CuO NFs.

been investigated that the main peaks are at  $\sim 933.4$  eV (Cu  $2p_{3/2}$ ) and  $953.2$  eV (Cu  $2p_{1/2}$ ), along with the presence of their characteristic shakeup satellite peaks at approximately  $941.3$  and  $961.5$  eV, respectively, which are due to the existence of pure CuO.<sup>51</sup> In the present study, the binding energy (BE) of Cu  $2p_{3/2}$  and Cu  $2p_{1/2}$  is at approximately  $933.77$  and  $953.71$  eV, indicating fully oxidized CuO<sup>52</sup> in mesoporous composite  $\text{Co}_3\text{O}_4$ -CuO NFs. The shakeup satellite peaks located at approximately  $940.75$ ,  $942.80$ , and  $961.37$ ,  $962.41$  eV are attributed to the Cu (3d) hole states.<sup>53</sup> These satellite peaks also confirm the Cu(II) bonding state of CuO in the composite  $\text{Co}_3\text{O}_4$ -CuO NFs.<sup>54</sup> The XPS spectrum of Co 2p in Figure 7 shows that the peaks at approximately  $780.52$ , and  $795.70$  eV correspond to Co(III)  $2p_{3/2}$  and  $2p_{1/2}$ , while  $783.4$  and  $798.6$  eV correspond to Co(II)  $2p_{3/2}$  and  $2p_{1/2}$ , respectively. The weak  $2p_{3/2}$  satellite peaks are found at approximately  $786.6$  and  $790.01$  eV, and those of  $2p_{1/2}$  satellite peaks are found at approximately  $802.00$  and  $805.30$  eV.<sup>55</sup> These peaks represent the existence of Co(II) in the tetrahedral sites and Co(III) in the octahedral sites of  $\text{Co}_3\text{O}_4$  in the mesoporous composite  $\text{Co}_3\text{O}_4$ -CuO NFs. The observed energy separation between Co(III)  $2p_{3/2}$  and Co(III)  $2p_{1/2}$  peaks is  $15 \pm 0.2$  eV, corresponding to the  $\text{Co}_3\text{O}_4$ . The O 1s BE in composite  $\text{Co}_3\text{O}_4$ -CuO NFs is observed at  $\sim 528.54$  eV and is due to the strong interaction of O with Co and Cu in mesoporous composite  $\text{Co}_3\text{O}_4$ -CuO NFs, whereas peak at  $\sim 530.63$  eV (lattice oxygen) is due to the surface hydroxyl present in the mesoporous composite  $\text{Co}_3\text{O}_4$ -CuO NFs.<sup>56</sup> From the above results it is concluded that the higher of BE of Co 2p as compared to pure Co 2p (Rakibuddin et al.)<sup>57</sup> and lower BE of  $528.54$  eV as compared to  $529$  eV pure  $\text{Co}_3\text{O}_4$ <sup>57</sup> are due to the electron transfer from Co(II)/Co(III) to Cu(II) through O atom and forming the -Co-O-Cu- linkage in the mesoporous composite  $\text{Co}_3\text{O}_4$ -CuO NFs.

The FTIR spectra of  $\text{Co}_3\text{O}_4$  NFs, CuO NFs, and  $\text{Co}_3\text{O}_4$ -CuO NFs are displayed in the Figure S5. The FTIR spectrum

of the mesoporous  $\text{Co}_3\text{O}_4$  NFs exhibits two distinct bands at  $567$  and  $663$   $\text{cm}^{-1}$  and are due to the stretching vibrations of the metal-oxygen bonds.<sup>58</sup> The first band at  $567$   $\text{cm}^{-1}$  is assigned to the  $\text{OB}_3$  vibration, where B denotes  $\text{Co}^{3+}$  in the octahedral hole. The band at  $663$   $\text{cm}^{-1}$  is assigned to the  $\text{ABO}_3$  vibration, where A denotes  $\text{Co}^{2+}$  in the tetrahedral hole. Hence, nature of the mesoporous  $\text{Co}_3\text{O}_4$  NFs is spinel lattice. The Cu-O stretching vibration observed at  $535$   $\text{cm}^{-1}$  is due to the formation of CuO in mesoporous CuO NFs.<sup>59</sup> The stretching frequencies  $553$  and  $661$   $\text{cm}^{-1}$  are due to the presence of  $\text{Co}^{3+}/\text{Cu}^{2+}$  and  $\text{Co}^{2+}$  in the mesoporous composite  $\text{Co}_3\text{O}_4$ -CuO NFs, respectively. That means the formation of spinel  $\text{Co}_3\text{O}_4$  and CuO is observed within composite  $\text{Co}_3\text{O}_4$ -CuO NFs.

The SERS experiment was performed by incident of laser ( $532$  nm) to the bare mesoporous  $\text{Co}_3\text{O}_4$  NFs, CuO NFs, and composite  $\text{Co}_3\text{O}_4$ -CuO NFs in comparison with addition of probe molecule rhodamine 6G (Rh 6G) mixed with mesoporous NFs separately (Figure S6). From Figure S6, it is investigated that all the mesoporous NFs mixed with Rh 6G have high enhancement of Raman intensity as compared to neat mesoporous NFs. The enhancement of Raman intensity is due to the presence of NPs within the mesoporous NFs. The NPs present in all the mesoporous NFs efficiently adsorb Rh 6G because of the quantum confinement phenomena, which leads to the absorption of the laser radiation and enhancement of the Raman intensity of Rh 6G. This enhancement of Raman intensity in the presence of NPs within mesoporous NF is called as SERS enhancement. The SERS signals due to the addition of Rh 6G are  $1646$   $\text{cm}^{-1}$ ,  $1567$   $\text{cm}^{-1}$ ,  $1495$   $\text{cm}^{-1}$ ,  $1362$   $\text{cm}^{-1}$ ,  $1309$   $\text{cm}^{-1}$ , and  $1184$   $\text{cm}^{-1}$ . These Raman signals arose from the totally symmetric modes of in-plane C-C stretching vibrations of Rh 6G.<sup>60</sup> The Raman band at  $772$   $\text{cm}^{-1}$  was assigned to the C-H out-of plane bending vibration of Rh 6G. The shift at  $608$   $\text{cm}^{-1}$  was assigned to the in-plane bending vibration of the Rh 6G C-C-C ring.<sup>61</sup> The SERS enhancement is based on the two types of mechanisms: the first is the electromagnetic enhancement, which is caused by a strong surface plasmon resonance of the nanoparticles metal surface coupled to the incident light.<sup>62</sup> The second is the chemical enhancement, which can be considered to be a resonance Raman process between the ground electronic state of the molecule-metal complex and its new excited levels arising from charge transfer between the metallic surface and the adsorbed molecule.<sup>63</sup> It should be noted that semiconductor materials have dominant contribution for the SERS signal, which must involve a charge-transfer mechanism since surface plasmon resonances lie far in the infrared.<sup>63</sup> In the present context, interconnected NPs situated in the mesoporous  $\text{Co}_3\text{O}_4$  NFs, CuO NFs, and composite  $\text{Co}_3\text{O}_4$ -CuO NFs belong to semiconductor, interacting with molecule (Rh 6G) through charge transfer mechanism. It has been noted that the quantum confinement phenomenon arises in the small size of the semiconductor NPs. The surface resonance happens only in the small size NPs which leads to excite the laser energy. Hence, the NPs size of the mesoporous  $\text{Co}_3\text{O}_4$  NFs, CuO NFs, and composite  $\text{Co}_3\text{O}_4$ -CuO NFs is  $16.5 \pm 4$ ,  $20.0 \pm 2$ , and  $15.2 \pm 3$ , respectively. The present interconnected NPs mediated mesoporous NFs have internal charge transfer phenomena and semiconductor behavior, proved from UV-vis absorption and UV-vis DRS spectra. These NPs adsorb Rh 6G molecule efficiently which provides the good platform for the excitation of the laser through charge transfer mechanism. The charge transfer process enhances the SERS signals. Conclusively,



**Figure 8.** Potential–current density curves of mesoporous  $\text{Co}_3\text{O}_4$  NFs, CuO NFs (a, b) and composite  $\text{Co}_3\text{O}_4$ –CuO NFs (c) under light condition. Mott–Schottky plots of mesoporous  $\text{Co}_3\text{O}_4$  NFs, CuO NFs, and composite  $\text{Co}_3\text{O}_4$ –CuO NFs (c–e) under dark. Nyquist plot for the  $\text{Co}_3\text{O}_4$  NFs, CuO NFs, and composite  $\text{Co}_3\text{O}_4$ –CuO NFs (f).

enhancement of SERS signals proves the quantum confinement effect of the interconnected NPs within all mesoporous NFs.

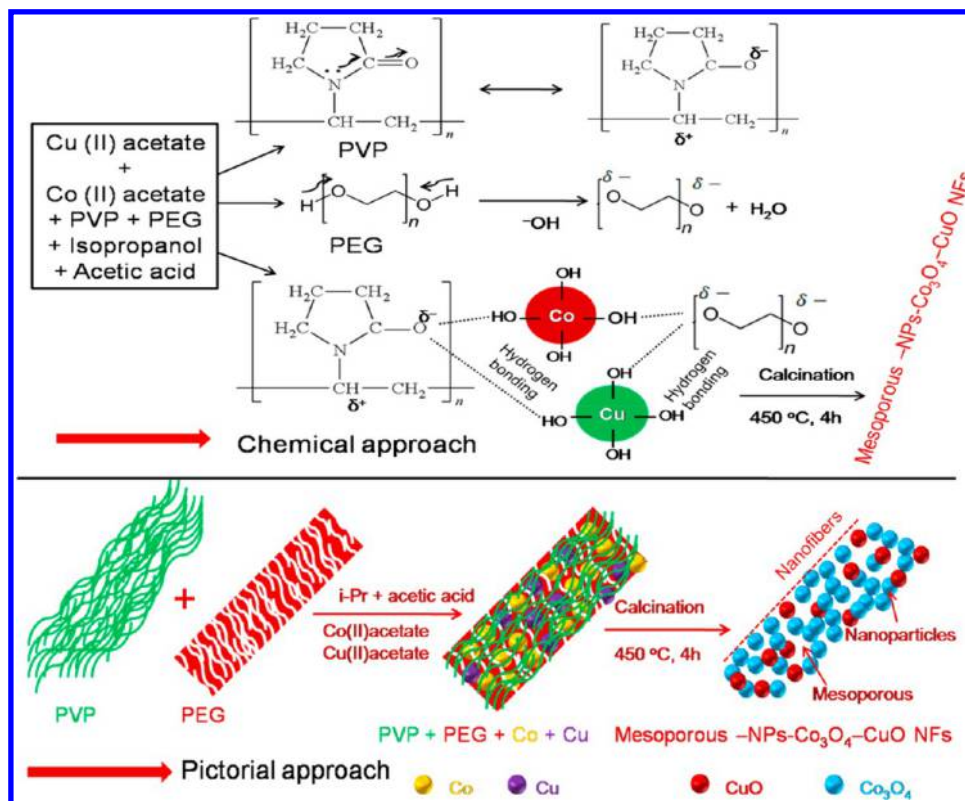
**3.5. Electrochemical Studies of Nanofibers.** In order to check the origin of the high photoactivity of the mesoporous composite  $\text{Co}_3\text{O}_4$ –CuO NFs along with neat mesoporous  $\text{Co}_3\text{O}_4$  NFs and CuO NFs, photocurrent measurement, Mott–Schottky (MS) plots and electrochemical impedance spectroscopy (EIS) were employed to investigate the current generation, the majority carrier density, and charge transfer rate at the electrode/electrolyte interface inside mesoporous NFs. The photocurrent under light illumination of mesoporous  $\text{Co}_3\text{O}_4$  NFs, CuO NFs, and composite  $\text{Co}_3\text{O}_4$ –CuO NFs was measured in 0.1 M  $\text{Na}_2\text{SO}_4$  at a scan rate of 10  $\text{mV s}^{-1}$  under light irradiation ( $\lambda \geq 400$  nm). Parts a and b of Figure 8 shows the photocurrent densities of the mesoporous  $\text{Co}_3\text{O}_4$  NFs, CuO NFs, and composite  $\text{Co}_3\text{O}_4$ –CuO NFs are measured to be  $-0.085$   $\text{mA/cm}^2$ ,  $-0.012$   $\text{mA/cm}^2$ , and  $-0.44$   $\text{mA/cm}^2$ , respectively. The negative values of the photocurrent densities of all NFs represent the cathodic current and are due to the p-type nature of the semiconductor. The generation of the photocurrent in all NFs photocathodes is due to the mesoporosity and NPs within NFs morphology, whereas high generation of photocurrent density in  $\text{Co}_3\text{O}_4$ –CuO NFs photocathode as compared to other NFs is due to the charge transport phenomena.<sup>64</sup> The charge transport phenomena of  $\text{Co}_3\text{O}_4$  could be improved by the CuO in composite  $\text{Co}_3\text{O}_4$ –CuO NFs. This charge transport phenomena prove the role of CuO as cocatalyst and enhancement of the photocatalytic performance.

The Mott–Schottky method helps to determine the band structure and recombination behavior, which are studied under dark conditions in three-electrode systems using Ag/AgCl as the reference electrode and platinum as the counter electrode dipped in 0.1 M  $\text{Na}_2\text{SO}_4$  at pH 6.5 in 500 Hz. The Mott–Schottky plots of mesoporous  $\text{Co}_3\text{O}_4$  NFs, CuO NFs, and

composite  $\text{Co}_3\text{O}_4$ –CuO NFs are represented in Figure 8c–e. The observed negative slope for all three materials indicates that all are in p-type semiconductor. The single slope for the composite  $\text{Co}_3\text{O}_4$ –CuO NFs indicates that semiconductor  $\text{Co}_3\text{O}_4$  is vitally involved in the photocatalytic performance with CuO as cocatalyst. Furthermore, from the intercept of the Mott–Schottky plots (Figure 8c–e), the flat band potentials ( $E_{fb}$ ) for mesoporous  $\text{Co}_3\text{O}_4$  NFs, CuO NFs, and composite  $\text{Co}_3\text{O}_4$ –CuO NFs are estimated to be 0.57, 0.48, and 0.78 V vs Ag/AgCl, respectively. As is well-known, a higher  $E_{fb}$  value for a p-type semiconductor implies a higher degree of band bending and a larger space-charge-region potential. Therefore, the high  $E_{fb}$  value of the composite  $\text{Co}_3\text{O}_4$ –CuO NFs provides a large driving force for the photoinduced electron–hole pairs to separate in the space charge region and then results in a high photoactivity toward phenolic compound degradation. The  $E_{fb}$  is approximately equal to the valence band potential for p-type semiconductors. Hence, the positions of the VB band for mesoporous  $\text{Co}_3\text{O}_4$  NFs and CuO NFs are 0.57 and 0.48 V, respectively. This band structure has greatly contributed toward the photocatalytic activities which are discussed in the next section.

The electrochemical impedance spectroscopy (EIS) is studied in Figure 8f. The Nyquist plots of the mesoporous  $\text{Co}_3\text{O}_4$  NFs, CuO NFs, and composite  $\text{Co}_3\text{O}_4$ –CuO NFs are showing semicircles at higher frequency level and almost a straight line inclined at a certain angle at lower frequency level. The semicircle at low frequencies features the charge transfer across the electrode/electrolyte interface, and the diameter of the semicircle represents the charge transfer resistance ( $R_{ct}$ ).<sup>65</sup> The electron charge transfer resistance ( $R_{ct}$ ) is estimated from the diameter of the semicircle. Here, the value of  $R_{ct}$  in different mesoporous NFs has the following trend:  $\text{Co}_3\text{O}_4$  NFs ( $\sim 225\Omega$ ) > CuO NFs ( $\sim 156\Omega$ ) >  $\text{Co}_3\text{O}_4$ –CuO NFs ( $\sim 40.5\Omega$ ). It has been observed that the larger is the semicircle

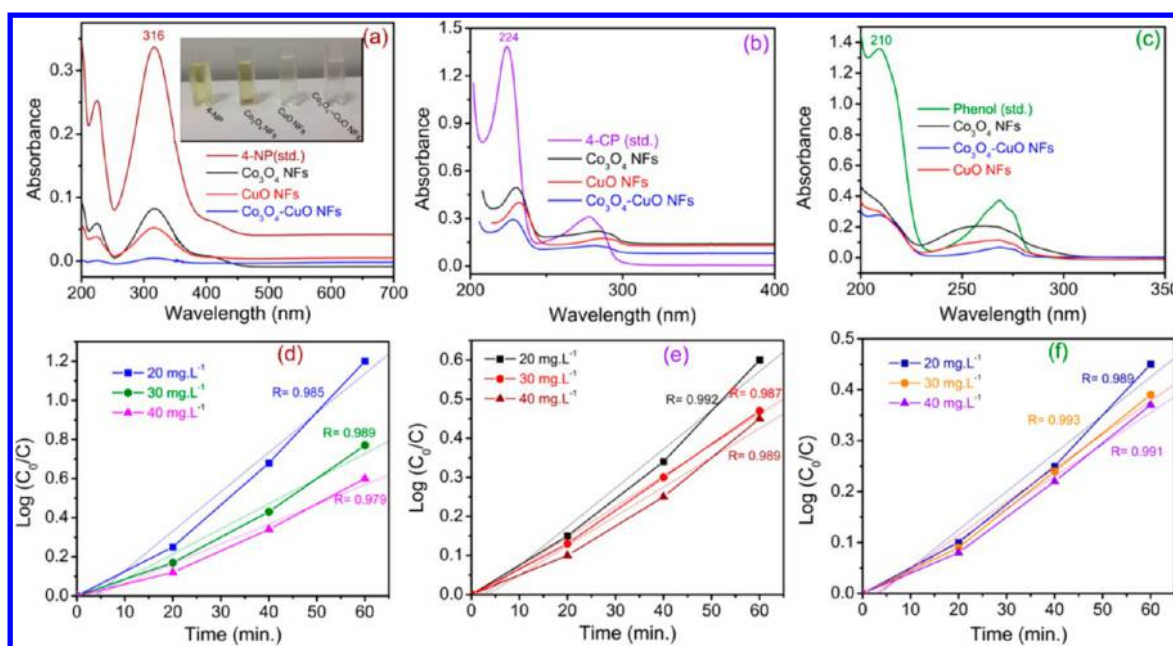


Scheme 2. Formation Mechanism of Mesoporous Composite  $\text{Co}_3\text{O}_4$ -CuO NFs in Both Chemical and Pictorial Approaches

diameter, the greater is the resistance at the interface and the lesser is the conductance. The lower  $R_{ct}$  value shows that charge transfer is more efficient at the electrode/electrolyte interface with the incorporation of the cocatalysts, thus enhancing generation of photocurrent.<sup>66</sup> That means the lower  $R_{ct}$  value of mesoporous composite  $\text{Co}_3\text{O}_4$ -CuO NFs has high charge transfer ability. The charge transfer process could be efficiently progressed from  $\text{Co}_3\text{O}_4$  to CuO, establishing the CuO as cocatalyst. Hence, the composite  $\text{Co}_3\text{O}_4$ -CuO NFs have high charge separation efficiency as compared to the constituent semiconductor NFs. Thus, the decreased value of  $R_{ct}$  increases the photocatalytic activity of the synthesized  $\text{Co}_3\text{O}_4$ -CuO NFs. Moreover, the straight line part in the lower frequency region is related to Warburg impedance ( $W_s$ ), which results from the transfer or diffusion of ions in the electrolyte.<sup>67</sup> In Figure 8f, it is clearly visible that the Warburg region is small in the cases of the mesoporous composite  $\text{Co}_3\text{O}_4$ -CuO NFs and larger in  $\text{Co}_3\text{O}_4$  NFs. The smaller Warburg region of the mesoporous composite  $\text{Co}_3\text{O}_4$ -CuO NFs as compared to single NFs indicates that the ion movement is more effective within the NFs through pores and favors the high photocatalytic performance of composite  $\text{Co}_3\text{O}_4$ -CuO NFs. Conclusively, the high current density, charge transfer, and ion movement in the composite  $\text{Co}_3\text{O}_4$ -CuO NFs due to the combination of CuO as cocatalyst and mesoporosity within NFs are responsible for the enhancement of photocatalytic performance.

**3.6. Formation Mechanism of Mesoporous Composite  $\text{Co}_3\text{O}_4$ -CuO NFs.** The possible formation mechanism of mesoporous composite  $\text{Co}_3\text{O}_4$ -CuO NFs is explained on the basis of the above experiments, which is illustrated in the Scheme 2. Scheme 2 depicts both the chemical approach and the pictorial approach. The chemical approach gives the idea about the chemical bonding interaction, whereas the pictorial

approach tells about surface arrangements. The mesoporous composite  $\text{Co}_3\text{O}_4$ -CuO NFs are fabricated on the basis of in situ electrospinning method. PVP is a highly soluble polar polymer mixed with another polar polymer PEG in isopropanol. When precursors of Co and Cu are dissolved in the mixed polymer solution and acetic acid, then the hydrolyzed Co and Cu precursors, i.e., hydroxides of Co and Cu, make hydrogen bonds with partial anionic form of oxygen present in PVP and PEG. The slow hydrolysis of metal precursors leads to strong interaction with PVP and PEG matrix, resulting in PVP/ $\text{Co}_3\text{O}_4$ -CuO/PEG (dried at 70 °C for 5 h). The mesoporous composite  $\text{Co}_3\text{O}_4$ -CuO NFs, i.e., architecture of “mesoporous-NPs- $\text{Co}_3\text{O}_4$ -CuO NFs”, are formed after calcinations of PVP/ $\text{Co}_3\text{O}_4$ -CuO/PEG at 450 °C for 4 h. The role of PVP, PEG, and acetic acid has great importance for the fabrication of the mesoporous composite  $\text{Co}_3\text{O}_4$ -CuO NFs. PVP is responsible for providing the nanofibers framework during the electrospinning process because high molecular weight long chain PVP produces the long chain hydrogen bonding with hydroxyl group of Co and Cu. PEG is a polymer of hydrophilic chain. The hydrophilicity of the PEG chain is responsible for the more significant pore formation effect.<sup>68</sup> It has been reported that the additives that are generating foam during hydrolysis aids to form mesoporous nanofibers.<sup>69</sup> In the present study, after addition of PEG, foams are generated in the solution, leading to formation of mesoporosity in the composite  $\text{Co}_3\text{O}_4$ -CuO NFs. It has been noted that the PEG is used as the capping agent for the fabrication of the nanoparticles.<sup>70</sup> Hence,  $\text{Co}_3\text{O}_4$  and CuO NPs are formed within mesoporous NFs by the role of PEG as capping agent through Ostwald ripening phenomena. The optimum concentration of the acetic acid helps to maintain the viscosity and solubilize the polymeric matrix. Conclusively,



**Figure 9.** UV–vis spectra of the solutions recorded after photocatalytic degradation of phenolic compounds such as (a) 4-NP, (b) 4-CP, and (c) phenol by different photocatalysts. The degradation process is carried out by using different 10 mg of photocatalyst in a 10 mL phenolic compounds solution at pH 6 in visible light for 70 min. The kinetic study is also performed by taking different concentrations (20, 30, and 40 mg·L<sup>-1</sup>) of phenolic compounds like (d) 4-NP, (e) 4-CP, and (f) phenol with different time intervals (20, 40, and 60 min) on mesoporous composite Co<sub>3</sub>O<sub>4</sub>–CuO NFs.

PVP, PEG, and acetic acid have great role in the fabrication of mesoporous composite Co<sub>3</sub>O<sub>4</sub>–CuO NFs or architecture of “mesoporous–NPs–Co<sub>3</sub>O<sub>4</sub>–CuO NFs”.

**3.7. Visible Light Photocatalytic Activity of Mesoporous Nanofibers.** Photocatalytic degradation of phenolic compounds such as 4-nitrophenol (4-NP), 4-chlorophenol (4-CP), and phenol took place in the presence of visible light. It is observed that mesoporous composite Co<sub>3</sub>O<sub>4</sub>–CuO NFs (Co:Cu = 1/4) have high photocatalytic degradation activity toward phenolic compounds. The reaction was carried out for 70 min in the presence of visible light, 20 mg·L<sup>-1</sup> of phenolic compounds, and 1 mg·mL<sup>-1</sup> of catalyst. The pH of the solution has a vital role for degradation of phenolic compounds. The pH is the most important factor affecting the photocatalytic degradation process. The effect of pH on the degradation of 4-NP is illustrated in Figure S7. The experiments are performed at pH values of 4, 6, 8, and 10 by using composite mesoporous Co<sub>3</sub>O<sub>4</sub>–CuO NFs. At pH 6, the degradation of 4-NP was maximum, nearly 100%, and the percentage of degradation gradually decreases as the pH increases from 6 to 10. This can be explained on the basis of the pK<sub>a</sub> value of 4-NP. The pK<sub>a</sub> value of 4-NP is 7.15. At the pH > pK<sub>a</sub>, the 4-NP dissociates; hence at the higher pH degradation decreases due to ionization of 4-NP molecules that leads to less photocatalytic degradation, while at acidic pH 6, the percentage of removal was higher because 4-NP is present in its undissociated state and the dispersion interaction predominates. Not merely 4-NP, the degradation percentage of 4-CP and phenol was higher at pH 6 due to the similar reasons as described above. In order to check the highly reactive composite Co<sub>3</sub>O<sub>4</sub>–CuO NFs having different Co and Cu ratios, the 4-NP degradation at pH 6 occurred in composite Co<sub>3</sub>O<sub>4</sub>–CuO NFs in the ratios Co:Cu = 1/4 (Co<sub>3</sub>O<sub>4</sub>–CuO NFs), Co:Cu = 1/3 (Co<sub>3</sub>O<sub>4</sub>–CuO NFs), and Co:Cu = 1/2 (Co<sub>3</sub>O<sub>4</sub>–CuO NFs). The highest degradation (100%) of 4-NP observed in composite Co<sub>3</sub>O<sub>4</sub>–

CuO NFs is shown in Table S2. The degradation of composite Co<sub>3</sub>O<sub>4</sub>–CuO NFs is highest due to the high surface area (Table S2) as compared to other Co:Cu = 1/3 and Co:Cu = 1/2. This is due to high surface area providing high active sites support for 4-NP degradation. Hence, the composite Co<sub>3</sub>O<sub>4</sub>–CuO NFs is the optimum system for the photocatalytic phenolic compound degradation.

The photocatalytic degradation of phenolic compounds such as 4-NP, 4-CP, and phenol by different mesoporous NFs is shown in Figure 9. The reaction is carried out in the presence of visible light at pH 6 for 70 min. The intensity of UV–visible absorbance spectra of 4-NP decreases gradually from mesoporous Co<sub>3</sub>O<sub>4</sub> NFs to composite Co<sub>3</sub>O<sub>4</sub>–CuO NFs as compared to standard 4-NP. The least absorbance intensity of mesoporous composite Co<sub>3</sub>O<sub>4</sub>–CuO NFs indicates high degradation of 4-NP by mesoporous composite Co<sub>3</sub>O<sub>4</sub>–CuO NFs (Figure 9a). The 4-NP color changes after degradation by different photocatalysts are pictured in Figure 9a (inset). The high degradation activity of 4-NP by mesoporous composite Co<sub>3</sub>O<sub>4</sub>–CuO NFs is due to the efficient visible light absorption and electron–hole separation properties. The efficient visible light absorption is due to the quantum confinement effect and electron–hole separation. Not only 4-NP degradation but the degradation of the 4-CP and phenol (Figure 9b,c) by different photocatalysts follows the same trend: mesoporous Co<sub>3</sub>O<sub>4</sub> NFs < CuO NFs < composite Co<sub>3</sub>O<sub>4</sub>–CuO NFs. But the % of degradation of 4-NP, 4-CP, and phenol varies with different photocatalysts (Table S3). The 4-NP has high % degradation as compared to 4-CP and phenol. This is due to the yellowish color of the 4-NP absorbing visible light more efficiently as compared to the colorless 4-CP and phenol.

The kinetics of phenolic compounds such as 4-NP, 4-CP, and phenol degradation by photocatalytic process on mesoporous composite Co<sub>3</sub>O<sub>4</sub>–CuO NFs is shown in parts d, e, and f of Figure 9, respectively. The results are plotted as



log  $C_0/C$  vs  $t$ . The kinetic study was performed by varying the concentration of phenolic compounds (20, 30, and 40 mg·L<sup>-1</sup>) with time (20, 40, and 60 min). With increase in the concentration of phenolic compounds, the degradation decreases. A linear relationship is observed between concentration of phenolic compounds and irradiation time. The photocatalytic degradation of phenolic compounds follows first order kinetics. The data are fitted to the first order rate model and the equations are given below.

$$\log \frac{C_0}{C} = Kt/2.303$$
$$K = \left( \log \frac{C_0}{C} \times 2.303 \right) / t$$

where  $K$  is the first order rate constant,  $C_0$  is the initial concentration of phenolic compounds, and  $C$  is the concentration at time  $t$ .

Trapping experiment has been performed in order to check the active species involve in the photocatalytic phenolic compounds degradation, in which the results were given in the Figure S8. For the trapping reaction, mesoporous composite Co<sub>3</sub>O<sub>4</sub>-CuO NFs is used as a photocatalyst for the degradation of the 4-NP. It has been well-known that the photogenerated holes (h<sup>+</sup>), hydroxyl radicals (•OH), and superoxide radicals (O<sub>2</sub><sup>•-</sup>) are treated as the vital reactive species involved in photodegradation reactions. It is seen that when IPA is used as scavenger, 4-NP degradation decreased less as compared to reaction performed without scavenger, indicating involvement of •OH species in the overall degradation process. The 4-NP degradation is significant when BQ is used as O<sub>2</sub><sup>•-</sup> scavenger, which indicates the involvement of species O<sub>2</sub><sup>•-</sup> in the reaction. In the Figure S8, the roles of both O<sub>2</sub><sup>•-</sup> and •OH are a little different for the degradation of the 4-NP. Moreover, the photocatalytic degradation of 4-NP decreased drastically when KI is used as scavenger. That means the photogenerated holes (h<sup>+</sup>) are the major reactive species for the degradation of 4-NP. Conclusively, h<sup>+</sup>, O<sub>2</sub><sup>•-</sup>, and •OH species play an important role for the degradation of 4-NP in visible light. The high involvement of h<sup>+</sup> in the reaction media indicates effective separation of photogenerated electron and hole. The trapping reaction supports the role of CuO as a cocatalyst.

To check the complete mineralization of 4-NP after degradation, the FTIR spectra of pure mesoporous composite Co<sub>3</sub>O<sub>4</sub>-CuO NFs, composite Co<sub>3</sub>O<sub>4</sub>-CuO NFs (after degradation), and pure 4-NP have been examined (Figure S9). The signature peaks at 1340 and 1495 cm<sup>-1</sup> are due the symmetric and asymmetric -NO<sub>2</sub> stretching vibration of 4-NP. The peaks at 1610 and 1590 cm<sup>-1</sup> correspond to the C=C valence and arene structure, respectively. The peak at 3332 cm<sup>-1</sup> is due to the O-H valence superimposed by C-H valence. The signature peaks of 4-NP are absent in mesoporous composite Co<sub>3</sub>O<sub>4</sub>-CuO NFs after degradation, indicating the complete mineralization of 4-NP during reaction. The FTIR spectrum of pure mesoporous composite Co<sub>3</sub>O<sub>4</sub>-CuO NFs is shown in Figure S9 for comparison.

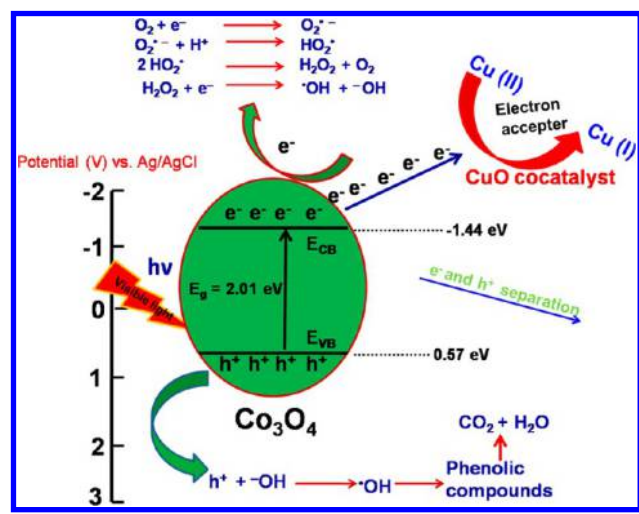
The stability study regarding 4-NP degradation by mesoporous composite Co<sub>3</sub>O<sub>4</sub>-CuO NFs is shown in Figure S10. The degradation reaction is evaluated by performing recycling experiments of 4-NP in similar conditions. The activity is found to be almost the same in three repeated runs, and then there is a slight decrease in the activity. The slight

decrease of the activity (4th run) may be due to little structural deviation in the composites Co<sub>3</sub>O<sub>4</sub>-CuO NFs which is proved by the SEM study after the fourth run (Figure S11a). The XRD of composite Co<sub>3</sub>O<sub>4</sub>-CuO NFs after the fourth run of 4-NP degradation is shown in the Figure S11b. A similar XRD result is obtained in the cases of raw composite Co<sub>3</sub>O<sub>4</sub>-CuO NFs (Figure 1b) and used composite Co<sub>3</sub>O<sub>4</sub>-CuO NFs (after the fourth run of 4-NP degradation). Conclusively, the FTIR and recycling experiment proves the complete mineralization of the 4-NP, whereas the SEM and XRD study (after 4-NP degradation) proves the stability of the photocatalyst mesoporous composite Co<sub>3</sub>O<sub>4</sub>-CuO NFs in visible light.

In order to check the morphological effect on 4-NP degradation, three kinds of composite Co<sub>3</sub>O<sub>4</sub>-CuO have been tested. These are mesoporous composite Co<sub>3</sub>O<sub>4</sub>-CuO NFs, composite Co<sub>3</sub>O<sub>4</sub>-CuO NPs, and composite Co<sub>3</sub>O<sub>4</sub>-CuO bulk. The synthesis procedure of composite Co<sub>3</sub>O<sub>4</sub>-CuO NPs and composite Co<sub>3</sub>O<sub>4</sub>-CuO bulk is given in the Supporting Information, and the morphologies are shown in the Figure S12a,b. Figure S12a shows the perfect NPs morphology of the Co<sub>3</sub>O<sub>4</sub>-CuO NPs, whereas Figure S12b shows the random bulk morphology of the Co<sub>3</sub>O<sub>4</sub>-CuO bulk. It has been observed that mesoporous composite Co<sub>3</sub>O<sub>4</sub>-CuO NFs have high percentage of 4-NP degradation (100%) as compared to composite Co<sub>3</sub>O<sub>4</sub>-CuO NPs and Co<sub>3</sub>O<sub>4</sub>-CuO bulk (Table S4). Hence, it is well understood that mesoporosity and NPs within NFs morphology have greater advantage over NPs and bulk. This is because in the case of mesoporous composite Co<sub>3</sub>O<sub>4</sub>-CuO NFs, mesoporosity gives surface active property, NPs provide quantum confinement and high surface-to-volume ratio, and NFs increase the morphological stability. The composite Co<sub>3</sub>O<sub>4</sub>-CuO NPs with only high surface-to-volume ratio property resulted in 55% 4-NP degradation. The Co<sub>3</sub>O<sub>4</sub>-CuO bulk lacks all surface active properties, which is why it shows the least 4-NP degradation (20%).

**3.7.1. Mechanism of Phenolic Compounds Degradation by Mesoporous Composite Co<sub>3</sub>O<sub>4</sub>-CuO NFs.** Suitable conduction band (CB) and valence band level of the individual semiconductors are necessary for making a suitable photocatalyst. From the Mott-Schottky plot, it is calculated that the  $E_b$  for Co<sub>3</sub>O<sub>4</sub> NFs is 0.57 V (Figure 8c) which is approximately similar to the valence band potential ( $E_{VB}$ ) of the Co<sub>3</sub>O<sub>4</sub> NFs.<sup>67</sup> So the  $E_{VB}$  of Co<sub>3</sub>O<sub>4</sub> NFs is 0.57 eV. From the UV-visible absorption spectrum, it is calculated that the band gap energy of the Co<sub>3</sub>O<sub>4</sub> NFs is 2.01 eV (Figure 5c). Hence, the conduction band potential ( $E_{CB}$ ) will be -1.44 eV. The band edge diagram of semiconductor Co<sub>3</sub>O<sub>4</sub> is shown in the Scheme 3. After visible light irradiation, the photogenerated electron (e<sup>-</sup>) and hole (h<sup>+</sup>) are formed. The h<sup>+</sup> and e<sup>-</sup> are utilized for the oxidation (•OH) and reduction (O<sub>2</sub>), respectively. But for photocatalytic system, electron and hole recombination retards the reaction efficiency. In order to solve the problem, a cocatalyst CuO is constructed that enables reduction of the e<sup>-</sup> and h<sup>+</sup> recombination. Katsumata et al. have observed that CuO has four electron reduction ability.<sup>71</sup> The high reducing nature of CuO reduces Cu(II) to Cu(I) by trapping extra electrons from the semiconductor Co<sub>3</sub>O<sub>4</sub>. This phenomenon helps to suppress the electron-holes recombination and makes CuO an effective cocatalyst. This is the vital reason that mesoporous composite Co<sub>3</sub>O<sub>4</sub>-CuO NFs show higher phenolic compounds degradation in visible light as compared to other NFs (Figure 9). The degradation mechanism of phenolic compounds occurs by the influence of electron and hole (Scheme 3).

**Scheme 3. Charge Separation and Electron–Hole Transfer in the Mesoporous Composite  $\text{Co}_3\text{O}_4$ –CuO NFs and Possible Reaction Mechanism**



The photogenerated electrons are reacting with surface molecular oxygen generating active species such as  $\text{O}_2^{\bullet-}$ ,  $\text{HO}_2^{\bullet-}$ , and  $\bullet\text{OH}$ . These oxidizing species are responsible for oxidizing/decomposing the phenolic compounds. Another way, the most reactive  $\bullet\text{OH}$  radical is formed when the hole is reacting with  $\text{OH}^-$ , which leads to oxidation/decomposition of the phenolic compounds, resulting in degradation products  $\text{CO}_2$  and  $\text{H}_2\text{O}$ . Thus, the charge separation and degradation mechanism have been well established in visible light.

**3.7.2. Factor Affecting the Photocatalytic Activity.** The photocatalytic degradation of phenolic compounds has been achieved by (a) the role of surface and textural properties which includes mesoporosity, surface hydroxyl group, NPs within NFs, and reactive facets, (b) quantum confinement and charge transfer phenomena, (c) lowering of electron–hole recombination and of oxygen vacancies, (c) role of cocatalyst, and (d) role of electrochemical study.

**3.7.2.1. Mesoporosity, Surface Hydroxyl Group, NPs within NFs, and Reactive Facets.** Mesoporous nanofibers have great deal of importance in the field of photocatalysis. These properties of materials help to increase the active site for the accommodation of a reacting molecule. It has already been examined that all mesoporous NFs show mesoporosity (Figure 1). Of the textural properties such as surface area, pore volume, and pore diameter, the mesoporous composite  $\text{Co}_3\text{O}_4$ –CuO NFs have high surface area as compared to  $\text{Co}_3\text{O}_4$  NFs and CuO NFs (Table S1), resulting in high photocatalytic phenolic compounds degradation compared with other NFs. This is due to the high interaction between the phenolic compounds with surface reactive sites of the mesoporous composite  $\text{Co}_3\text{O}_4$ –CuO NFs. Another factor is the role of particle size with the NFs. If the particle is small, i.e., below 100 nm (nanoparticles), it will act as an efficient photocatalyst because of high surface to volume ratio. Generally, the surface area increases with decreasing particle size, and if the surface area is more, then the availability of the active site will be more, leading to an increase in the catalytic activity. It has been seen from TEM studies (Figures 3 and 4) that the NPs are present within all NFs. Table S1 shows that all mesoporous NFs exhibit small size NPs, resulting in high phenolic compounds degradation in visible light. Among all mesoporous NFs, mesoporous

composite  $\text{Co}_3\text{O}_4$ –CuO NFs have high degradation activity. This is due to the small size NPs as compared to other NFs. Surface hydroxyl groups have high impact for an efficient photocatalytic application.<sup>72</sup> This is because surface hydroxyl groups are creating more  $\bullet\text{OH}$  radicals for the photocatalytic application.<sup>73</sup> Presently, from the XPS study of O 1s core-level spectrum (Figure 7), the presence of the surface hydroxyl groups in the mesoporous composite  $\text{Co}_3\text{O}_4$ –CuO NFs is proved. The generated  $\bullet\text{OH}$  radicals from the surface hydroxyl groups enhanced the phenolic compounds degradation. Recently, Ma et al. and Su et al. investigated the exposed reactive facets of the {110} and {001} for  $\text{Co}_3\text{O}_4$  and CuO, respectively.<sup>21,28</sup> In the present study, the FFT image (HR-TEM study) of the  $\text{Co}_3\text{O}_4$  and CuO showing the {110} and {001} completely matched the result from Ma et al. and Su et al. Hence, the enhanced degradation activity is also due to the exposed reacting facets of the mesoporous  $\text{Co}_3\text{O}_4$ , CuO, and composite  $\text{Co}_3\text{O}_4$ –CuO NFs. Conclusively, the mesoporosity, surface hydroxyl group, NPs within NFs, and reactive facets have great role in the enhancement of the phenolic compounds degradation by mesoporous  $\text{Co}_3\text{O}_4$ , CuO, and composite  $\text{Co}_3\text{O}_4$ –CuO NFs.

**3.7.2.2. Lowering of Electron–Hole Recombination and Oxygen Vacancies.** The PL intensities of the mesoporous composite  $\text{Co}_3\text{O}_4$ –CuO NFs are the lowest as compared to mesoporous  $\text{Co}_3\text{O}_4$  NFs and CuO NFs. Generally, the PL emission describes the recombination of excited electrons and holes under light irradiation. Hence, the lower PL intensity indicates a lower recombination rate of electron–hole.<sup>74</sup> The highest photocatalytic activity is directly proportional to the lowest recombination of electron–hole. The Schottky barrier appears to be due to the mixing of the CuO into the  $\text{Co}_3\text{O}_4$  in mesoporous composite  $\text{Co}_3\text{O}_4$ –CuO NFs, resulting in lowering of electron–hole recombination as compared to  $\text{Co}_3\text{O}_4$  and CuO NFs. Hence, the lowest electron–hole recombination of mesoporous composite  $\text{Co}_3\text{O}_4$ –CuO NFs indicates the highest photocatalytic degradation of phenolic compounds in visible light. It has been observed that oxygen vacancies help to enhance the photocatalytic application by trapping the electrons, resulting in suppression of the electron–hole recombination.<sup>75</sup> Hence, in the present study, the oxygen vacancies present in the mesoporous composite  $\text{Co}_3\text{O}_4$ –CuO NFs could act as an electron trapping center, which may be responsible for an efficient degradation activity.

**3.7.2.3. Quantum Confinement and Charge Transfer Phenomena.** Quantum confinement phenomena enhance light absorption property and quick transport of photo-generated electrons and holes. Quantum confined effect favors higher rates of interfacial charge transfer.<sup>48</sup> This phenomenon favors the high photocatalytic activity.<sup>76</sup> The quantum confinement phenomena of mesoporous  $\text{Co}_3\text{O}_4$  NFs, CuO NFs, and composite  $\text{Co}_3\text{O}_4$ –CuO NFs have been examined by SERS, UV–visible absorption spectroscopy, UV–visible–NIR DRS spectroscopy, and PL spectra. Quantum confinement arose because of the small NPs present within the aforesaid NFs. Due to quantum confinement phenomena, all three NFs show high photodegradation of phenolic compounds in visible light. Among them, mesoporous composite  $\text{Co}_3\text{O}_4$ –CuO NFs show high degradation activity than others. This is due to the small size NPs of  $\text{Co}_3\text{O}_4$ –CuO NFs, which enables harvesting of more light through quantum confinement phenomena. It has been reported that interparticle transfer of charge carriers contributes to the enhanced photocatalytic efficiency.<sup>77</sup> The



charge transfer transition favors an internal oxidation–reduction process which facilitates photocatalysis in the visible region. The mesoporous composite  $\text{Co}_3\text{O}_4$ –CuO NFs show high degradation activity which is due to the internal charge transfer between  $\text{Co}^{2+}$ ,  $\text{Co}^{3+}$ , and  $\text{Cu}^{2+}$ .

**3.7.2.4. Role of CuO as Cocatalyst in Mesoporous Composite  $\text{Co}_3\text{O}_4$ –CuO NFs.** The lowest PL spectrum (Figure 6) of mesoporous composite  $\text{Co}_3\text{O}_4$ –CuO NFs as compared to neat  $\text{Co}_3\text{O}_4$  NFs and CuO NFs proves the presence of CuO as cocatalyst. That is the vital reason that mesoporous composite  $\text{Co}_3\text{O}_4$ –CuO NFs show higher phenolic compounds degradation than neat  $\text{Co}_3\text{O}_4$  NFs and CuO NFs. As a cocatalyst, CuO enhances the photocatalytic activity of mesoporous composite  $\text{Co}_3\text{O}_4$ –CuO NFs by (i) absorbing the photoexcited electrons, as an electron acceptor, which leads to high suppression of electron–hole recombination. The lowering of electron–hole recombination increases the photocatalytic phenolic compounds degradation. It has been reported that Cu(II) species in nanoclusters are very efficient photocatalyst in visible light.<sup>78</sup> (ii) Enhancement is also done by trapping of electrons by Cu(II) center of CuO and reduction to Cu(I) in composite  $\text{Co}_3\text{O}_4$ –CuO NFs. This process is also helps to suppress the electron–hole recombination and enhanced the phenolic compounds degradation.

**3.7.2.5. Role of Electrochemical Study.** The electrochemical study favors the high phenolic compounds degradation by mesoporous composite  $\text{Co}_3\text{O}_4$ –CuO NFs as compared to the neat mesoporous  $\text{Co}_3\text{O}_4$  NFs and CuO NFs. The generation of high current density from the composite  $\text{Co}_3\text{O}_4$ –CuO NFs favors the high photocatalytic degradation through harvesting of electrons and holes. The high  $E_{fb}$  value of the composite  $\text{Co}_3\text{O}_4$ –CuO NFs as compared to neat NFs suppresses electron–hole recombination and hence increases the photocatalytic degradation of the phenolic compounds. From EIS study, a lower value of  $R_{ct}$  and smaller value of  $W_s$  in composite  $\text{Co}_3\text{O}_4$ –CuO NFs as compared to neat NFs support the high phenolic compound degradation through charge transfer, separation of electron–hole by CuO as cocatalyst, and ion movement.

## 4. CONCLUSION

Here, fabrication of electrospun mesoporous  $\text{Co}_3\text{O}_4$  NFs, CuO NFs, and composite  $\text{Co}_3\text{O}_4$ –CuO NFs was performed by using a mixed PVP/PEG polymeric template. Design and reaction of NPs and mesoporosity within NFs are a vital achievement in the present study. The mesoporosity provides the high surface area whereas NPs provide quantum confinement phenomena, which is the significance of the present investigation. The generation of the quantum confinement effect is well-established and proved by enhancing SERS signals, red shifting from optical study and blue shifting from PL study. It has been examined that mesoporous composite  $\text{Co}_3\text{O}_4$ –CuO NFs show higher photocatalytic activity for degradation of phenolic compounds under visible light as compared to mesoporous  $\text{Co}_3\text{O}_4$  and CuO NFs. The photocatalyst mesoporous composite  $\text{Co}_3\text{O}_4$ –CuO NFs are stable up to fourth runs and there is minimal change in degradation activity, proved by stability test. The FTIR study proves the complete mineralization of the 4-NP. That means the photocatalyst will be stable, reactive, and reusable for multiple runs. The high photocatalytic activity by mesoporous composite  $\text{Co}_3\text{O}_4$ –CuO NFs is ascribed to the role of low cost CuO as cocatalyst in visible light. The suggested CuO as cocatalyst is proved by the lowest PL

intensity in the composite  $\text{Co}_3\text{O}_4$ –CuO NFs which conclude that CuO acts as an electron sink/trapper, resulting in an efficient electron–hole separation. Moreover, quantum confinement phenomena from NPs within NFs, provide high visible light absorption property which enhances the degradation activity. That is the main reason that all fabricated mesoporous NFs have good degradation activity. Mesoporosity with NFs increases the surface reactivity for an efficient photodegradation of the phenolic compounds. Oxygen vacancies within NFs help to enhance the photocatalytic activity by separating electron–hole. The electrochemical properties of the mesoporous NFs support the high photocatalytic activity and prove the role of CuO as cocatalyst. Internal charge transfer phenomena between  $\text{Co}^{2+}$ ,  $\text{Co}^{3+}$ , and  $\text{Cu}^{2+}$  and exposed reactive facets {110} for  $\text{Co}_3\text{O}_4$  and {001} are also helpful for harvesting visible light efficiently for photocatalytic application. The aforesaid factors within mesoporous NFs are responsible for an efficient photocatalytic degradation of phenolic compounds under visible light.

## ■ ASSOCIATED CONTENT

### 📄 Supporting Information

The Supporting Information is available free of charge on the ACS Publications website at DOI: 10.1021/acsami.7b09026.

Table summarizing textural properties of mesoporous NFs, SEM picture of electrospun porous CuO nanoflowers, HR-TEM micrographs of mesoporous  $\text{Co}_3\text{O}_4$  NFs and CuO NFs, STEM–EDX spectra of mesoporous NFs, UV–vis–NIR DRS FTIR spectra of mesoporous NFs, Raman spectra of all mesoporous NFs, effect of pH, table summarizing % of phenolic compound degradation, trapping experiment, 4-NP degradation, and recycling study (PDF)

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A.C.P. has conducted the experiments, and A.C.P. and T.U. have both contributed to the writing of the manuscript. All authors have given approval to the final version of the manuscript.

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### Notes

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